Spectroscopic Ellipsometry

User Guide

P/N: 31 087 134



Explore the future





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HORIBA Jobin Yvon Headquarters	16-18, rue du Canal 91165 LONGJUMEAU CEDEX (France) Tel: +33 (0)1 64 54 13 00 Fax: +33 (0)1 69 09 93 19 - +33 (0)1 69 09 90 88 Internet site: http://www.jobinyvon.com
HORIBA Jobin Yvon Thin Film Division	ZA de la Vigne aux Loups 5, Avenue Arago 91380 - CHILLY-MAZARIN Tel: +33 (0)1 64 54 13 00 Fax: +33 (0) 69 74 88 61 Email: tfd-sales@jobinyvon.fr - tfd-services@jobinyvon.fr
HORIBA Jobin Yvon Inc.	3880 Park Avenue, EDISON - New Jersey 08820 (USA) Tel: +1 (732) 494 86 60 - Fax: +1 (732) 549 51 25
HORIBA Jobin Yvon GmbH	Bretonisher Ring 13 - 85630 GRASBRUNN (Germany) Tel: +49 (89) 46 23 17-0 - Fax: +49 (89) 46 23 17-98
HORIBA Jobin Yvon Ltd	2 Dalston Gardens - STANMORE Middlesex HA7 1BQ (Great Britain) Tel: +44 (20) 8204 8142 - Fax: +44 (20) 8204 6142
HORIBA Jobin Yvon Srl	Via Cesare Pavese 35/AB - 20090 OPERA (Milano) (Italy) Tel: +39 (2) 57 60 30 50 - Fax +39 (2) 57 60 08 76
HORIBA Jobin Yvon China	Room 1801, Capital Tower - No.6 Jia Jianguomenwai Avenue Chaoyang District - Beijing 100022 - P R China Tel: +86 (0) 10 8567 9966 - Fax: +86 (0) 10 8567 9066
HORIBA Jobin Yvon Japan	Higashi-kanka Daiji Building 1-7-8 Higashi-kanka, chiyoda-ku TOKYO, 101-0031 (Japan) Tel: +81 (0) 3 3861 8231
HORIBA Jobin Yvon Korea	5, Yangjae-Dong - SEOCHO-GU - SEOUL 137-150 (Korea) Tel: +82 (0) 2 753 7911

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1 Introduction

1.1 General Aspects

1.1.1 What is ellipsometry?

Ellipsometry is an optical technique for surface and thin films analysis. It is based on the measurement of the change in polarization state of a light beam caused by the reflection on the material surface or the transmission through the material. From the change in state of polarization one can deduce film thickness and optical properties of the material. The principle was discovered already more than a century ago. However, over the

past few decades the technique has progressed rapidly due to the availability of computers and thus automation of high accuracy ellipsometers. The field of applications is wide: microelectronics, display, optoelectronics, photovoltaics, chemistry, metallurgy etc. Ellipsometry is especially suited to the characterization of semiconductors where we have



massive substrates, thin film layers ranging from few nanometers to tens of micrometers, and also multi-layer structures. Furthermore ellipsometry is a fast and non-destructive technique that allows the monitoring of film growth in real time.

1.1.2 What does it measure?

Ellipsometry measures Ψ and Δ

Ellipsometry measures the change in state of polarization of a light beam caused by the reflection on the sample surface (or the transmission through the sample). The measured parameters are the so-called ellipsometric angles Ψ and Δ . They are related to the ratio of the complex Fresnel reflection coefficients r_s and r_p . Thereby r_s is the reflection coefficient for light polarized perpendicular (German: "senkrecht") to the plane of incidence and r_p is the reflection coefficient for light polarized parallel to the plane of incidence. This is expressed by the fundamental equation of ellipsometry:

fundamental equation of ellipsometry

$$\rho = \frac{r_p}{r_s} = \tan \Psi e^{i\Delta}$$

where

$$\tan \Psi = \frac{|r_p|}{|r_s|} \qquad 0^\circ \le \Psi \le 90^\circ$$

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and



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$$\Delta = \delta_p - \delta_s \quad 0^\circ \le \Delta \le 360^\circ$$

From the measured parameters Ψ and Δ the optical and structural properties of the sample can be determined by appropriate modeling. Sample parameters of interest are for example:

- optical constants in near UV, visible, and near IR wavelength ranges
- mono- or multi-layer thickness
- thickness of layers in superlattice structure
- surface and interface roughness
- anisotropy
- gradient of optical constants in single layers

Ellipsometry is a highly accurate and precise metrology technique because it measures the change in state of polarization (expressed by the ratio of reflection coefficient magnitudes and difference in phase change) rather than simply an intensity.

1.1.3 Limits of applicability

Very rough and inhomogeneous samples are difficult to analyze by ellipsometry The major limiting factors come from the state of the sample surface. Ellipsometry excludes all scattering effects and only deals with specular reflection. Therefore the roughness of the surface and also of interfaces between two media must not exceed the wavelength of the light. Otherwise non-specular scattering of the incident beam occurs which causes depolarization of the reflected light beam.

The thickness of the film must have low variation over the width of the light spot otherwise the assumption of parallel interfaces is no more valid; it means that the surface must be as plane as possible. That is why the different steps of surface preparation must not be neglected.

Assuming a perfect optical surface the minimum thickness of a film that can be determined is related to the sensitivity of the ellipsometer. This is in the sub-monolayer range.

The maximum thickness of a film that can be determined depends mainly on the wavelength range and the spectral resolution that is used. Typically films up to several tens of µm can be analyzed.

1.1.4 Procedure of ellipsometric experiments

1)Acquire 2)Model 3)Fit 4)Results An ellipsometric experiment follows always the same procedure:

- 1. an ellipsometer measurement yields the ellipsometric angles Ψ and Δ not the required sample parameters like thickness and optical constants.
- 2. It is necessary to create a model of the sample to determine the sample parameters.
- 3. Once the model is built, calculated data must be fitted to the experimental data and the best match between the two sets has to be found.
- 4. The user has to evaluate the best-fit model in order to decide whether the predictive model is physically reasonable and whether the different parameters are unique and not correlated.



SE data analysis flowchart:



1.1.5 Accuracy of ellipsometric measurements

Ellipsometry is a very sensitive and highly reproducible technique, but the absolute accuracy is hard to define Ellipsometry is a very sensitive and reproducible technique. However, it is very difficult to determine the absolute accuracy of an ellipsometer. One way is to compare results with **certified standard samples**. However, the thickness of these samples is usually certified not better than +/-5Å. This values are well above the sensitivity of an ellipsometer. Another way is to compare results with complementary methods such as AFM or RBS microscopy.

1.1.6 Final words

Ellipsometry is a very powerful but not an all mighty technique: we cannot determine the structure of a complex sample without any input information of the sample. There are some limits of applicability regarding maximum and minimum thickness and absolute accuracy. Nevertheless ellipsometry is one of the most sensitive and precise tool for thin film thickness measurements. Furthermore, ellipsometry allows the determination of the optical constants (n and k) over a broad spectral range, from near IR to UV. And this can be done all in a non-destructive way which is important for research and much more for industrial applications.



1.2 General Introduction to DP2 Software



The graph below shows the general user interface of DP2 software:



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TreeView and the Main Screen. This area displays an icon for every file which has been loaded. Thus, it becomes easy to take a look and switch between the currently loaded files. The Show/Hide *TreeView* icon is a toggle command which hides the TreeView menu. This is useful as it magnifies the working View screen. This command is a short-cut which closes the opened *TreeView Explorer* folders. 3 4 Open the Options Software screen. 5 The *TreeView Explorer* simplifies the access to the TreeView. Click on the field to activate the scroll bar, then select the node to open it. 6 The Manual Measurement icon launches the Views screen . The Views screen is in fact 021 the Real Time Acquisition panel. Depending on your System(s), an intermediate screen could ask for the acquisition to launch.

The Show/Hide icon palette is a toggle command which hides the area located below the

- 7 Creation icons: from these «short-cut» icons, the user can cre-Mat. Acq. Mod. Grd. Rec. Tem. ate a new material, acquisition routine, model, grid, recipe or report template.
- 8 Launch the *Export* files package feature.
- Launch the *Import* files package feature.
- 10 The Check Files Integrity is a very powerful feature which is recommended for experimented users only. This feature analyzes the files integrity and verify the links between built models, recipes etc...



Click on the Search feature to find a DeltaPsi2 file. The following screen will be displayed:



Activates the search

- Enter the query string to find
 - Select specific criteria

Select the searched file in the list, then click on the Go icon to select this file in the TreeView. Double-click goes directly to the file.

Close the Search screen and select the searched file in the TreeView

Close the Search screen

NOTICE: the * sign must not be used in the search query string

12	<i>TreeView</i> below:	Main Menu allows the user to access files. It contains nodes which are detailed				
	Functional node which is used to browse through the files of the software. There are three possibilities to open/close the functional node:					
		• Using the left button of the mouse, double-click the node name,				
		• Left-click on the +/- indicator in front of the icon. If this indicator does not exist, the node is empty,				
	@ @	• DeltaPsi files: each file node represents a DeltaPsi file located on the hard disk. To open the file, double-click with the left button of the mouse. Each type of file is marked on the icon: i.e. SPE for spectroscopy, REF for reference				
	?	• Unknown file: this icon signifies that an unknown file has been found.				
	NOTICE : Right-clicking on an element name will display a popup menu. This menu contains specific options related to the node. Do not hesitate to try features included in this menu.					
13	Shell Title tion) and t	<i>Bar</i> : includes the <i>Print</i> command, the <i>About</i> button (DeltaPsi version informathe <i>Help</i> button which is not active.				
14	🕐 Helj	p online is available by clicking the icon located on the <i>Button Bar</i> .				
<mark>16</mark>	16 Views screen: this is the main working area. This screen is in fact an independent window which can be resized using the following icon located on the upper right hand side of the screen:					
	Window is minimized and reduced to simple title bar					
	Back to full size screen					
		If several windows have been opened, these two buttons switch between the windows.				



Additional options can be accessed by clicking on the icon located on the upper left hand side of the window. Depending on your size status, some options cannot be activated (greyed). Here below is an example of such of screen:





2 Basic Principles of Optics

2.1 The electromagnetic wave according to Maxwell equations

An electromagnetic wave is a transverse wave consisting of an electric field vector and a magnetic field vector. The magnitude of both vectors is a function of position and time. These two vectors are mutually perpendicular and both are perpendicular to the direction of propagation. The frequencies are ranging from ca. $3 \ 10^{11}$ to $3 \ 10^{16}$ Hz. The important quantity describing the light wave is the electric field vector. The magnetic field vector is not important because the induced magnetization in the material can be neglected since the motion of magnetic moments of electrons and nuclei is too slow to follow the rapid optical oscillations.

The electromagnetic field is described by Maxwell's equations:

Maxwell equations

$$\vec{\nabla} \cdot \vec{E} = 0$$
$$\vec{\nabla} \cdot \vec{B} = 0$$
$$\vec{\nabla} \times \vec{E} + \frac{1}{c} \frac{\partial \vec{B}}{\partial t} = 0$$
$$\vec{\nabla} \times \vec{E} - \frac{\mu \varepsilon}{c} \frac{\partial \vec{E}}{\partial t} = 0$$

where E and B are the electric and magnetic field, c is the speed of light, μ is the permeability and is the dielectric function.

Combining these equations leads to the electric field wave equation:

$$\vec{\nabla}^2 \vec{E} - \frac{1}{\upsilon^2} \frac{\partial^2 \vec{E}}{\partial t^2} = 0$$

with the optical impedance

$$\upsilon = \frac{c}{\sqrt{\mu\epsilon}}$$

The electromagnetic plane wave is a solution of the electric field wave equation

$$\vec{E}(\vec{r},t) = \vec{E}_0 \exp\left(\frac{i2\pi \cdot \widetilde{n}}{\lambda} \vec{q}.\vec{r}\right) \exp\left(-i\omega \cdot t\right)$$

where \vec{r} is the position vector along the direction of propagation, t is the time, the vector q is a unit vector along the propagation of the wave, $\tilde{n} = n + ik$ is the complex index of refraction, is the angular frequency of the wave, is the wavelength and is \vec{E}_0 a complex vector constant specifying the amplitude and polarization state of the wave. A plane wave propagating along the z-axis of an orthogonal coordinate system can be described by:



2.2 Interaction of light with matter: complex index of refraction

When a light beam hits an interface between two media several phenomena can be observed. The wave in general slows down, changes direction, can be absorbed and/or reflected by the second medium. Optically the media are described by their complex index of refraction \tilde{n} and their thickness.

The complex index of refraction $\tilde{n} = n + ik$ consists of the index of refraction n and the extinction coefficient k. It describes the interaction of light and matter. Both, n and k, are functions of the wavelength. The index of refraction describes the phase velocity of light in a material. The extinction coefficient describes how the intensity decreases as the light passes through the material.



depth penetration

The decrease in intensity in an absorbing medium can be described by the following law:

$$I = I_0 exp(-\alpha \cdot z)$$

k: coefficient of absorption

n: index of refraction



With the absorption coefficient α :

$$\alpha = \frac{4\pi \cdot k}{\lambda}$$

The penetration depth is defined by:

$$D_{p} = \frac{\lambda}{4\pi \cdot k}$$

When the thickness of the film is greater than ca. 4x the penetration depth the film can be treated as a substrate because too less light reflected from the interface reaches the detector. In this case the thickness cannot be determined, only the optical constants n and k.

There are mainly three processes causing the interaction of light with matter:

- 1. electronic transitions
- 2. molecular or lattice vibrations
- 3. free carrier absorption





2.3 Reflection and Refraction of Light

2.3.1 Law of reflection

The angle of reflection θ_r is equal to the angle of incidence θ_i :

$$\theta_r = \theta_i$$

2.3.2 Law of refraction:

The light that is transmitted through the interface changes its direction according to the Snell-Descartes law:

$$\widetilde{\mathbf{n}}_0 \sin \theta_0 = \widetilde{\mathbf{n}}_1 \sin \theta_1$$

The knowledge of this law will allow us to determine the Fresnel coefficients.



2.4 Polarization

Polarization refers to the behaviour with time of the electric field vector observed at a fixed point in space.

2.4.1 Linearly polarized light

Light with its electric field oriented in a constant transverse direction - although its magnitude and sign vary with time - is called linearly polarized. The transverse direction could be for example the x- or y- direction or any other direction perpendicular to the direction of propagation (z). In general any linearly polarized light can be described by the superposition of components polarized in the x-direction and in the y-direction. The amplitudes can be different however they must be in phase.





2.4.2 Circularly polarized light

If the amplitudes of the two components are equal and their relative phase difference is for example equal to /2 then the light is circularly polarized.

If the phase difference is equal to -/2 + 2m (m=0, $\pm 1, \pm 2,...$) then the electric field vector is rotating clockwise at the frequency of the light as seen by an observer toward whom the wave is moving. This is called right-circularly polarized light. The electric field vector makes one complete rotation as the wave advances through one wavelength.

If the phase difference is equal to +/2 + 2m ($m=0, \pm 1, \pm 2,...$) then the electric field vector is rotating counterclockwise at the frequency of the light as seen by an observer toward whom the wave is moving. This is called *left-circularly polarized* light.



2.4.3 Elliptically polarized light

In general the two components of the light wave have different amplitudes and are out of phase. The result is elliptically polarized light. The tip of the electric field vector describes an ellipse in the projection on the x-y plane.

The most common components to describe the state of polarization are the component along the x-direction and the component along the y-direction. However, in general any two orthogonal polarization states can be used to describe the polarization of the light wave. For example also the two oppositely polarized circular waves can be used.

Elliptically polarized light can be formed for example by the reflection of linearly polarized light on a surface.



So usually in ellipsometry we see elliptically polarized light after reflection on the sample at the detector when the incoming light is linearly polarized:



2.4.4 Partially polarized light

The electric fields of photons emitted from an incandescent light source are oriented in all different directions. This is called *unpolarized light*. In general the electric field varies in a way that is neither totally regular nor totally irregular, it is *partially polarized*. This partial polarization can be expressed by the *degree of polarization*.

2.5 The Fresnel coefficients

Assuming a plane wave arrives at the sample surface, one part of the wave is reflected and the other part is transmitted. The electric field vector can be decomposed in two components as mentioned above, in a component E_s perpendicular to plane of incidence (s – for German "senkrecht") and in a component E_p parallel to the plane of incidence. Both

$$r_{p} = \frac{E_{p}^{r}}{E_{p}^{i}} = |r_{p}| \exp(i\delta_{p}) \qquad r_{s} = \frac{E_{s}^{r}}{E_{s}^{i}} = |r_{s}| \exp(i\delta_{s})$$



The modules of the coefficients give the amplitude modifications of the electric field components. Their phases give the phase shift caused by the reflection.

The ratio ρ of these two coefficients is exactly what an ellipsometer measures:

Fundamental equation of ellipsometry

with

$$\rho = \frac{r_p}{r_s} = tan\Psi e^{i\Delta}$$

$$\tan \Psi = \frac{|\mathbf{r}_{\rm p}|}{|\mathbf{r}_{\rm s}|}$$
$$\Delta = \delta_{\rm p} - \delta_{\rm s}$$

and $\in [0;90^{\circ}] \& \in [0;360^{\circ}].$

The parameters Ψ and Δ are called the *ellipsometric angles*. The analysis of Ψ and Δ allows to obtain optical properties of the sample.

2.6 Relation of Fresnel coefficients to material properties

For a homogeneous sample the Snell-Descartes equation is

$$\widetilde{n}_0 \sin \theta_0 = \widetilde{n}_1 \sin \theta_1$$

The complex Fresnel coefficients

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where θ_0 and θ_1 are the angle of incidence and refraction and is \tilde{n}_1 the refractive index of the sample and \tilde{n}_0 f the surrounding material. By using this equation and the properties of the electromagnetic field at an interface one obtains the Fresnel coefficients for reflection and transmission:

reflection

$$\begin{split} r_{p} &= \frac{\widetilde{n}_{1}\cos\theta_{0} - \widetilde{n}_{0}\cos\theta_{1}}{\widetilde{n}_{0}\cos\theta_{0} + \widetilde{n}_{1}\cos\theta_{1}} \\ r_{s} &= \frac{\widetilde{n}_{0}\cos\theta_{0} - \widetilde{n}_{1}\cos\theta_{1}}{\widetilde{n}_{0}\cos\theta_{0} + \widetilde{n}_{1}\cos\theta_{1}} \end{split}$$

transmission

$$t_{p} = \frac{2\widetilde{n}_{0}\cos\theta_{0}}{\widetilde{n}_{1}\cos\theta_{0} + \widetilde{n}_{0}\cos\theta_{1}}$$
$$t_{s} = \frac{2\widetilde{n}_{0}\cos\theta_{0}}{\widetilde{n}_{0}\cos\theta_{0} + \widetilde{n}_{1}\cos\theta_{1}}$$

In this simple case of a semi-infinite sample without top layer it is possible to calculate the dielectric function of the material directly from the ratio $\rho = r_p/r_s$:

$$\widetilde{\epsilon} = \widetilde{n}_1^2 = \widetilde{n}_0^2 sin^2 \theta_0 \left[1 + \left(\frac{1 - \rho}{1 + \rho} \right)^2 tan^2 \theta_0 \right]$$

2.7 The Brewster angle

The *reflectance* R is calculated from the reflection coefficients r by:

 $R^{s} = |r^{s}|$ and $R^{p} = |r^{p}|$

If we plot the reflectance *R* versus the angle of incidence (AOI) we see that R^p shows a minimum for a certain AOI and gets equal to 0 for a certain AOI in the case of a dielectric (k =0) whereas R^s increases with AOI.

The AOI where R^p shows a minimum is called the *principle angle* and in the case of dielectric materials ($R^p = 0$) the *Brewster angle*.





The Brewster angle can be calculated from the indices of the two media by:

$$\tan \theta_{\rm B} = \frac{n_1}{n_0}$$

The phase retardation of the p-wave shifts abruptly from zero to 180° at the Brewster angle for dielectrics.

The Brewster angle for some materials is shown in the next graph:



2.8 Reflection with films



In the case of a deposited film on a substrate, the part of light which is transmitted cannot be ignored as it is the case for a bulk material because it is reflected on the film-substrate interface.

The resulting reflected wave returning to the medium 0 is composed of light which is reflected from the first interface as well as light transmitted by the first interface, reflected by the second one and then transmitted by the first one, and so on. Assuming the previous derivation for the Fresnel reflection and transmission coefficients of a bulk material is locally valid for the reflection and transmission of a beam from any interface, we can now calculate the Fresnel coefficients which are functions of the optical properties of the film and the ambient. The calculation of the coefficient for the p-polarization is independent from the one for the s-polarization; any arbitrary incident light beam can be described as a linear combination of the two polarization states. For a single film, the ratio of the amplitude of the outgoing resulting wave to the amplitude of the incoming light beam is defined as total reflection coefficient:

Total reflection coefficients

$$r_{\text{total}}^{p} = \frac{E_{p}^{r} t_{\text{total}}}{E_{p}^{i}} = \frac{r_{01}^{p} + r_{12}^{p} \exp(-i \cdot 2\beta)}{1 + r_{01}^{p} r_{12}^{p} \exp(-i \cdot 2\beta)}$$
$$r_{\text{total}}^{s} = \frac{E_{s}^{r} t_{\text{total}}}{E_{s}^{i}} = \frac{r_{01}^{s} + r_{12}^{s} \exp(-i \cdot 2\beta)}{1 + r_{01}^{s} r_{12}^{s} \exp(-i \cdot 2\beta)}$$

where

Fundamental equation of ellipsometry

 β is the phase change from the top of the film to the bottom of the film

$$\beta = \left(\frac{2\pi \cdot d}{\lambda}\right) \quad \widetilde{n}_1 \cos\theta_1$$



2.9 Effect of film thickness

For normal incidence the total reflection coefficient for a single film system is give by:

$$r_{total} = \frac{r_{01} + r_{12} \exp(-i \cdot 2\beta)}{1 + r_{01} r_{12} \exp(-i \cdot 2\beta)}$$

When the thickness and wavelength are such that the light reflecting from the top surface and the light reflecting from the interface layer-substrate or layer1-layer2 are 180° out of phase, which is the equivalent to $2\beta = \pi$, the exponential term of the reflectance is -1 and we have the greatest amount of destructive interference. It results in the first order minimum on the reflectance spectrum.

When $2\beta = 2\pi$, the value of the exponential term is +1, the reflectance is the same as the reflectance of a film free substrate and we have a first order maximum.

The minima occur for $2\beta = \pi$, 3π , 5π , 7π ... and the maxima occur for $2\beta = 2\pi$, 4π , 6π , 8π ...

For the following illustration we plotted the reflectance of a 1000Å thick SiO_2 layer on a substrate whose index of refraction "n" is 3.87 and whose extinction coefficient "k" is 0.0165.



If we increase the thickness of the SiO_2 layer several higher-order maxima and minima appear successively. The reflectance spectrum of a 3000Å SiO_2 layer on the same substrate as above is shown in the next figure:





We can make the following observations:

- the spacing between maxima and minima increases with wavelength
- increasing the thickness causes all the positions of the maxima and minima to move toward a longer wavelength
- thicker films provide more maxima and minima than thinner films for any wavelength range

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3 Ellipsometry Basics

3.1 Definition Ellipsometry

Ellipsometry measures the change in state of polarization in reflection and transmission experiments.

3.2 Fundamental equation of ellipsometry

Fundamental equation of ellipsometry

The relation between ellipsometric angles Ψ and Δ with the reflection coefficients for parallel and perpendicular polarization is given by:

$$\rho = \frac{r^{\rm p}}{r^{\rm s}} = \tan \Psi \cdot e^{i\Delta}$$

This equation is called the fundamental equation of ellipsometry.

3.3 Dependence on Sample Properties (single wavelength)

In this chapter Psi-Delta curves are shown as functions of substrates, film thickness and optical constants.

3.3.1 Substrates

For pure substrates the measured Psi-Delta values must be in the following intervals: $0 \le \Psi \le 45^{\circ}$ $0 \le \Delta \le 180^{\circ}$





This is shown for some examples in the following graph:

3.3.2 Single films: thickness variation

When growing a thin film on a substrate the $\Psi-\Delta$ trajectory describes a closed curve if the film is transparent. This is shown for SiO₂ on Si substrate (AOI: 70°, 633nm):



After one «period» (P) the curve reaches the starting point. The period depends on wavelength, AOI and refractive index of the material:

$$P = \frac{\lambda}{2\sqrt{n_1^2 - n_0^2 \sin(AOI)}}$$

For SiO₂ on Si substrate (AOI = 70°, 633nm) the period is ca. 280nm. So it is actually not possible to distinguish a SiO₂ layer of a certain thickness t and a layer of thickness t + nP. For example the following SiO₂ layers would give the same - values: 20nm, 300nm, 580nm, ...

This is called the «period problem» of single wavelength ellipsometry.

The period does not depend on the substrate. For example the next graph shows $\Psi - \Delta$ trajectories of SiO₂ on different substrates. Although the curves look quite different the period length is always the same.



3.3.3 Single films: variation in n

The dependency of the Ψ - Δ trajectories on the refractive index n is shown in the next graph. As long as k = 0 the curves are closed with a period that depends on n.





3.3.4 Single films: variation in k

If k > 0 the $\Psi - \Delta$ trajectories are not closed any more they converge into the $\Psi - \Delta$ values that represent the film as pure substrate:



For films thicker than the ca. 4x the penetration depth it is not possible any more to determine the thickness of the film. It can be regarded as a substrate. That means that one can determine n & k of this film directly from Ψ - Δ values.

Remember the penetration depth is defined by:

$$D_{p} = \frac{\lambda}{4\pi \cdot k}$$

3.4 Spectroscopic information

The spectroscopic information eliminates the period problem because the period is wavelength dependant. The following graph shows the Ψ -values of SiO₂ films on Si substrate (red: 50nm, blue: 330nm, green: 610nm) at AOI = 70°:



One can see that the curves coincide at 633nm but are different at the other wavelengths. Again, the thicker the layer the more "interference fringes" appear.

3.5 Different ellipsometer types

The different ellipsometers have components in common

There are three major techniques to measure the ellipsometric angles Ψ and Δ :

- null ellipsometry
- rotating element analyzer
- polarization modulation ellipsometer

All types of ellipsometers consist of a light source of some sort, a device that defines the state of polarization before the light strikes the sample and a device that determines the state of polarization after reflection.

3.5.1 Null ellipsometer

This kind of ellipsometer is based on the principle of signal extinction. The azimuths of a polarizer and an analyzer are manually adjustable to obtain the configuration where the light intensity that is detected is minimized.

3.5.2 Rotating element ellipsometer

This is a very common type of ellipsometer. One of the polarizing elements (polarizer, analyzer or compensator) rotates at a frequency of ca. 10 to 300Hz. This leads already to an improvement of acquisition time compared to null ellipsometers but acquisition is still rather slow. The range of frequencies is prone to noise caused by mechanical vibrations, requiring an elaborate signal treatment.

Rotating polarizer (RP) or analyzer (RA)



Rotating polarizer (RP) or analyzer (RA) with compensator (C)







3.5.3 Polarization modulation ellipsometer (PME)

The PEM allows very fast measurements This ellipsometer shows optimized acquisition speed and a large spectral range from the IR to the UV. The high acquisition speed is due to the presence of a photoelastic modulator (PEM) whose acquisition frequency is around 50kHz. The drawbacks of this ellipsometer are the chromatic behaviour of the modulator and the sophisticated optical calibrations. This type of ellipsometer allows extremely accurate results, thanks to a high performing electronic system.



3.5.4 Summary

Technique	RA / RP	RA / RPRA / RP and compensatorRC		РМЕ	
Modulation frequency	10 - 300Hz	10 - 300Hz	10 - 300Hz	50kHz	
Measured parameters	tanΨ - cos∆	$ an\Psi$ - $ an\Delta$	$ an \Psi - an \Delta$	sin2Ψ - cosΔ sin2Ψ - sinΔ cos2Ψ	

Strength	Simple design Easy calibration Large spectral range No wavelength dependent optics CCD detection possible	High accuracy also at Δ≈ 0° or 180° (thin transparent films)	High accuracy CCD detection possible	No rotating elements Wide spectral range High accuracy for all Ψ and Δ Fast acquisition: 1ms
Weakness	Not accurate for ∆ ≈ 0° or 180° Errors due to residual polarization Moving part	Complicated calibration CCD detection not possible Moving part	Difficult technology Difficult calibration Moving part	Difficult technology Difficult calibration CCD detection not possible

3.5.4.1 Detailed description of the Polarization modulation (PM) ellipsometer

The layout of a classical PM ellipsometer is as follows:



Elliptically modulated polarized light

Source

The typical light source is a Xenon arc lamp of 75 Watt with a spectral range from the near IR to the UV.

Optical fibers

The optical fibers are used to couple the light beam from the output of the light source to the input of the polarizer and from the output of the analyzer to the input of the monochromator. Generally the optical fibers have a 1mm core diameter and are of two types:

- UV fibers which cover the range 190 880nm
- NIR fibers which cover the range 260nm 2100nm

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Explore the future

Polarizer

The polarizer defines the state of polarization before the light strikes the sample.

Photoelastic Modulator (PEM)

The polarization modulator induces a modulation of the state of polarization that means that the ellipticity of the polarization varies as a function of time.

The modulator used in the spectroscopic ellipsometers of HORIBA JOBIN YVON is based on the photoelastic effect: Stress is applied to a silica bar so that its optical properties are modified. In its equilibrium state, the modulator is optically isotropic with one index of refraction, and becomes birefringent (with two indices) under mono-axial stress. A cosine variation of the stress modulates the birefringent state of the silica bar (resonance frequency $f = \Omega/2\pi \approx 50$ kHz). The phase retardation between the two components of the electric field is:

$$\delta(t) = A \sin \Omega t$$

whereby

$$\mathbf{A} = \frac{2\pi \cdot \mathbf{d}(\mathbf{n}_1 - \mathbf{n}_0)}{\lambda}$$



The applied stress is wavelength dependent. It is regulated with a modulation voltage, so that the modulation amplitude is kept constant.

The modulator temperature must be controlled and kept constant. The advantages of the photoelastic modulation are: very fast acquisition, high accuracy and working frequency far from ambient noise sources.



Analyzer

The analyzer is similar to the polarizer of the entry and is located just after the sample and photoelastic modulator.

Monochromator

The monochromator separates the light into its various components before it reaches the detector.

Detector

The detection system is composed of various elements:

- photomultipliers for the UV and the visible spectral range
- a InGaAs detector for the IR spectral range

Acquisition interface

Except for the sample placement, the operation of the ellipsometer is entirely automated. The computer interface owns command modules to control:

- step motors for azimuths of analyzer, polarizer and modulator
- goniometer for the angle of incidence
- amplitude of the photoelastic modulator
- shutters and filters
- high voltage of the photomultipliers
- slits of the IR monochromators



4 Measurement QuickStart

Preparation of System

- Switch on the controller unit and the computer where the *DeltaPsi2* software is installed.
- Turn on the *light source* (usually a Xe arc lamp) and wait 15min to let it stabilize.
- Open the *DeltaPsi2* software and click on the "software options" tab:

The following window will appear:

Software Options				X		
User Library	User Library file folders Current User Library profile : • • Sort by name • • Sort by creation order					
Application Library	Default					
	👷 Create a new j	orofile	Delete the current profile			
File name formats	Folders :					
	Acquisition Routines	C:\DeltaPsi2\L	Jser Library\Default\Acquisition Routines\	^		
File explorer	Data Manipulation	C:\DeltaPsi2\L	Jser Library\Default\Data Manipulation\			
	Substrates	C:\DeltaPsi2\L	Jser Library\Default\Substrates\			
Tool bar	Materials	C:\DeltaPsi2\User Library\Default\Materials\				
	Samples	C:\DeltaPsi2\User Library\Default\Samples\				
Prograss screens	Models	C:\DeltaPsi2\User Library\Default\Models\				
Trogress screens	Multimodels	C:\DeltaPsi2\User Library\Default\Multimodels\				
Onlands d	Modeling Scenarios	C:\DeltaPsi2\L	Jser Library\Default\Modeling Scenario\			
calculated	Grids	C:\DeltaPsi2\L	Jser Library\Default\Grids\			
observables	Recipes	C:\DeltaPsi2\L	Jser Library\Default\Recipes\			
Acquisition	Report Templates	C:\DeltaPsi2\User Library\Default\Report Templates\				
parameters	Acquisition Data	C:\DeltaPsi2\L	Jser Library\Default\Acquisition Data\			
Observables	Model Results	C:\DeltaPsi2\User Library\Default\Model Results\				
precision	Multimodel Results	C:\DeltaPsi2\U	Jser Library\Default\Multimodel Results\			
Modeling output	Modeling Scenario Results	C:\DeltaPsi2\L	Jser Library\Default\Modeling Scenario Results\			
- modeling output	Recipe Results	C:\DeltaPsi2\L	Jser Library\Default\Recipe Results\			
	Model Reports	C:\DeltaPsi2\L	Jser Library\Default\Model Reports\	~		
		<u>.</u>				

- Choose "user library" and choose an existing user library profile in the drop-down menu or "Create a new profile...".
- Close the "software options" window and click the "manual measurement" tab to open the "Views" screen:

Sample Positioning

- Place your *sample* on the *stage* and switch on the *autocollimator illumination*.
- Look through the eyepiece of the autocollimator and align the crosshairs so that the centers overlap by adjusting the two knobs underneath the stage.
- In the views screen set the monochromator to 2.75eV (or 450nm if you prefer to work with wavelength).
- Adjust now the height of the stage by turning the corresponding knob underneath the stage. Try to find the position where S0 is at its maximum
- In the views screen click "S0 adjustment" or manually set the high voltage so that S0 is between 60 and 80mV.

DeltaPsi2 Software: Acquisition Interface

• Click on the «Create a new acquisition routine...» tab:


• The following options will appear (they will be explained for scanning systems – for MWL systems the main points are mostly identical):

Create new ac	quisi	tion routine	X
User Lib	rary	Application Library	
Ellipsometer	Refle	ctometer LCE	
چ	100	Ellipsometer Calibration	
5	109	Spectroscopic Mono	
5	icq	Spectroscopic MWL	
5	icq.	Kinetic Mono	
5	100	Kinetic MWL	
5	100	R&T Mono	
<u>چ</u>	100	R&T MWL	
		X Cancel	

They allow you to choose the data type you wish to acquire.

- > Ellipsometric Calibration: For calibration of your system
- **> Spectroscopic Mono:** Choose this option if you want to acquire ellipsometric data over a certain wavelength range.

Spectroscopic mono acquisition setup Graph setup Graph			
Acquisition si	mulation	–Spectrum range	a
Configuration ch	oice	User unit	eV 🔹
No merge	•	Start :	1.500 eV
II: M=0° A=+	45° 🔻	Grad.	6,000
End: 6.000 eV			
		Increment :	0.050 eV
Mode: Standard			
o 111 - 11 - 1	No dete	Mode : Standa	ard 🗾
Calibration index	No data	Mode: Standa	ard 💌
Calibration index	No data	Mode: Standa	ard 💌
Calibration index Micro-Spot :	No data	Mode: Standa	ard 💌
Calibration index Micro-Spot : Background :	No data	Mode: Standa	ard 💌
Calibration index Micro-Spot : Background : Protection UV :	No data	Mode : Standa	ird 💌
Calibration index Micro-Spot : Background : Protection UV :	No data	Mode : Standa	ird 💌
Calibration index Micro-Spot : Background : Protection UV :	No data	Mode : Standa	ard •
Calibration index Micro-Spot : Background : Protection UV : High Voltage :	No data	Mode : Standa	e: 70.00 *
Calibration index Micro-Spot : Background : Protection UV : High Voltage : SO Min :	No data	Mode : Standa	ard ▼ =: 70.00 • :: 1 × 200 ms

> Kinetic Mono: Choose this option if you want to acquire ellipsometric data as a function of time.

Kinetic Acquisition Routine Mono - Non	ame1.ack
Kinetic mono acquisition setup Result	
Kinetic parameters	Acquisition conditions
Sampling Time : 0_s	Configuration : II: M=0° A=+45
Integration Time : 200 ms	
Total Duration	Inclaence angle : 70.00
Points Number : 10	Micro-spot :
Time: 0 h 0 min 0 s	Micro-spot.
Acquisition result format	Background : Background on
Binary format	Protection UV : Off
C Text format	
Spectral position	High voltage : Automatic 💌 250 V
Ilser unit	S0 Min : 60 mV
	S0 Max : 80 mV
Position : 1.500 eV	

> R & T Mono: Choose this option if you want to acquire reflection or transmission data over a certain wavelength range.

👜 Acquisition Rou	tine - Noname1.acq	
R&T mono acquisi	tion setup Graph	
Calibration index: Spectrum range User unit : Start : End : Increment :	No data ▼ eV ▼ 1.500 eV 6.000 eV 0.050 eV	Mode : Transmission Transmission Mode: Unpolarized New DC signal reference Drag-and-drop a reference file here :
Micro-Spot : Background :	▼ Background on ▼	
Protection UV :	Off	
High Voltage : SO Min :	Fixed 250 V 60 mV	Incidence angle : 90.00 • Integration time : 1 X 200 ms
SO Max :	80 m∨	Accumulation : 1

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5 Advanced Acquisition

5.1 Introduction

The UVISEL system is able to acquire data basically from any given sample. Moreover besides ellipsometric data we can also acquire reflectance and transmittance data.

If you click on the Acq. Tab on top of the page:

★ ★ ★ ★ ★ ★ Mat. Acq. Mod. Grd. Rec. Ten

The following window will appear:

Create new acquisition routine		
User Library	C Application Library	
Ellipsometer Refle	ectometer LCE	
acq	Ellipsometer Calibration	
(poo	Spectroscopic Mono	
(Dag	Spectroscopic MWL	
(Dec	Kinetic Mono	
(Dag	R&T Mono	
(Dag	R&T MWL	
	X Cancel	

The different options for the UVISEL ellipsometer are:

Ellipsometric Calibration: system calibration

Spectroscopic Mono: acquisition of ellipsometric data in reflection and/or transmission with a scanning monochromator

Spectroscopic MWL: acquisition of ellipsometric data in reflection and/or transmission with detection at fixed multiple wavelengths

Kinetic Mono: acquisition of ellipsometric data in reflection and/or transmission at a specified wavelength as a function of time

R¢**T Mono**: acquisition of reflectance and/or transmittance data with a scanning monochromator

R¢*T MWL*: acquisition of reflectance and/or transmittance data with detection at fixed multiple wavelengths

- spectral range
- angles of incidence
- spectral step size: equal increments in wavelengths or energy?
- detector settings
- spot size
- measurement parameters (experimental data types)

This parameters will be discussed in detail for the most important acquisition mode, the spectroscopic mono, but most of the information is also relevant for the other acquisition modes.

5.2 Sample preparation

Before the data acquisition is started the sample has to be prepared and positioned on the sample support.

5.2.1 Surface cleaning

Ellipsometry is very sensitive to the microscopic surface state whereas it is rather insensitive to macroscopic particles: large objects only scatter the light away from the detector. But surface roughness or oxidation layers may have strong effects on the specularly reflected beam.

Overlayers including native oxides or surface roughness can be removed by chemical or mechanical treatment. The user should refer to a library of surface treatments for the considered material.

If the overlayer cannot be removed it can be included in the layer model.

5.2.2 Back surface reflections

In the case of transparent samples that are polished on both sides the light beam reflected from the backside reaches the detector and affects the measurement.





One way to avoid this backside reflection is **roughening** the backside surface. Roughening provides a scattering effect and a non-specular surface which prevents the penetrating beams to be reflected from the back.

Backside roughening can be applied for example to samples based on plastic substrates especially when these are optically anisotropic.

Roughening can be achieved with sandpaper assuming that it does not leave residue such as inclusions in the sample.

Note that backside reflections are not necessarily a problem. In some particular cases (thin transparent substrates with both sides polished) the backside reflections can be included in the layer model.

Assuming the optical properties of the transparent material are isotropic we can distinguish three cases:

- the case of substrates whose thickness is < 1 mm:

The detection head will collect several incoherent reflected beams; the exact amount of them is not critical as far as there are at least two collected beams. The modeling will assume infinite number of backside reflected beams. The problem can therefore be solved by proper modeling.

- the case of substrates whose thickness is between 1 and 2 mm:

In this case the backside reflected beams tend to be separated. The backside reflected beam can be suppressed mechanically with an appropriate spot size and angle of incidence. If this is not possible the only solution is to roughen the backside.

- the case of substrates whose thickness is > 2 mm:

In this case, the back surface beams are sufficiently separated so that the backside reflected beam can be clearly suppressed mechanically with an appropriate spot size and angle of incidence.

5.2.3 Sample positioning

The levelling of the sample is performed with the **Autocollimator**. Adjust the screws of the stage so that crossing points of both reticules coincide:



5.3 Ellipsometric calibration acquisition mode

This mode is used for calibration of system devices. It is used only in extraordinary situations when the ellipsometer needs to be fully calibrated. For this type of calibration, it is advised to ask for the assistance of HJY Service Engineers. Do not perform this calibration for routine measurements.

Calibration Routine - Noname1.acq	
✓ System initialization	 Full ellipsometer calibration Monochromator
Wavelength monochromator calibration	Multichannel
Wavelength multichannel calibration	
Polarizer & analyzer zero setting	
PMT gain calibration	Fast ellipsometer calibration
Background calibration	Multichannel

5.4 Spectroscopic Mono acquisition mode



Spectroscopic mono acquisition setup Graph se	etup Graph	
Acquisition simulation Configuration choice No merge II: M=0° A=+45°	Spectrum range User unit : Start : End : Increment :	eV ▼ 1.500 eV 6.000 eV 0.050 eV
Calibration index: No data	Mode : Standard	•
Micro-Spot :		
Background : Background on 💌		
Protection UV : Off		
High Voltage : Automatic 💌 250 V	Incidence angle :	70.00 •
SO Min : 60 mV	Integration time :	1 X 200 ms
SO Max : 80 mV	Accumulation :	1

5.4.1 Spectral range

1 Set wavelength range

The spectral range option allows the user to define the range of wavelength or photon energy over which he wants to have measurement points.

The choice of optimal spectral range depends directly on the **material** under study and of course on the information that is needed (thickness and/or optical constant in a certain spectral region).

5.4.1.1 Optimal spectral: thickness and/or optical constants

If the user is attempting to determine the thickness(es) of layer(s) in the sample, few well chosen wavelengths might be enough to obtain the correct information. Of course, the acquisition over multiple wavelengths allows the overdetermination of the thickness and thus lead to a higher accuracy.

If the user is attempting to determine the optical constant at certain wavelengths, data must be acquired in the corresponding spectral range. Assuming DeltaPsi2 linearly interpolates optical constant between two points of measurement, one has to see how sharp absorption peaks are. The wavelength or energy increments have to be set so that these peaks can be sufficiently resolved. Typically, the increment in energy for data acquisition ranges between 0.1 eV to 0.905 eV.

It is always preferable to acquire data in the transparent spectral range of the material, as this allows to reduce the correlation between the different fitting parameters and to eliminate the determination of material absorption from the analysis. It is important to repeat that ellipsometric data must be acquired in the transparent spectral range of the material to determine the thickness of the film. Therefore the appropriate spectral range for various types of materials is discussed in the following section.

5.4.1.2 Optimal spectral range for various material types

Dielectric materials

Dielectric materials are, along with semiconductors, the most commonly encountered class of materials. Examples for dielectric materials are SiO_2 , $SrTiO_3$, SiC, Si_3N_4 , HfO_2 , LaO_3 , Ta_2O_5 , TiO_2 , MgO or Al_2O_3 These materials are transparent over a wide spectral range from IR to deep UV. Ellipsometric measurements from IR to UV The measurement of these materials can be done over a very broad part of the spectral range because they are transparent from IR to UV and even deep UV. Ellipsometric measurements from IR to UV provide a wealth of information. Al_2O_3 for example is transparent for IR to deep UV due to its wide energy band-gap. Silicon dioxyde instead is transparent for wavelengths > 250nm.

The transparent region for low-k materials is in general smaller than that for high-k materials ranging from the near IR to the visible/near UV, thus for energies < 3 - 4eV. Transparent organic materials have an absorption tail in the UV region which can be described by dispersion functions.

Semiconductors

Semiconductor materials exhibit a band gap. In general we can basically decompose the spectra of semiconductors into two regions:

- energy > band gap: the material begins to absorb and the absorption increases smoothly
- energy < band gap: the absorption coefficient of the material approaches zero (material becomes transparent) and the index of refraction decreases monotonically with decreasing photon energies.

The absorption processes for semiconductors are mainly due to interband transitions above the energy band gap.

Semiconducting material can have crystalline or amorphous structure. For **crystalline semiconductors** the transition from the transparent region to the absorbing region at the band gap is very **sharp**. **Amorphous semiconductors instead have broad absorption bands**. In fact, amorphous solids exhibit absorption tails below the band gap (Urbach tails), which are far more extensive than the absorption tails of crystalline solids.

Metals

The classical model used to describe metals is the Drude model assuming a gas of free carriers. The absorption of energy of an incident beam by the free carriers can be explained by the fact that the electric field of the light beam induces a conducting current whose dissipation leads to the absorption of incident energy. When resolving the Maxwell equation, we obtain the following dependency:

$$\varepsilon = 1 - \frac{\omega_p^2}{\omega^2 + i\gamma\omega}$$

where $\omega = 2\pi \frac{E}{h}$ and ω_p as the plasma frequency:

$$\omega_p^2 = \frac{Ne^2}{m\epsilon_0}$$

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with:

- N the density per volume unit of free carriers
- ε_0 the dielectric permittivity of the void
- m the mass of the electrons
- $e = 1.6 \ 10^{-19} C$

This frequency is the resonance frequency of the free electron gas around the equilibrium state. For common metals, it is located in deep UV and is the frontier between two regions:

- $\omega < \omega_p$ (IR-visible area): the metal has a complex index and is highly absorbing and reflecting.
- $\omega > \omega_p$ (UV region): the index of the metal is almost real, k ≈ 0 : the metal is transparent.

In general, most of the **metals** are highly **absorbing in the IR region** and absorbs lesser and lesser as the photon energy approaches the UV region. However, even if most of the metals become transparent in the UV, the analysis and the modeling operations remain quite difficult.

In general, ellipsometers enable to measure film thickness of metal ranging from 0 to 500Å, depending on the absorption of the metal understudy.

Glasses and plastics

Glasses could be equally regarded as dielectric materials. Most of glasses are transparent over a large part of the spectral range but some of them have an absorption edge in the UV when wavelength gets below 300nm or so. Colored glasses have absorptions in the visible area caused by dopants in the glass matrix. Float glass can be absorbing even in the whole spectral range.

For measurements of thin layers on glass it is always advantageous to determine the dispersion of the pure glass substrate in advance. The dispersion of the substrate must be known very precisely because usually the "optical contrast" (difference in dispersions) between deposited layer and glass substrate is very low.

Plastics are similar to glasses in terms of transparent regions, however, their characterization is much more complicated because they can be highly anisotropic.

Transparent conducting oxides

Transparent conducting oxides like ITO (indium tin oxides) show **absorption in the UV** due to interband optical transitions and also **in the IR** due to free carrier absorption. The modelling of the dispersion is rather complicated near these absorption bands. However, there is a transparent region in the visible part of the spectrum (400nm-760nm) where modelling is relatively straight forward.

Carbon-based materials

Carbon can bond in both, planar (sp²) and tetrahedral (sp³) configuration leading to graphite or diamond form. Diamond-like-carbon (DLC) is mainly used as protective coating due to its hardness and low-friction. **Diamond itself has a high energy band gap of ca. 7.1eV whereas the energy band gap of graphite is much lower.** Annealing in DLC deposition processes causes a decrease of the band gap so that it approaches more and more the one of graphic-like-carbon (GLC). For both, DLC and GLC, a strong absorption occurs at the energy bang gap and transparency is found only well below the band gap energy.

Summary:

MATERIALS	TYPES	SPECTRAL RANGE
Dielectrics	High-k	IR to deep UV
Diciccuits	Low-k	IR to near UV
	Crystalline	Below band gap
Semiconductors		Low absorption
Semiconductors	Amorphous	Below band gap
	Amorphous	broad band of absorption
Metals	ω<ω _p	Highly absorbing in IR and VIS
	ω>ω _p	Less absorptive in UV
Glasses	Normal	IR to visible
	Float glasses	Absorbing
Plastics	PET, PEN, PP, PC	Transparent VIS
Transparent conducting oxides		Transparent in Vis
	Diamond	IR to UV
Carbon	DLC	Transparent Below band gap: IR to UV
	GLC	Transparent Below band gap: IR to VIS

5.4.2 Choice of optimal step size

2 Set step size

After the setting of the spectral range, the user must adjust the step size. The step size must be low enough to resolve spectral features like sharp absorption peaks or interference patterns in thick layers and large enough to make the acquisition as fast as possible.

Step size commonly ranges between 0.1 eV to 0.005 eV depending on the measured sample.

In chapter 1 it was shown that the separation of maxima and minima of the interference pattern decreases with increasing film thickness. This means that with increasing thickness the step size has to be decreased to resolve properly the interference pattern and thus determine unambiguously the film thickness.

The chart below shows typical step size values for different thickness ranges:

Film thickness	Step size (eV)	Step size (nm)
0nm to 200nm	0.1eV	20nm
200nm to 500nm	0.05eV	10nm
500nm to 800nm	0.025eV	5nm, well resolved for wavelengths > 300nm



800nm to 1.2µm	0.02eV	5nm, well resolved for wavelengths > 800nm
1.2μm to 2μm	0.01eV	2nm, well resolved for wavelengths > 400nm
2µm to 30µm	0.005eV	2nm, well resolved for wavelengths > 500nm

5.4.3 Choice of optimal range units

We gave step size values in nanometers for the wavelength scale and in electron-Volts as an equivalent for the photon energy scale. However, keep in mind that acquisition of data with a spectral range in nanometers and acquisition of data with a spectral range in eV do not provide the same spectrum shape and the same distribution of points along the spectrum.

The conversion from wavelength to energy is expressed by:

$$\mathrm{E}(\mathrm{eV}) = \frac{1240}{\lambda(\mathrm{nm})}$$

Note that this conversion formula can only be applied to the endpoints of the spectral range but not to the step size! The following example shows that this would not make sense:

From: 1eV	From:248nm
To: 5eV	To: 1240nm
Every:0.025eV	 Every:49600nm !!!

The software applies the conversion formula automatically only to the endpoints of the spectral range and not to the increments.

To pass from a spectrum in eV to a spectrum in nanometers, or the contrary, the software converts the range bounds and calculates the increments so that the number of points is kept constant, for example:

From: 1eV	From:248nm
To: 5eV	To: 1240nm
Every:0.025eV	 Every:6.2nm

This gives 161 data points in both cases.

Note that for thick layers the interference pattern cannot be resolved for short wavelengths when data is collected in nanometers but they can be resolved for corresponding high energies when collected in photon energies because there the minima and maxima are separated nearly equally over the whole spectral range.

As an example of a 450nm thick sample of SiO_2/Si is shown in the energy range 1eV to 5eV corresponding to a wavelength range of 248nm to 1240nm (data collected in photon energies with increments of 0.05eV. This gives 81 points):



Then converted into wavelength scale, the points still remain equally spread over the interference structure:



Data collected in wavelengths with increments of 12.4nm. This gives again 81 points:



It is obvious that data collected in nanometers does not resolve features in the short wavelength range.

Then converted into a photon energy scale, the points are not equally spread over the interference structure:



Wavelengths or photon energies?

In the following chart, we suggest which scale to use depending on the type of sample:

Sample	Features	Optimal units choice
Thick, transparent films	Rapidly changing features in Ψ and Δ at short wavelengths	Best to measure in terms of eV
Thick, UV absorbing films	Data features flat at short wavelengths.	Best to measure in terms of nm
Very thick and transparent films	Difficult to resolve short wavelength features	Measure only long wavelengths in terms of nm

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5.4.4 Choice of optimal angles of incidence

The goal is to work as close as possible to the Brewster angle (BA) of the substrate because there the highest sensitivity can be achieved.

The BA can be determined by calculation $\tan(BA) = n_{substrate}$ or by monitoring Ψ as a function of the AOI because $\Psi = 0$ at the BA!

Note that you do not need to be exactly at the Brewster angle to perform the acquisition, being near the Brewster angle is sufficient to enhance sensitivity.

In the following chart, we provide reasonable angle of incidence combinations for different types of samples:

Samples	Combinations
Thin films on Si	65°, 70° , 75°
Thick films on Si	60°, 70° , 75° or 55°, 65°, 75°
Other films on glass	55° , 65°, 75°
Films on metals	70°, 80°
Anisotropic and graded films	55°, 65° , 75° or 45°, 60° , 75°

Variable angles of incidence

The variable angles coupled with spectroscopic ellipsometry are used to collect a great quantity of data. However the user must take these informations with a certain care, especially in the case of samples with an overlayer or a roughness layer, because acquisition with variable angles does not always produce completely independent information.

The advantages of this mode are:

- for a given wavelength, the acquisitions at different angles of incidence can give independent information and allow to reduce the measurement noise (improve of S/N).
- additional data statistically improves the determination of confidence limits
- the variation of the angle allows the determination of the angle for which the values of Ψ and Δ are the most sensitive.

5.4.5 Configurations (Azimuth setting of modulator and analyzer)

The UVISEL can acquire data for various azimuth settings of the modulator M and the analyzer A. The two most important configurations are described in the following table:



3 Set angle of incidence

highest sensitivity around the Brewster angle

Configurations	Modulator M	Analyzer A
Π	0° or 90°	±45°
III	±45°	±45°

- configuration II: allows accurate determination of Δ in the whole range but cannot be determined precisely at ca. 45°.
- configuration III: allows accurate determination of Ψ in the whole range but cannot determine Δ precisely at around 90° and 270°.

5.4.6 Configuration modes

In general we recommend to measure only in configuration II and use (Is,Ic) as fitting parameters.

The DP2 software allows automatic measurements in one or more configurations to get the optimal set of data.

Config	jurati	on choice	1
No m	ierge	·	
II: M	= 0°	A=+45° ▼	Í

No merge

This is the standard mode of acquisition. The user can select configuration II or III. As a result one file is obtained either from configuration II or from configuration III. Note that the fit for an anisotropic material can only be done in configuration II (with A=45° and $M=0^{\circ}$).

The merge function

There are three types of merge: Standard merge, High accuracy merge and Smart merge which provide an acquisition in two scans. The principle of these merges is to measure in configuration II & III and calculate the best Ψ and Δ values using both acquisitions.

Standard merge

The standard merge function automatically generates three result files: xxx-CII.spe, xxx-CII.spe and xxx-STM.spe.

For the example, a measurement of thermal oxide on a Si substrate gives the following and for configuration II (blue) and III (red):







There are obviously mirror effects for around 45° and for around 90° and 270°. The cause of these effects occurs in the last step of acquisition, the recreation of Ψ and Δ from Is and Ic:

- in configuration II:

 $Is = sin 2\Psi sin \Delta$ $Ic = sin 2\Psi cos \Delta$

- in configuration III:

$$Is = \sin 2\Psi \sin \Delta$$
 $Ic = \cos 2\Psi$

The two couples (Is,Ic) are shown below; notice Is does not depend on the chosen configuration.





- in configuration II only $\sin(2\Psi)$ is known. It only defines Ψ in the interval [0,45], the values above are flipped down around the line Ψ =45°. On the other hand, in configuration III, the term $\cos(2\Psi)$ allows determination of Ψ over the complete domain.
- in configuration III only sin Δ is known. It defines Δ on the right half of the trigonometric circle, the left-side values are flipped right around the vertical axis. On the other hand, in configuration II, the terms sin Δ and cos Δ allow determination of over the complete domain.

That is why the resulting standard merge file xxx-STM.spe contains Δ from configuration II file and Ψ from configuration III file.



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High accuracy merge

Like the standard merge, this one includes two scans. The final spectrum is composed of:

- Δ taken from configuration II
- Ψ is calculated with Is and Ic from configuration III and Δ from configuration II

with the following formula: $\Psi = \frac{1}{2} \arctan\left(\frac{Is^{III}}{Ic^{III}sin\Delta^{II}}\right)$

Smart merge

Again the first scan is in configuration II and the second in configuration III. The final spectrum is then composed of:

- Δ taken from configuration II
- Ψ is composed of several segments: the limits of the segments are determined by specified Ψ threshold values. The segments with Ψ less than this threshold give Ψ from configuration II and the segments with Ψ greater than the threshold give Ψ from configuration III.

Remarks

A fit can give a bad result after a merge acquisition when:

- configuration II gives wrong Ψ values: it is not possible then to fit on Ψ and Δ
- configuration III gives wrong Δ values: it is not possible then to fit on Ψ and Δ
- configuration III gives wrong Ic values: it is not possible then to fit on Is and Ic

In general we recommend to measure only in configuration II and use (Is,Ic) as fitting parameters.

The only interest of the merge method is to extract exact values for Ψ and Δ but does not improve the fitting itself.

5.4.7 Detector settings

5.4.7.1 Integration time

The integration time is the amount of time allowed for the signal to be acquired for each measurement point. In general select the shortest integration time that will give sufficient S/N in the interesting spectral range. Greater integration time provides better S/N but longer acquisition time.

For standard acquisition we use an integration time of 200ms, for measurements of samples with low reflectivity we recommend a value of 2000ms.

5.4.7.2 Background

Background off: does not take care of the background signal.

Background on: two acquisitions are carried out for each data point. The system closes the shutter located on the light source path and performs an acquisition. A signal plus dark acquisition are performed with the shutter open. The dark acquisition is then subtracted from the signal plus the dark acquisition; this technique minimizes the effects of any drift that may occur.

Automatic background: if the offsets calibration was performed, a results file (background.ini) was saved on the hard disc. If an acquisition is launched using this mode, the $[S0_{background noise}$ (HV)] will be then subtracted from the acquisition data.

5.4.7.3 High voltage

The high voltage box contains the value the user has to set to optimize the gain; this is the voltage applied to the photomultiplier tube (PMT). If you do not know what value to put notice that before launching an acquisition, the user has to acquire the signal in real time through the **VIEWS** window and make some settings described further so as to be in optimal conditions of acquisition. Then by clicking on **S0 adjustment** button, the high voltage value will be **automatically** adjusted.

5.4.7.4 Measurement spot size

The Micro-spot feature allows the user to set the size of the incident beam diameter. The choices are 1200, 250, 120 and $60\mu m$. By reducing the spot size, the signal will decrease, though small spot size may be needed for small or structured samples.

5.5 Spectroscopic MWL acquisition mode

Acquisition Routine - Noname1.acq	
Spectroscopic MWL acquisition setup HV values	Graph setup Graph
Acquisition simulation	Spectral range User unit : eV Start : 1.50 eV
	End: 6.00 eV
II M=0° A=+45° ▼	Use only one modulation
	Number of wavelength :
	Number of mirror shifts :
	Default values
Calibration index: No data	
Micro-Spot :	Mode: Standard
Background : Background on 💌	
Protection UV : Off	
High Voltage : Automatic ▼ SO Min : 60.00 mV SO Max : 80.00 mV	Incidence angle : 70.00 * Integration time : 1 × 200 ms

The following figure shows the DP2 page for setting the acquisition parameters:

Most of the parameters are identical to the Spectroscopic Mono mode. However some additional parameters have to be set:

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Number of wavelengths:

A limited number of wavelength points can be selected.

Number of mirror shifts:

This is the minimum number of shifts necessary to cover the selected wavelength range.

Use only one modulation:

A fixed modulation amplitude is used: the measurement is faster but the precision decreases in the IR.

5.6 Kinetic Mono acquisition mode

This mode allows the user to follow the growth of a material, in a reactor, in real time and for a given wavelength. This acquisition mode is very appropriate for industrial applications like process control. Usually this mode is used to determine thickness of the layer as a function of time.

🚳 Kinetic Acquisition Routine Mono - Nona	me1.ack	
Kinetic mono acquisition setup Result		
Kinetic parameters	Acquisition co	onditions
Kinetic parameters Sampling Time : 0 s Integration Time : 200 ms <u>Total Duration</u> Points Number : 10 Time : 0 h 0 min 0 s <u>Acquisition result format</u> Binary format Text format Spectral position	Acquisition co Configuration : Incidence ang Micro-spot : Background : Protection UV High voltage :	Il: M=0° A=+45' Ie: 70.00 Background on Off Automatic 250 V
	S0 Min :	60 mV
Position : 1.500 eV	S0 Max :	80 mV
Prepare 🖉 Run 🗉 Stop 🐓 Views	<u>S</u> ave	Save <u>As</u> Save as <u>D</u> efault

Sampling time:

Select here the time interval between measurement points.

Integration time:

Set integration time according to S/N.

Total duration:

Defined either by the number of points or by the total time of the experiment.

Spectral position:

Choose the wavelength (energy) where the data are taken.

5.7 R&T Mono acquisition mode

This acquisition mode measures the intensity of a transmitted or a reflected beam:

- in transmission I_T/I_0
- in reflection I_R/I_0



The transmission mode is very useful to decorrelate the analysis of thin absorbing films on transparent substrates or multilayers on transparent substrates.

The measurement in transmission requires several conditions:

- the sample must be placed perpendicular to the normal incident light beam
- The user must pay attention that the transmission measurement of a thick sample (>2mm) may not provide accurate data.
- in this acquisition mode the user must pay attention that fluctuations in lamp intensity or bad light collection may introduce errors in the measurement.
- for both, the reflectance and transmission mode, the user has to create a reference file for each wavelength.

Nevertheless, this measurement type is very important since it can be applied to glass substrates, which are of course very important in display technology etc.

R&T mono acquisi	tion setup Graph	
Calibration index: Spectrum range User unit : Start : End : Increment : Micro-Spot : Background :	No data	Mode : Transmission Transmission Mode: Unpolarized New DC signal reference Drag-and-drop a reference file here :
High Voltage : SO Min : SO Max :	Fixed 250 V 60 mV 80 mV	Incidence angle : 90.00 • Integration time : 1 × 200 ms Accumulation : 1

Note that the High Voltage must be fixed during the whole experiment because we want to compare absolute intensities.

Mode:

Choose transmission or reflection mode. Both can be performed in polarized or unpolarized mode.

Reference file:

Transmission and reflection experiments need a reference measurement of a known sample. This can be simply air (transmission) or Si (reflection). Choose whether you want to acquire a new DC signal reference or use an already existing measurement (Drag-and-Drop).

5.8 R&T MWL acquisition mode

R&T MWL acquis	ition setup HV values	Graph
Spectral range- User unit : Start : End : Use only one Number of wave Number of mirro Default value:	eV ▼ 1.50 eV 6.00 eV e modulation elength : ▼ s	Graph Mode : Transmission Transmission Mode: Unpolarized New DC signal reference Drag-and-drop a reference file here :
Micro-Spot : Background : Protection UV :	▼ Background on ▼ Off ▼	
High Voltage : SO Min : SO Max :	Fixed 60.00 mV 80.00 mV	Incidence angle : 90.00 • Integration time : 1 × 200 ms Calibration index: No data

Most of the parameters are identical to the R & T Mono mode. However some additional parameters have to be set:

Number of wavelengths:

A limited number of wavelength points can be selected.

Number of mirror shifts:

This is the minimum number of shifts necessary to cover the selected wavelength range.

5.9 Summary: measurement strategies

Sample type	Spectral range/ Step size	AOI	General settings	Additional measurements
Thin transparent layer/Si	UV-VIS/0.1eV	70°	IT = 200ms	
Thick transparent layer/Si	VIS-IR/0.005eV	70°	IT = 200 ms	
Absorbing layer/Si	Transparent region	70°	IT = 200ms	

The following table shows typical settings for important applications:



Multilayer/Si	UV-VIS/0.05eV	70°	IT = 200 ms	VASE: 65°, 75° R
Organic layer/Si	UV-VIS-IR/0.05eV	70°	IT = 200 ms	VASE: 65°, 75° R
Glass substrate	UV-VIS-IR/0.1eV	ca. 55° (BA)	IT = 1000ms	
Thin layer/glass	UV-VIS/0.1eV	ca. 55° (BA)	IT = 1000ms	
Multilayer/glass	UV-VIS-IR/0.05eV	ca. 55° (BA)	IT = 1000ms	VASE: 60°, 65° T
Organic layer/glass	UV-VIS-IR/0.05eV	ca. 55° (BA)	IT = 1000ms	VASE: 60°, 65° T
Metal substrate	UV-VIS/0.05eV	70°	IT = 200 ms	VASE: 65°, 75°
Thin Transparent layer/metal	UV-VIS/0.05eV	70°	IT = 200 ms	VASE: 65°, 75°
Thick Transparent layer/metal	VIS-IR/0.005eV	70°	IT = 200ms	
Plastic substrate	UV-VIS/0.05eV	ca. 62° (BA)	IT = 1000ms Config. II	VASE: 60°, 65° T
Polymer/plastic	UV-VIS/0.05eV	ca. 62° (BA)	IT = 1000ms Config. II	VASE: 60°, 65° T



6 QuickStart Modeling

6.1 Introduction

The collected data of an ellipsometry measurement are the intensities of the harmonics of the reflected or transmitted signal in a predefined spectral range. These harmonics are mathematically treated to extract values of Is and Ic for each selected wavelength thus providing spectra Is = $f(\lambda)$ and Ic = $f(\lambda)$. These values are not material parameters or thicknesses thus ellipsometry is an indirect technique of measurement.

Starting from the values Is and Ic, the software can calculate quantities of interest such as Ψ and Δ .

To extract parameters of interest like optical constants and thickness from the measurement a model has to be set up that allows theoretical calculation of Is and Ic (or Ψ and Δ etc.). The parameters of interest are determined by comparison of theoretical and experimental data (fitting: detail in chapter 6), the true optical constant values of the different parts composing the sample structure.

A layer model can includes:

- layer structure: bare substrate, mono- or multilayer, super lattices...
- optical dispersions for each material: literature values, dispersion formulae, alloys, effective media
- thickness of the different layers
- optical anisotropy, index gradients,...

This abundance of modeling parameters is provided in order to match the experimental data as close as possible. Once the unique best fit is obtained the given model is considered the most sensible representation of reality. Thus the values for the modeling parameters (thickness, optical constants,...) obtained by the fitting operation will be considered as being the real ones.

Obviously, ellipsometry does not provide 100% exact values but which technique does? However, despite of errors due to modeling or measurement, ellipsometry is still an incredibly accurate, reproducible and precise optical technique of measurement. Furthermore, the present thin film calculation software has been designed to fulfil the analysis requirements for a great variety of users: the DP2 software can be used just as well by thin films research workers and by operators in semiconductor fabs.

Never forget that you have in hand a tool that is sample sensitive down to the atomic scale, just by putting your sample under light, in air environment without destroying.



6.2 Pre-requirements for modeling

Before starting the modeling step, there are some questions the user should answer to correctly proceed through this fundamental step of ellipsometric data analysis. Indeed, to build a model the user has to know the maximum of information about:

- the substrate, its composition, its historic (cleaning, annealing,...), its surface state (roughness, native oxide,...)
- the coating material(s) composing the sample, the number of films, deposition method, the type(s) of interface(s), post-treatment(s) like annealing
- homogeneity of the sample, isotropy or anisotropy (birefringent material, preferential polarization,...), index gradient
- presence of overlayer(s) like thin oxide film or roughness layer

The user has to keep in mind that all ellipsometric analyses are model dependent because it is not possible to measure physical and optical quantities directly. The desired parameters such as dielectric functions, refractive indices, material compositions, and film thicknesses are determined by reference to a model that is an idealized mathematical representation of the sample. So, the model must be physical and note that any inaccuracies in the fixed parameters will lead to inaccuracies in the final results.

Ellipsometric modeling provides a real challenge and is the part of job that is the most stimulating to the user of ellipsometry. The technique has a reputation of time sharing 10% for measurement and 90% for analysis. This is partly true but the pleasure or benefit consecutively obtained afterward just gets consecutively higher.

6.3 DeltaPsi2 software environment – Modeling Interface

In this section we review a number of important software concepts or menus which are used during data analysis with DP2. You should be familiar with all of these environments as they are used throughout the analysis examples of this document. Discover each element of modeling software environment before proceeding onward.

6.3.1 Configurate and discover the working environment: User Profile

DP2 is a multi-user working platform. Thanks to the *user's profile* feature, all files to work with remain in a local environment. Consequently you can chose to create a profile for each user, for a group of users, for a specific application...To create a profile click on *software options* button and *create a new profile*...



There are no limits to the number of defined profiles. Once a new profile is created or an existing one is selected, the central node for handling analysis is the TreeView. Its general overview is detailed in the *DP2 reference manual* and its functionalities are similar to those of windows explorer.



6.3.2 TreeView folders



There are 4 main folders in this tree. They are visible directly when you launch the software. Two of these are libraries and two others are for data:

- User library
- Results
- Reports
- Application library

The secondary folders are visible or hidden according to a selection made in "software option/file explorer".

Application library and User library have the same sub-structure. The first one is the pre-set HORIBA Jobin Yvon library containing model examples, reference files, material dispersion files delivered with the installation disk. The latter should be the result of your work. It starts containing empty folders every time you create a new profile.

We suggest your work to be saved in the User library. Copy from the application library files of interest in your user library.

- Results folder is the place where raw data are saved. These are either measured data or modeling result data.
- Reports folder is the place where formatted data are saved. Modeling results or recipe results are saved here following a template format for printing or displaying HTML or microsoft word DOC documents.

6.3.3 Creation of a new model

Create a new model by clicking on the Mod. tab.:



Make sure «User Library» is selected and then click on the «Spectroscopic Model» button.



6.3.3.1 Built your model using layers for each material

Drag acquisition data, reference, dispersion, etc. files from the tree view selection menu and drop it into the sample structure



Drag the experimental measurement file from the directory *results – acquisition data* to the grey bar "Exp file".

Materials	Z Model - Polymer_on_SL_dessical.mdl	40 -
anatrial anatrial and Ag and Ag an		
	Photomerant (mean and before the for Char graph before insider (* Sheer al Hing dags	
Moosi Ni NPD PEDOT PET PILPGA Dulumer	The set of	2

A default angle of incidence of 70° is given. You can change it by clicking on " 70° ". Generally do not fit the angle of incidence.

From "Modeling description" menu, click "Edit...". From the "Modeling conditions" tab, the "Experimental file spectral range" fits your model to the spectral range of the used experimental file. However, if you want to use a smaller range, you can use a "User defined spectral range". Click "Ok".

6.3.3.2 Set the unknown properties of the sample as model fit parameters

Check the boxes in the sample structure, and an "F" will appear

¹ 🖪 2800.00	Epolymer_classical.dsp	×	×
S	C-si_isa.ref	×	

6.3.3.3 Launch the fitting process

Now the fitting process can be launched. The software will automatically adjust the selected fit parameters to improve the agreement between the measured and the model-generated data.

Click on the "Fit" tab

Epolymer_classical.dsp			x ×
C-si_isa.ref			×
Pitting choice Fitting procedure Modeling description Output description Ortfold v Single Step v Default v	Simulation	Save	
Add 🔮 Edt Add 😤 Edt Add 😤 Edt	Fit	Save As	

6.3.3.4 The results are displayed graphically and in numbers.

The fit quality can be checked with the value of χ^2 (a smaller value indicates a better fit to the experimental data) and visually compare the measured experimental data and the final best-fit calculated data.



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6.4 Materials files

In both the application and the user library there is a folder called *materials*. The folder contains files that represent the optical properties materials. These optical properties can be either expressed as complex refractive indices N = n + ik or as dielectric functions $\varepsilon = \varepsilon_1 + i\varepsilon_2$. The two expressions are related according to $\varepsilon = N^2$.

Note that optical properties for a material are physically meaningful if $k \ge 0$ and $\varepsilon_2 \ge 0$ and that these optical properties are given only for a limited spectral range.

DP2 offers different "file" possibilities to handle optical properties. Here is a short overview.

6.4.1 Material file types

6.4.1.1 Reference (*.ref)

A reference file contains tabulated values of optical constants (n,k) as a function of wavelengths taken from literature.

These files have text format in a structure similar to all deltapsi2 experimental data files. They contain a table with 3 columns, wavelength or energy, real part then imaginary part of optical properties. Most of Application library files are from this type and contain energy and dielectric function. We will often mention these files as reference files. One can already keep in mind that **optical properties in such files are fixed and consequently cannot be adjusted**. A complete list of application library available reference files is given in the appendix.

Example:



Access:

In the left-hand selection menu, expand «Application Library» «Materials»

Modelling Strategy:

Using reference files means that the optical constants are fixed values. The advantage is that only the thickness will vary during the fitting process. The disadvantage is that it may not exactly represent the real optical constants of your material.

When to use it ?

Often used materials like c-Si, SiO₂, Al, Al₂O₃, etc. are present in the application library. Use them as a starting step for modelling. A low value of the χ^2 implies a good representation of the optical constants and thickness of your sample under analysis. In the contrary case modify your model or your fit parameters until the best description is found.

Examples:



Transparent Dielectric

Absorbing Dielectric



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Semiconductor







6.4.1.2 Dispersion Formula (*.dsp)

Optical dispersion means dependence of optical properties on wavelength or photon energy. Sometimes this dependence can follow an explicit analytical formula, called dispersion formula. These dispersion formulae mostly are physically related to optical transitions in materials, however, they also can be purely empirical without any physical meaning.

DP2 offers pre-defined dispersion models which can be used straight forward. A legitimate question is "which one should I use?". Chapter 6 will explain in detail the most important dispersions. Appendix A give the theoretical background of all dispersions.

Note that a dispersion file contains the following information:

- the type of dispersion equation
- the dispersion parameters, real numbers
- the dispersion parameters boundaries: an optional constraint when fitting these parameters. When min = max = 0, the fitting routine considers no boundaries.

Example:

🚳 SiNx.dsp

Access:

Click on the "Mat" tab to access the dispersion files.

Examples for often used dispersions:

Main dispersion formula	Materials family	Materials examples	
Lorentz oscillator	Transparent and weakly absorbing dielectrics	Oxides (SiO ₂ , SiON, Al ₂ O ₃ , HfO ₂ , MgO, ZrO ₂), polymers, organics	
Drude oscillator	Metals	Al, Ag, Cr, Cu, Ti	
New amorphous, Tauc Lorentz	Semi-absorbing dielectrics and semiconductors	SiN, a-C, oxides (TiO ₂ , Ta ₂ O ₅ , ZnO), AlN, photoresist	
Kato-Adachi, Adachi-New Forouhi	Crystalline semiconductors	AsGa, InP, InAs, HgCdTe, InGaP, ZnTe, CdTe	

A dispersion file is used when a reference file for the material is not available or if the reference file does not lead to good fitting results (high χ^2 value).

Comment:

Weigh the benefit gained (goodness of fit) when using a dispersion based model instead of a reference file against the added complication (more regression parameters). As a rule of thumb the χ^2 should be improved by an order of magnitude when using a dispersion file.

6.4.1.3 *.ALY files

In Deltapsi2, the optical properties of four families of semiconductor materials can be described by an alloy modeling. These families are:

- Al_xGa_{1-x}As
- Hg_{1-x}Cd_xTe
- Si_{1-x}Ge_x
- In_{1-x}Ga_xAs_yP_{1-y}

The three first are ternary alloys, the last one is a quaternary alloy. For the three first, the composition is defined by the stoechiometry x, a value that can vary between 0 and 1. The fourth family of alloy is the quaternary material $In_{1-x}Ga_xAs_yP_{1-y}$. Its optical properties are given for the case of x = 0.47 y. This stoechiometry relation restricts this family to the alloys that are lattice matched to InP. For this case, composition is defined by stoechiometry y, also between 0 and 1. For example:

y = 0 corresponds to the material InP

y = 1 corresponds to the material In_{0.53}Ga_{0.47}As.

For a given family of alloys, the modeling is based on existing reference files of known composition between 0 and 1. The software applies an appropriate interpolation procedure to generate optical properties for intermediate compositions within the interval [0, 1].

Note that an alloy file contains the following information:

- the type of alloy
- its composition

Using this type of material file illustrates how ellipsometry can give access to the composition of materials.

*.PPC files

PPC means **point-by-point calculation**. This kind of material file is devoted to cases where fitting extracts the optical properties directly from experimental data at each wavelength. Assumptions are made regarding all other parameters that are kept constant.

Modeling with PPC file types will be illustrated in detail later in this manual.

*.SPE or *.CLC files

SPE files are directly generated by **acquisition routines**. **CLC** files are **calculated files** that can be generated in the simulation environment. These two types of files are text files with standard DP2 data format. We consider in the following explanation ellipsometric data (remember DP2 also handles Reflectance or Transmittance data).

It is known from ellipsometry basic theory that angles Ψ and Δ for a sample are related to the "pseudo-index" of a sample according to the following equation:

$$\rho = \tan \psi e^{i\Delta}$$
$$\varepsilon = n^2 = \sin^2 \theta_0 \left[1 + \tan^2 \theta_0 \frac{(1-\rho)^2}{(1+\rho)^2} \right]$$

with θ_0 as the angle of incidence.

When the sample is a bulk material, its pseudo-index is simply the complex index of refraction of the material.

An ellipsometric data file can be seen as a material file, both, SPE and CLC files, can be used in the materials folder. and consequently in the model.

6.4.2 Creation of a dispersion file in the DeltaPsi2 software

For the creation of a dispersion Click on *Mat* in the top bar:



6.4.2.1 Creation of a dispersion file with a pre-defined dispersion formula

Select *Dispersion* in the following window if you want to use a given dispersion formula to describe your material dispersion:

Create new material				
User Library C Application Library				
aly	Alloy			
dsp	<u>D</u> ispersion			
Ūdf	<u>U</u> ser Defined Formula			
্বর্ণ	Point-by-Point Calculated			
	X Cancel			

A dispersion window is displayed with a rolling menu to select a dispersion among the 32 existing, a list of adjustable fitting parameters and a graph that plots the dielectric function or other optical parameters (n, k,...) as a function of photon energy or wavelength.

Adjust the fitting parameters to your convenience and save as your dispersion.

The *.dsp file will be placed in the *materials* folder in the *User Library* of the TreeView. You can then drag and drop it in the layer model.

Each dispersion formula is an adjustable function. The parameters of the function are fitted when the corresponding check box in the model window is activated as shown below:



A fitting window appears by clicking twice on the check box allowing you to select the parameter(s) to fit. An example is shown below for the dispersion formula "Classical" that takes into account only the single lorentz (transparent) oscillator set fot the SiO₂ material.



Dispersion	Fit Parame	eters					
Dispersion file name : SiO2.dsp Dispersion formula name : Cauchy Absorbent							
Paramet	ters						
		Nominal	Check :	Absolute	C Rela	ntive	
	Fit	Value	Min	Мах	Internal	Output	Fitting mode for A :
A	_	1.0000000	0.0000000	0.0000000	Г		Nominal value
В	v	0.0000000	0.0000000	0.0000000			O Multiguess absolute :
С	v	0.0000000	0.0000000	0.0000000		-	O Multiguess relative :
D	<u> </u>	0.0000000	0.0000000	0.0000000			O Multistart absolute :
E	_ _	0.0000000	0.0000000	0.0000000			Multistart relative :
F		0.0000000	0.0000000	0.0000000	Γ		
1							J
Close							

6.4.2.2 Creation of a dispersion file with an alloy function

The dispersion for alloyed materials is created by an interpolation technique. The interpolation is based on real measurements that were performed for a series of compositions for the four alloy types of the library: $Al_xGa_{1-x}As$, $Hg_{1-x}Cd_xTe$, $Si_{1-x}Ge_x$ and $In_{1-x}Ga_xAs_yP_{1-y}$.

To create an alloy material with the Alloy feature of DP2, perform the following steps: In the model window, go to *mat* and select *Alloy*

Create new material				
User Library O Application Library				
💩 Alloy 🔫				
dsp Dispersion				
User Defined Formula				
Point-by-Point Calculated				
X Cancel				

In the Alloy window, select the alloy type. According to the x value the user set, the software will generate a spectrum of the given alloy by interpolation between the existing spectra of this alloy measured for well defined compositions.

Alloy type :	
Al(x)Ga(1-x)As	•
Default X :	0.730

Save the material in the *material* folder of the User Library with the extension *.aly.



You can drag and drop it in your model, like the other material files in DP2

6.4.2.3 Creation of a dispersion file with the PPC feature

The Point-by-Point Calculated feature (also called NK fitting routine) fits n and k on a point-by-point basis. NK fitting does not use dispersion equations for the optical constants but it requires the exact knowledge of the film thickness.

It was shown that spectroscopic ellipsometry measures two parameters, and as a function of wavelength. The optical properties of a sample (n, k) vary with wavelength, but the thickness (d) is of course independent of wavelength. If the thickness is known then a point-by-point calculation can invert the ellipsometric equations and give n and k for the sample at each wavelength.

The DeltaPsi software achieves this by performing the following steps:

From the main menu, select Mat. choose Point-by-Point Calculated.

Create new material 🔀				
User Library C Application Library				
من <u>Alloy</u>				
(dsp) Dispersion				
udf) User Defined Formula				
Point-by-Point Calculated 🔫				
× Cancel				

Then the following options appear:

Default n&k gu	iess value sou	rce :			
Fixed inde	ex guess value	: ` n : [1.000		
O Previous	guess value :	∫ k: [0.000		
O Material 1	:				
Drag-and	l-drop a materi	al file here	壷		
🔿 Material 2	:				
Drag-and	l-drop a materi	al file here	壷		
O Material 3	:				
Drag-and	l-drop a materi	al file here	壷		
C Table values : Clear table					
Energy	n	k			

The *fixed index guess value* feature uses fixed optical constants, n and k, as the guess for each point.

The previous guess value feature uses the previous NK fit as the new guess for each point.

The *material 1, 2 & 3* feature uses one or several materials created by dispersion as starting point for the guess at each point. To perform this, drag and drop the material(s) to the appropriate box and check the box that contains the material you want to fit with.

When you save your PPC material you create a *.ppc file in the *material* folder of the *User Library*.
If you want n and k over all the spectral range or over one specified region, you will surely notice that, for most materials there are regions of the spectral range where the fit is not perfect. That is why we recommend to decompose the spectrum:

- for each region of interest, create a different *.ppc file
- for region(s) of the spectral range where the fit is good, use your current dispersion as starting point for the guess and be sure to perform the fit only over the specified region. To do this go to *Modeling description > edit > Modeling conditions* then check the box *User defined spectral range*, here you can select the interval and the increment; we recommend to set the same increment you used during acquisition.
- for region(s) of the spectral range where the fit is not satisfying try another dispersion which provides a better fit over the specified region(s), then use it as starting point for the guess and fit over the specified region(s).
- in the fitting step, fit only on the PPC material; don't fit the thickness (it is supposed to be constant)
- fit only using the Marquardt algorithm: go to *Modeling description > edit* >*fitting params*, choose *Marquardt* in *Minimization algorithm*.
- after each fit, click on *Update* above the graphic displayed in the model window: it updates your *.ppc file, so when you open your PPC material, you get access to the optical constants for each photon energy or wavelength. If you created a *.ppc file for each region you can have n and k over the complete range of acquisition.
- to see if you performed correctly the entire process, you should obtain a χ^2 equal to zero.

6.4.2.4 Creation of a dispersion file from analytical expressions

Select User Defined Formula in the following window if you want to compose your dispersion formula from analytical expressions to describe your material dispersion:

Create new mat	Create new material	
User Library Application Library		
aly	Alloy	
dsp	<u>D</u> ispersion	
Udf	<u>U</u> ser Defined Formula	
99	Point-by-Point Calculated	
	X Cancel	

A window is displayed that allows to compose the dispersion formula from pre-defined blocks:

This function is dedicated to advanced users



🚇 User defined formula - None	ame1.udf
Spectral range and comments	Formula A 🖏 🔍 🤍 1:1 🖨 😫 🕻
Build standard formula Add new fo	mula block 🔗 Left : 💌 X :
Build from dispersion file	
Fixed parameters : Value Minir	num Maximum
Function blocks :	
Δ Select the formula block	to add
Select the formula block :	Formula block detail :
Epsilon	
Oscillator Stat	$f_i\omega_{0i}^2$
Drude model Oscillator	$\mathcal{E} = \frac{1}{2} \frac{1}{$
Amorphous	$\omega^2 - \omega^2 + i \chi \omega$
Sellmeier Transparent Sellmeier Absorbent	$\omega_{0j} \omega + i\gamma_j \omega$
Kinf	
Cauchy Transparent Cauchy Absorbent	
Conrady	
Fixed Index	
Sqrt(Eps)	
Excition E0D	
Transition E1D	
DH0 E2	
New Amorphous	
Tauc Lorentz Foroubi	
Transition E0	
Transition E1 Excition Wannier	
DHO E2a	
Excition E0 New Ecroubi	
Hartman	
Tanguy Tanguy Eutondod	
Afromovitz	
Lorentz Oscillators - General	
Gauss	
Cody Lorentz	
MicroSiL	
PolySX PolySX	
Fuyar	
1	
	UK Cancel

The example below shows a new dispersion formula composed of two blocks, a Drude and a Tauc Lorentz term to represent micro-crystalline silicon:



6.5 Creation of a Layer Model

Model files are kept in model folder with the extension: *.MDL. In the main window of DP2, click on *Mod* button

券 券 券	⇔	⇔	⇔
Mat. Acq. Mod.	Grd.	Rec.	Tem.

and select Spectroscopic Model:

Create new mode	l	X
User Library	C Application Library	
mdl	Spectroscopic Model	
tem 🕅	Bound Multimodel	
	🗙 Cancel	

The following window is displayed:

DeltaPsi2	Jobin Yvon / Horiba	Print	Help	About
🛅 🚼 😭 📝 🐋 🕅 at. 🕈	aq. 🕺 d. Šrd. Šrd. 👬 👘 💕 🥬 🔟		*/*	0
-	😹 Model - Noname1.mdl			- 8 ×
 User Library Results Application Library 		Bave results file	Update	Show report
	Thickness unit: A P Clear results before fit P Clear graph before simulation Show al A01: 70.000 • Exp. File:	I fitting steps		<u></u>
	S			
	Fitting choice Fitting procedure Modeling description Output description Default	Simulation Fit	Save Save As	
₩ Noname1.mdl				
Current profile : Default				
			10m main	09:57:32
📴 nan	ier 🛛 👔 👘 manual - Microsoft Word 🛛 💆 The model window - Micro 🛛 🗛 Jobin Yvon 7	Horiba	10	209:57

with:

- 1 TreeView
- 2 model part
- **3** result window
- **4** fit/simulation graph window

In the TreeView go to the Results/Acquisition data folder and drag and drop the experimental file in the Exp. File box.

Then go in the TreeView to the materials folder of the User Library (for created materials) or of the Application Library (for reference materials) and select the material(s) for the substrate layer (S).

Drag and drop the material(s) from the TreeView window to the substrate layer

To create layers on top of the substrate layer just drag and drop a material file above the substrate layer.

For example a two-layer model look like this:

2 🗌 1	000.0	B.dsp	×	×
1 🔲 1	000.0	A.dsp	×	×
S		Substrate.spe	×	

For all layers a mixture (effective medium approximation: EMA – details in chapter 5) of maximum three materials per layer can be created. For example a layer consisting of material A and B would look like this:



In the grey fields, you can:

- set thickness and/or composition values. A click on these values allows to open the "Fitting Choice Properties" window where you can set the fitting conditions.
- delete a layer by clicking the cross in the grey part on the right side.
- create/delete a layer repetition by a right click in the grey part
- change the type of layer by a right click in the grey part (gradient layer, anisotropic uniaxial/biaxial layer or gradient anisotropic uniaxial/biaxial layer).

In the coloured fields you can find all that concerns the material(s) and:

- access the material file by a double click on the material name in the layer. This possibility allows the user to see and/or modify the optical properties of the material. The modification of optical properties is not possible with reference files, you can only modify *.dsp, *.aly and *.ppc files. Remember to save your material file after modifications!
- delete a material by clicking on the cross in the colored parts
- set "Fitting parameters" by opening the "Fitting Choice Properties" window by a right click on the colored fields.

The modeling step and the fitting step are connected: some of the actions made in the modeling step are part of the fitting step. The fitting of a model to data is explained in detail in the corresponding chapter.



6.6 Thin Film Layer Structures

6.6.1 Modelling Strategy for Substrate

A film free substrate is characterized by: $0 < \Psi < 45^{\circ}$ and $0 < \Delta < 180^{\circ}$. A measured value of greater than 45° implies that the sample is not a simple substrate, but has at least one film on it.

6.6.1.1 Opaque Substrate

The ellipsometric measurement of an opaque bulk substrate provides directly its optical constants (n,k) (assuming that the substrate is uniform and isotropic). An unknown surface film or roughness will cause the (n,k) values to be slightly different from the true ones.

The ellipsometric measurement of the substrate can be used as a reference file. Just drag and drop the ellipsometric measurement (.spe) inside the substrate.

¹ F 6000.0	F sio2-osabs.dsp	×	×
S	substrat lah 66.spe	×	

6.6.1.2 Transparent Substrate

A transparent substrate exhibits basckside reflections.

Case1: The substrate was measured with its backside reflections.



Use the **«backside reflection**» model. This model allows the optical treatment of the backside reflections collected during the measurement.

It consists in the use of void in the substrate, then add a layer that represents the transparent substrate (do not forget to mention its thickness), then continue your sample structure under analysis. In the example below, thicknesses are in Å.

³ F 28.0 🗆 %	F sinb.dsp	50.00 % × void.dsp	50.00 % × 3
2 F 1782.6	F sinb.dsp		X
1 500000.0	Bk7.ref		X
S	Void.ref		×

Case2: The substrate was measured without backside reflections (the surface at the bottom of the substrate can be roughened: the light is scattered or the backside reflections are spatially separated from the first reflection and masked).







Sample with a mask

Use a simple substrate containing the reference or dispersion of your transparent material substrate.



6.6.2 Modelling Strategy for Single Layer

If the layer consists of a transparent material then we see interference fringes over the whole spectral range:



A Semi-absorbing material would show damping of the oscillations in a part of the spectrum. The following example shows a layer that is transparent from 1.5-3.5 eV and absorbing for energies above 3.5 eV:





In the absorbing region the film behaves like a substrate. Light must penetrate the film to determine its thickness. In the example of the semi-absorbing material, you can determine the thickness in the spectral range 1.5 - 3.5 eV.

The thicker the transparent films is the closer the interference fringes are:











To get a first guess of the thickness, change the thickness of the layer and press the simulation button to see whether the calculated interference period matches the measured one:

1 F 20000.0	Sio2_isa.ref	
S	C-si_isa.ref	
Fitting choice Fitting procedure	Modeling description Output description Simulation	Save
Add 👘 Edit Add 👘 Ed	fit Add 👘 Edit Add 👘 Edit Fit S	ave As

Alternatively multiguess or multistart functions allow to vary automatically the initial fitting values of thickness in a range defined by the user.

Click on the thickness value of the layer. The fitting choice properties screen will appear, then select the multiguess or multistart options available in the thickness tab. The larger the range and the smaller the increment, the longer the model calculation will take.

Fitting Choice Properties	for layer #1	×
Thickness Materials Spec	cial functions L.B.R. Ch	iecks
🗹 Fit		
🔿 Nominal value :	20	000.0 Å
Multiguess absolute :	Start 1	000 0 Å
O Multiguess relative :	End: 2	000.0 Å
Multistart absolute : Multistart relative :	Increment :	100.0 Å
C Correlated to : Layer	number: 1	
Ratio f	actor :	
Fixe	ed :	1.000
🔘 Vari	iable Start :	1.000
	End :	1.000
	Increment :	0.000
0	k Cancel	

6.6.3 Modeling Strategy for a Layer Deposited on both Sides of the Substrate

This case often happens when a dip coating technique is used to deposit the layer.

If the substrate is transparent (like glass in the visible or c-Si in the NIR) and the backside reflections are collected during the measurement (that is there is no mask that hides the backside reflections, or the surface at the bottom of the substrate is not roughened) then the layer model has a structure as shown below:

Activate Multiguess or Multistart functions



⁴ F 314.7	FGST-2_tl.dsp	×	×
³ 1028.0	sio2.dsp	×	×
2 3000000.0	C-si_isa.ref	×	×
1 1028.0	sio2.dsp	×	×
S	void.dsp	×	

The model is based on the backside reflection model, so it implies void as the substrate. In this example, there is on both sides of c-Si a SiO_2 layer.

6.6.4 Modeling Strategy for Multilayers

It is good to know the approximate thicknesses of the individual layers. If this information is not available the multiguess or multistart options can help, but this might be very timeconsuming for more than two layers.

6.6.5 Roughness

Roughness is a small-scale variation in the height of the sample surface. It is often due to the thin film deposition technique or sample preparation.



Surface roughness is described as a discrete thin film at the surface. This layer can be described with the effective medium approximation (EMA) as a mixture of the material of the layer below with void (air) in a 50:50 ratio.

For this purpose add another layer and drag and drop the void reference file and the material of the layer below in this new layer. The default composition for a two-material layer is 50:50.

2	F 50.0	% F SiNx.dsp	50.00 % × Void_asp.ref 50.0	0 % ×	×
1	F 1148.0	F SiNx.dsp		×	×
S		Csi_wor.ref		×	

Modelling Strategy:

For small roughness thickness starting value is ca. 50Å. Generally do not adjust the composition. For large roughness the thickness starting values might be 150 - 500 Å. The EMA that is mainly used is the Bruggeman model (it is set by default and called EMA in DP2).

Thick roughness layers might scatter and depolarize light. Compare the χ^2 achieved with a fixed composition of 50:50 and a fitted composition parameter. Does the result make physically sense (for example 100% void: 0% material does not)?

Always begin with a simple model and increase complexity step by step. Add a rough overlayer on top of the structure and see whether the goodness of fit (χ^2) improves significantly.

Thin film deposition techniques that often produces a rough overlayer are sol-gel, anodizing, evaporation, sputtering, etc. The oxidation (oxide layer < 100Å) of a surface can show the same effect as a roughness layer on top of the sample.

The use of a roughness layer can improve drastically the goodness of fit (χ^2) in a model. Typical examples give a goodness of fit of $1.5 < \chi^2 < 10$ without roughness layer for a film thickness < 5000Å. Adding a roughness layer can lead to $\chi^2 < 1$.

Note that surface roughness extracted by spectroscopic ellipsometry correlates with roughness measured by other techniques, such as AFM.

The example below shows how adding a roughness layer can improve the fit:

• Model without roughness: $\chi^2 = 6.23$





• Model with roughness: $\chi^2 = 0.46$



6.6.6 Interfacial Regions

An interface describes the common boundary of two layers. In a thin film structure a mixture of the material below and above the interface is assumed (using the Effective Medium Approximation EMA theory).

In the example below we want to create an interface layer between layer 3 (P-si-as) and the top layer (sio2_cl):

In the blue area of layer 3 click right and select copy layer to clipboard and then paste layer to clipboard. Another layer will be inserted above the layer 3 that contains the material of this layer. Then drag and drop the material of the layer 5 inside this new layer. A default composition of 50:50 is set:

	1807.0	Sio2_isa.ref		x ×
	4000.0	P-si-as_jel.ref		x x
E	100.0 F %	P-si-as_jel.ref	50.00 % × Fsio2_cl.dsp	50.00 % × ×
• F	2000.0	F sio2_cl.dsp		××

A reasonable starting value for the thickness of the interface layer is: 100Å.

In a first step keep the composition fixed and adjust the thickness of the interface. It is possible to automatically vary the thickness by using the multiguess or multistart option.

In a second step fit on both, the thickness and the composition and see whether there is a significant improvement in the goodness of fit.

An interface layer is usually used when a homogeneous layer model gives a large χ^2 and adding a roughness layer still does not give a satisfactory χ^2 .

Thin film deposition technique that produce inhomogenous coatings and thus interface layers are sol-gel, anodizing, evaporation, sputtering, etc.

The example below shows how an interface layer can improve the fit:

4 F 1904.4	E sio2_cl.dsp	×	: ×
³ F 3591.7	P-si-as_jel.ref	×	×
2 🗌 790.0	sin_new_tt_w8.dsp	×	: ×
1807.0	Sio2_isa.ret	×	X
S:	C-si_isa.ref	×	:

• 4 layer model without interface: $\chi^2 = 24.5$





• 5 layer model with interface: $\chi^2 = 8.7$

⁵ F	1950.7	F sio2_cl.dsp		x ×
1 F	110.1 F %	P-si-as_jel.ref	82.19 % × F sio2_cl.	dsp 17.81 % × ×
³ F	3505.6	P-si-as_jel.ref		x ×
2	790.0	sin_new_tt_w8.dsp		x ×
1	1807.0	Sio2_isa.ref		××
S		C-si_isa.ref		×





7 Advanced Modeling

7.1 The most important dispersion formulae

As already said the real and imaginary parts of the optical properties of solids are wavelength / energy dependent. This dependency can be described by so-called dispersion formulae. These dispersion formulae can be classified in the following groups:

- empirical formulae
- classical oscillator models
- amorphous and multiple oscillator models
- transition based models (excitons,...)

The dependency of the optical properties on wavelength is due to different types of optical absorption processes in the materials. The most important are:

- interband absorption

This type of absorption occurs when a bound electron absorbs a photon from the light beam and is transferred to a higher energy level of the material. This absorption process appears in the range from NIR to UV. This process is typical for nearly all semiconductors and dielectric materials. Metals also show interband absorption, but additionally a strong absorption of photons by free carriers.

- intraband absorption

This is the second most important type of absorption process in which an electron absorbs a photon from the light beam and is transferred to a different energy state within the same band.

- free carrier absorption

Conducting materials such as metals contain free electrons which are not bound to any specific atom of the material. The absorption is usually proportional to the concentration of carriers. Because such materials have no bandgap, as Fermi energy for a metal lies within one of the electron energy band, the free carrier absorption is an intraband absorption phenomenon.

7.1.1 Empirical Formulae

These formulae are mostly based on polynoms. The most important examples are Cauchy, Sellmeier, Conrady and Briot. They can be used in the case that the refractive index changes simply with wavelength. Note that they are not Kramers-Kronig consistent (Appendix A).

7.1.1.1 Cauchy dispersion

Usually the Cauchy dispersion is only used for transparent materials. The refractive index is then given by (transparent Cauchy):

$$n(\lambda) = A + \frac{B \cdot 10^4}{\lambda^2} + \frac{C \cdot 10^9}{\lambda^4}$$



 $k(\lambda) = 0$

However, the DeltaPsi2 software allows to take a small absorption into account (absorbent Cauchy):

$$n(\lambda) = A + \frac{B \cdot 10^4}{\lambda^2} + \frac{C \cdot 10^9}{\lambda^4}$$
$$k(\lambda) = D \cdot 10^{-5} + \frac{E \cdot 10^4}{\lambda^4} + \frac{F \cdot 10^9}{\lambda^4}$$

$$\lambda^2 = \lambda^4$$

A, B, C, D, E and F are the fitting parameters.

These formulae can provide reasonable description of simple materials, but complicated dispersions cannot be determined and sometimes fitting of the coefficients does not converge to a good solution.

7.1.1.2 Sellmeier dispersion

This dispersion is very similar to the Cauchy dispersion. The expressions are:

$$\varepsilon = n^2(\lambda) = A + B \cdot \frac{\lambda^2}{\lambda^2 - \lambda_0^2}$$
 and $k(\lambda) = 0$ (transparent)

or

$$\varepsilon = n^{2}(\lambda) = \frac{1+A}{1+\frac{B\cdot10^{4}}{\lambda^{2}}} \text{ and } k(\lambda) = \frac{C}{n\cdot D\cdot10^{-2}\cdot\lambda + \frac{E\cdot10^{2}}{\lambda} + \frac{1}{\lambda^{3}}} \text{ (absorbent)}$$

7.1.1.3 Conrady

The Conrady dispersion is a variation of the Cauchy Transparent dispersion and can also be applied to transparent and simple materials:

$$n\!\left(\lambda\right)\!=A+\frac{B\!\cdot\!10^2}{\lambda}\!+\!\frac{C\!\cdot\!10^9}{\lambda^{3,5}} \text{ and } k\!\left(\lambda\right)\!=\!0$$

7.1.1.4 Briot (Schott)

This dispersion formula is based on mathematics of Abelian and elliptic functions:

$$n(\lambda) = A_0 + A_1 \cdot 10^{-2} \cdot \lambda^2 + \frac{A_2 \cdot 10^{-2}}{\lambda^2} + \frac{A_3 \cdot 10^{-4}}{\lambda^4} + \frac{A_4 \cdot 10^{-6}}{\lambda^6} + \frac{A_5 \cdot 10^{-7}}{\lambda^8}$$

$$\mathbf{k}(\lambda) = \mathbf{0}$$

Note that empirical dispersion relations such as Sellmeier, Cauchy, Conrady and Briot (Shott) have been widely used to describe transparent substrates and thin films till now. The constants in these formulae have no physical meaning.

7.1.1.5 Haltman

A further empirical model is described by the Haltman formula:

$$n(E) = A + \frac{C}{\lambda(E) - B}$$
$$k(E) = 0$$
$$\lambda(E) = \frac{h \cdot c}{E}$$

7.1.1.6 Fixed index

This model is mainly used to describe the dispersion of the ambient (air, liquid,...):

$$n(E) = const$$

 $k(E) = const$

7.1.2 Classical oscillator models

Classical oscillator models are derived from the equation of motion of a bound charge. In the DP2 software the classical oscillator model contains:

- Lorentz oscillator
- Drude oscillator
- high frequency constant
- and additional oscillators

7.1.2.1 Lorentz oscillator

A Lorentz oscillator describes an electron bound to a heavy ion by a "spring" with a restoring force.

The single Lorentz oscillator is mostly used to describe transparent dielectric materials ($\Gamma_0=0$) such as SiO₂, SiON, Al₂O₃, HfO₂, MgO or weakly absorbing materials ($\Gamma_0\neq 0$).

The formula for the Lorentz oscillator is:

$$\frac{\left(\epsilon_{s}-\epsilon_{\infty}\right)\omega_{t}^{2}}{\omega_{t}^{2}-\omega^{2}+i\Gamma_{0}\omega}$$

where is the light energy (eV), ε_{∞} is the dielectric constant for high frequencies, ε_s is the static dielectric constant, ω_t is the transition energy of the absorption process and Γ_0 is the damping constant.

For example in the next graph a Lorentz oscillator with $\varepsilon_{\infty}=1$, $\varepsilon_s=2$, $\varepsilon_t=3eV$ and $\Gamma_0=0.7$ is shown (Im(ε) in red and Re(ε) in blue):





7.1.2.2 Drude term

The Drude free electron model is used mainly for metals. It can be derived from the Lorentz oscillator model by assuming that the restoring force is zero ($\omega_t = 0$). That means that all electrons are free to move throughout the material. The equation for the Drude oscillator is given by:

$$\frac{\omega_{\rm p}^{2}}{-\omega^{2}+i\Gamma_{\rm D}\omega}$$

where is the light energy (eV), ω_p is the plasma energy of the free electron gas and Γ_D is the damping constant. The damping force describes the scattering of the electrons associated with electrical resistivity.

7.1.2.3 Additional oscillators

Sometimes it is necessary to use additional oscillators for example in the case of materials that have several absorption bands such as organic polymers.

These additional oscillators have the form of Lorentz oscillators and are given by:

$$\sum_{j=1}^{2} \frac{f_{j} \omega_{0j}^{2}}{\omega_{0j}^{2} - \omega^{2} + i\tau_{j} \omega}$$

where ω is the light energy (eV), f is the oscillator strength, ω_0 is the transition energy of the absorption process and is the damping constant.

7.1.2.4 Summary: classical formula

The equation of the classical formula in the DP2 software is therefore given by:

$$\varepsilon = \varepsilon_{\infty} + \frac{\left(\varepsilon_{s} - \varepsilon_{\infty}\right)\omega_{t}^{2}}{\omega_{t}^{2} - \omega^{2} + i\Gamma_{0}\omega} + \frac{\omega_{p}^{2}}{-\omega^{2} + i\Gamma_{D}\omega} + \sum_{j=1}^{2}\frac{f_{j}\omega_{0j}^{2}}{\omega_{0j}^{2} - \omega^{2} + i\tau_{j}\omega}$$

When the user checks the fitting box of the classical dispersion in the Model window, the software will ask the user to select the type(s) of oscillator:



7.1.2.5 Quatre Oscillators

This dispersion formula is just an extension of the double oscillator formula of the classical dispersion and is used for materials with several absorption bands such as polymers.

$$\varepsilon(\omega) = \varepsilon_{\infty} + \sum_{j=1}^{4} \frac{f_{j}\omega_{0j}^{2}}{\omega_{0j}^{2} - \omega^{2} + i\gamma_{j}\omega}$$

7.1.3 Oscillator models for amorphous materials

In thin film technologies a lot of materials exist in an amorphous or nearly amorphous structure. The optical properties of these materials can depend strongly on the growth conditions so that there can be a great variation of optical parameters. In the following we describe the most important dispersion models for this type of materials.

Common to these models is that there is an energy band gap below which there is no absorption.

7.1.3.1 New Amorphous

Probably the most commonly used dispersion formula for amorphous materials is based on a parameterization introduced by Forouhi & Bloomer:

$$n(\omega) = n_{\omega} + \frac{B(\omega - \omega_{j}) + C}{(\omega - \omega_{j}) + \Gamma_{j}^{2}}$$
$$k(\omega) = \begin{cases} \frac{f_{j}(\omega - \omega_{g})^{2}}{(\omega - \omega_{j}) + \Gamma_{j}^{2}} & \omega > \omega_{g} \\ 0 & \omega < \omega_{g} \end{cases}$$

where

$$B = \frac{f_j}{\Gamma_j} \left[\Gamma_j^2 - (\omega_j - \omega_g)^2 \right]$$
$$C = 2f_j \cdot \Gamma_j (\omega_j - \omega_g)$$

with the following fitting parameters:

- n_∞ the long wavelength (low energy) refractive index
- ω_i the energy of maximum absorption
- fj the oscillator strength
- ω_g the band gap energy
- Γ_i the broadening factor

The New Amorphous dispersion formula, which can be regarded as a single oscillator above a certain energy band gap below which there is no absorption, can be extended to allow 2 or 3 oscillators above the band gap:

- Double New Amorphous
- Triple New Amorphous

The fitting parameters are thus n_{∞} , ω_g , f_j , ω_j , Γ_j with $1 \le j \le 3$.

This is necessary if there are additional absorption bands present in the material.

7.1.3.2 Amorphous

This dispersion is based on an earlier parametrization found by Forouhi & Bloomer:

$$n(E) = \sqrt{\varepsilon_{\infty}} + \frac{B_0 \cdot E + C_0}{E^2 - B \cdot E + C'}$$

$$k(E) = \begin{cases} \frac{A(E - E_g)^2}{E^2 - B \cdot E + C} & E > E_g \\ 0 & E < E_g \end{cases}$$

where

$$B_{0} = \frac{A}{Q} \left(-\frac{B^{2}}{2} + E_{g} \cdot B + C \right)$$
$$C_{0} = \frac{A}{Q} \left[\left(E_{g}^{2} + C \right) \frac{B}{2} - 2 \cdot E_{g} \cdot C \right]$$
$$Q = \frac{1}{2} \sqrt{4 \cdot C - B^{2}}$$

The fitting parameters are: ε_{∞} , E_g , A, B and C. The formulae can be extended to use up to 4 oscillators (Double-, Triple- and Quadruple Amorphous).

The drawback is that not all parameters have a proper physical meaning. ε_{∞} is the high frequency dielectric constant and E_g is the energy band gap. A is simply a factor but B and C are expressions of physical parameters:

$$B = 2E_0$$
 and $C = E_0^2 + \Gamma^2$

This makes it very difficult to guess proper start values for the dispersion.

The recommendation is to use the New Amorphous formula which is much more intuitive.

Although the Amorphous or New Amorphous parameterization works for a lot of materials and is very useful for spectroscopic ellipsometry there are some physical problems:

- the approximation k(E) = 0 for $E < E_g$ is not physically realistic
- for E >> E_g the (New) Amorphous formula predicts k(E) = constant which of course cannot be true for a real material.

A consequence is that these dispersions are not Kramers-Kronig consistent.

7.1.3.3 Tauc Lorentz formula

The Tauc Lorentz dispersion formula corrects the unphysical problems of the (New) Amorphous dispersion. This formula is based on the Tauc expression and Lorentz oscillator model. The imaginary part of the dielectric function near the band edge is given by:

$$\varepsilon_{2}(E) = 2n(E)k(E) = \begin{cases} \frac{1}{E} \cdot \frac{A \cdot E_{0} \cdot C(E - Eg)^{2}}{(E^{2} - E_{0}^{2})^{2} + C^{2} \cdot E^{2}} & E > E_{g} \\ 0 & E < E_{g} \end{cases}$$

There are 5 fitting parameters: the energy band gap E_g the peak in the joint density of states E_0 , the high frequency dielectric constant ε_{∞} in the ε_1 expression (see below), the factor A, which includes the optical transition matrix elements, and C the broadening factor.

The real part ε_1 is determined from ε_2 with the Kramers-Kronig relation:

$$\varepsilon_1(\mathbf{E}) = \varepsilon_{\infty} + \frac{2}{\pi} \cdot \mathbf{P} \int_{\mathbf{E}_g}^{\infty} \frac{\xi \varepsilon_2(\xi)}{\xi^2 - \mathbf{E}^2} d\xi$$

For details on this calculation refer to appendix A.



The Tauc Lorentz expression is consistent with known physical phenomena and is obviously Kramers-Kronig consistent.

7.1.4 Transition based dispersion models

These dispersion models are mainly used to describe transitions near the band gap in semiconductors. These are mostly band-to-band and excitonic transitions.

7.1.4.1 Adachi Model Dielectric Function (MDF)

The dielectric function of a material is composed by contributions from all transitions between the valence band and the conduction band.

The Adachi-MDF is mainly applied to transitions appearing in semiconductors near the band gap, namely E_0 , E_0+_0 , E_1 , E_1+_1 .

The complex dielectric function of the MDF model is composed of different terms corresponding to these transitions:

$$\varepsilon = \varepsilon_{\infty} + \varepsilon_1 + \varepsilon_2 + \varepsilon_3$$

The assumption is again that for energies lower than the fundamental band gap the absorption is zero: Im (ϵ) = 0 for E < E₀

For the transitions at E_0 and at $E_0 + \Delta_0$ we get:

$$\varepsilon_{1} = \mathbf{A}_{0} \left(\frac{f(\chi_{0})}{E_{0}^{3/2}} + \frac{f(\chi_{s0})}{2(E_{0} + \Delta_{0})^{3/2}} \right)$$

with

$$f(\chi) = \frac{2 - \sqrt{1 + \chi} - \sqrt{1 - \chi}}{\chi^2}$$
$$\chi_0 = \frac{E + i\Gamma_0}{E_0}$$
$$\chi_{s0} = \frac{E + i\Gamma_0}{E_0 + \Delta_0}$$

The contribution of the excitonic transitions is described by:

$$\varepsilon_{2} = \sum_{n=1}^{3} \frac{A_{0x}}{n^{3}} \left(\frac{1}{E_{0} - \left(\frac{G_{0}}{n^{2}}\right) - E - i\Gamma_{0}} \right)$$

In theory, the sum is done for an infinite number of excitonic states, but most of the time, only the first excitonic states are observable. The others are included in the band to band contribution due to their proximity.

For transitions E_1 and $E_1 + \frac{1}{1}$ the contribution is:

$$\varepsilon_3 = -\frac{\mathbf{B}_1}{\xi^2} \cdot \ln(1-\xi^2)$$

with

$$\xi = \frac{\mathbf{E} + \mathbf{i}\Gamma_1}{\mathbf{E}_1}$$

The fitting parameters are A_0 , A_{0x} , B_1 , G_0 , E_0 , E_1 , Γ_0 , Γ_1 , Δ_0 and ϵ_{∞} .

7.1.4.2 Excitonic model

This model describes the contribution of excitons at the transitions E_0 , $E_0+\Delta_0$, E_1 and $E_1+\Delta_1$.

$$\begin{split} & \varepsilon = \varepsilon_{ex,0} + \varepsilon_{ex,1} \\ & \varepsilon_{ex,0} = \sum_{n=1}^{\infty} \frac{A_{0x}}{n^3} \left[\frac{1}{E_0 - \left(\frac{G_0}{n^2}\right) - E - i\Gamma_0} + \frac{1}{2} \left(\frac{1}{E_0 + \Delta_0 - \left(\frac{G_0}{n^2}\right) - E - i\Gamma_0} \right) \right] \\ & \varepsilon_{ex,1} = \sum_{n=1}^{\infty} \frac{1}{(2n-1)^3} \left[\frac{\frac{B_{1x}}{E_1 - \left[\frac{G_1}{(2n-1)^2}\right] - E - i\Gamma_0}}{E_1 - \left[\frac{G_{1x}}{(2n-1)^2}\right] - E - i\Gamma_0} \right] \\ & + \frac{B_{2x}}{E_1 + \Delta_1 - \left[\frac{G_{1s}}{(2n-1)^2}\right] - E - i\Gamma_0} \end{split}$$

With the following parameters:



ε _m	High frequency
~	
E_0	E_0 transition
D_0 , also called Δ_0	$E_0 + D_0$ transition
G ₀	Broadening parameter
A_0	Strength parameter
A_{0x}	Exciton strength parameter
G_0	Exciton binding parameter
E ₁	E ₁ transition
D_1 , also called Δ_1	E_1+D_1 transition
G ₁	Broadening parameter
B ₁	Strength parameter
B_2	Strength parameter
B _{1x}	Exciton strength parameter (E_1)
B _{2x}	Exciton strength parameter (E_1+D_1)
G ₁	Exciton binding parameter
G _{1s}	Exciton binding parameter
E ₂	E2 transition
C ₀	Strength parameter
G ₂	Broadening parameter

7.1.4.3 Tanguy model

The tanguy dispersion formula is used for alloyed III-V-semiconductors such as InGaAsP to describe the transitions near the band gap energy. The formula of the Tanguy model is:

$$\begin{split} \epsilon(E) &= \epsilon_0 + \frac{a}{b - (E - i\Gamma)^2} \\ &+ \frac{A\sqrt{R}}{E + i\Gamma} \left[ln \frac{E_g^2}{E_g^2 - (E - i\Gamma)^2} + 2\sqrt{\frac{E_g}{R}} - \sqrt{\frac{E_g + E + i\Gamma}{R}} - \sqrt{\frac{E_g - E - i\Gamma}{R}} \right] \\ &+ \frac{A\sqrt{R}}{E + i\Gamma} \left[+ 2\sum_{k=1}^{\infty} \left(\frac{2}{\sqrt{\frac{R}{E_g}} - k} - \frac{1}{\sqrt{\frac{R}{E_g} + E + i\Gamma}} - k - \frac{1}{\sqrt{\frac{R}{E_g} - E - i\Gamma}} - k \right) \right] \end{split}$$

The fitting parameters are:

- ϵ_0 : low frequency dielectric constant
- Eg: band gap energy
- Γ : broadening parameter
- a, b: dispersion parameters for $E \le E_g$
- c, d: dispersion parameters for k at E > Eg
- R: energy of exciton
- A: coefficient proportional to the square of the Kane momentum



7.1.4.4 Tanguy Extended Model

The Tanguy Extended dispersion formula is used as the Tanguy formula, however, it is mainly applied in the near-IR spectral region. The formula is:

$$\varepsilon(E) = 1 + \frac{a}{b - E^2} + \frac{A\sqrt{R}}{E + i\Gamma} \left[g[\zeta(E + i\Gamma)] + g[\zeta(-E - i\Gamma)] - 2g[\zeta(0)] \right]$$

with

$$g(\zeta) = 2\ln(\zeta) - 2\pi \cdot \cot(\pi\zeta) - 2\Psi(\zeta) - \frac{1}{\zeta}$$
$$\Psi(z) = \frac{d}{dz} \ln[\Gamma(z)]$$
$$\zeta(z) = \sqrt{\frac{R}{E_g - z}}$$

so that we obtain:

$$n = \operatorname{Re}(\sqrt{\varepsilon})$$

$$k = \begin{cases} \operatorname{Im}(\sqrt{\varepsilon}) & \operatorname{E} < \operatorname{E}_{g} \\ \operatorname{Im}(\sqrt{\varepsilon}) + c(\operatorname{E} - \operatorname{E}_{g}) + d(\operatorname{E} - \operatorname{E}_{g})^{2} & \operatorname{E} > \operatorname{E}_{g} \end{cases}$$

The fitting parameters are the same as the one from Tanguy formula

- ε_0 : low frequency dielectric constant
- Eg: band gap energy
- Γ: broadening parameter
- a, b: dispersion parameters for $E < E_g$
- c, d: dispersion parameters for k at E > Eg
- R: energy of exciton
- A: coefficient proportional to the square of the Kane momentum

7.1.5 Further models and combination of previous models

There are further dispersion models implemented in the DP2 software. Some are extensions or combinations of already discussed models. The most important are:

- Afromowitz
- Kato-Adachi
- Kato-Adachi IR
- Adachi-n Forouhi
- Adachi-New Forouhi

They are all described in detail in appendix A!



7.1.6 Summary

The following table summarizes again the discussed dispersion models and their applications:

Dispersion type	Dispersion	Materials	Fitting
2 isperioron type	21000101		parameter
		Transparent or weakly	
		absorbent materials: SiO ₂ ,	
Classical	classical	SiON, Al ₂ O ₃ , HfO ₂ , MgO,	4 up to12
oscillators		ZrO ₂ .	
		Metals with Drude Oscillator	
	Quatre oscillators	Polymers, TPD materials	9
	Cauchy	Transparent or weakly	3 up to 6
	Sellmeier	absorbent thin materials: glass,	3 up to 5
Polynomial		ullii 310 ₂ lliilis	2
	Conrady	Transparent and simple	3
	Briot	materials	6
	Haltman		3
	New Amorphous	Al_2O_3 , TiO_2 , Ta_2O_5	5 up to 11
	Amorphous	Transparent or complex	
Amorphous		absorbent materials, Alq3,	5 up to 14
		AlN, pentacene	
	Tauc Lorentz	a-Si _x N _y :H,	5 up to 11
	Excitonic		19
		Semiconductors II-VI, ZnTe,	
	Adachi-MDF	HgCdTe	8
T :: 1 1	75	Semiconductors III-V, InGaP	7
I ransition-based	Tanguy	near band gap energy	/
	Tangay Extended	Semiconductors III-V in the	0
	Tanguy Extended	IR region of the spectra	0
	Afromovitz	Semiconductors III-V	6



	Kato-Adachi	Semiconductors III-V till 3eV	11
Combination of	Kato-Adachi IR		8
previous	Adachi n Fourouhi		8 up to 14
dispersion	Adachi-New	PZT: lead zirconate titanate,	0
	Fourouhi		0

7.1.7 Dispersion parameter manual adjustment - Examples

During the fitting process the DP2 software automatically adjusts dispersion parameters to obtain the best fit. In complex cases the solution of the fitting process might be far from the best solution. Therefore it is crucial to set the starting parameters for the dispersion models as close to reality as possible.

For this reason it is important to get the maximum of information of the optical properties of the material to be modelled by consulting literature values and/or the reference files of the DP2 software.

In the next chapters dispersion models and corresponding start values for various materials are discussed.

7.1.7.1 Classical/Drude Oscillator dispersion for Aluminium

The following graph shows in the dispersion window:

- n of the Drude Oscillator in blue
- k of the Drude Oscillator in red
- n of the Aluminium reference file in green
- k of the Aluminium reference file in black



The above graph is obtained when you open a classical dispersion with the aluminium reference superposed on it. The starting fitting parameters are:



E^{∞}	1.0000000
ε _s	2.5000000
ω_t	12.0000000
ω _p	0.0000000
Γ_0	5.0000000
Γ _d	0.0000000
F ₁	0.0000000
ω_{01}	0.0000000
y ₁	0.0000000
F ₂	0.0000000
ω ₀₂	0.0000000
Y ₂	0.0000000

Aluminium is a metal, so we use the Drude Oscillator dispersion whose parameters are the plasma frequency ω_p and the damping constant Γ_d . ϵ_{∞} is generally set to 1 and we do not fit on it. All other parameters, except for ω_p , Γ_d and ϵ_{∞} are set to 0.

 ω_p and Γ_d are now adjusted to match the reference curve. The following set of parameters is a good guess for matching the dispersion with the reference from 2.5eV to 6eV:

ε _∞	1.0000000
ω _p	15.6500000
$\Gamma_{\rm d}$	0.6200000



9/2/08

Aluminium, like all metals, is strongly absorbent in the IR. Usually it does not matter if the dispersion is not appropriate in the IR region, because most of time the visible and UV region are of interest. However, if the IR region is important a more sophisticated dispersion formula is necessary.

7.1.7.2 New Amorphous dispersion for TiO₂

The following graph shows in the dispersion window:

- n of New Amorphous in blue
- k of New Amorphous in red
- n of TiO₂ reference file in green
- k of TiO2 reference file in black



The goal is to adjust parameters so that the dispersion matches the reference. Initial parameters of the New Amorphous dispersion are for example:

$$\begin{split} \mathbf{n}_{\infty} &= 1.0000000\\ \mathbf{\omega}_{g} &= 3.0000000\\ \mathbf{f}_{j} &= 0.1000000\\ \mathbf{\omega}_{j} &= 5.0000000\\ \boldsymbol{\Gamma}_{j} &= 0.5000000 \end{split}$$

Look at n of the reference file near the lowest energy: n(1.5eV) = 2.265. We can put this value to n_{∞} .





According to the reference there is no absorption (k = 0); That means that we have to set ω_j and ω_g high enough so that there is no absorption in the given spectral range. For example set $\omega_g = 4.9$ eV and $\omega_j = 6$ eV; by definition $\omega_g < \omega_j$. You will obtain:



Then adjust the strength and the broadening parameter. An increasing oscillator strength leads to a higher value of n. A decreasing broadening parameter leads to a steeper increase of n with energy. Set for example $f_j = 0.25$ and $\Gamma_j = 0.10$ eV thus we obtain a correct curve profile:



Finally we adjust again the value of n. For example set n = 1.55:



The final start values in this case are then:

 $n_{\infty} = 1.5500000$ $\omega_g = 4.9000000$ $f_j = 0.2500000$ $\omega_j = 6.0000000$

Explore the future

9/2/08

$\Gamma_{i} = 0.1000000$

It does not matter if the starting parameters lead to a dispersion that does not exactly match the reference values because they will be adjusted in the fitting process. Real TiO₂ samples can show different optical properties for example they can show absorption for lower energies which means lower values for ω_g and ω_j . For example for one of our samples we found after the fitting process the following parameters:

$$\begin{split} \mathbf{n}_{\infty} &= 1.9922565 \\ \boldsymbol{\omega}_{g} &= 3.0720084 \\ \mathbf{f}_{j} &= 0.1764574 \\ \boldsymbol{\omega}_{j} &= 4.2158175 \\ \boldsymbol{\Gamma}_{j} &= 0.4805436 \end{split}$$

7.1.8 Effective medium approximation

Effective medium approximations (EMAs) are used to describe the optical constants of mixtures of materials of known optical constants.

EMAs are applied successfully when inhomogeneities are at scale somewhere in between atomic distances and the wavelength. The most important applications are:

- surface roughness layers
- interface layers
- porous materials
- mixture of materials

7.1.8.1 The effective medium models

The DP2 software uses several models of effective medium, according to the phase proportions in the mixture:

- MG for Maxwell-Garnett model
- EMA for Bruggeman model
- EMAMG for Maxwell-Garnett & Bruggeman model
- SSC for Sen, Scala & Cohen model

Maxwell-Garnett model

This model is based on the Clausius-Mosetti relationship which describes the dielectric function of a set of spheres that are at a certain distance to each other and homogeneously immerged in a host medium. This model defines the dielectric function ε of the effective medium according to the following formula:

$$\varepsilon = \varepsilon_0 \frac{\varepsilon_1 (1+2f) + 2\varepsilon_0 (1-f)}{\varepsilon_1 (1-f) + \varepsilon_0 (2+f)}$$

With the dielectric constant of spherical particles ε_1 immerged in a host medium whose dielectric constant is ε_0 . f is the volume fraction of the spheres.



The Maxwell-Garnett model is valid in cases where there are inhomogeneities in granular form. It is only valid for a two-component mixture. It is for example very useful for certain nanocrystals embedded in hosts and for cermats. However, the proportion of the inclusions in the host medium must be low, otherwise the Bruggeman model, which is discussed next, must be used.

EMA-Bruggeman model

In this model the different components of the mixture are treated equivalently, without preliminary assumption on the relative proportions. This model is self-consistent: the mixture of the different materials forms the host medium, which means that the dielectric function of the host medium is the EMA-Bruggeman dielectric function.

In this model the number of phases can exceed 2. The dielectric functions of the different materials are ε_i (with i=1,2,3,...) with corresponding volume fractions f_i . The dielectric function of the effective medium fulfills the following equation:

$$\sum_{i} f_{i} \frac{\varepsilon_{i} - \varepsilon}{\varepsilon_{i} + 2\varepsilon} = 0$$

This polynomial equation has a degree equal to the number of material components. For example, in the case where i = 2, $f_2 = 1-f_1$ and the following equation has to be solved:

$$-2\varepsilon^{2} + [3f_{1}(\varepsilon_{1} - \varepsilon_{2}) + 2\varepsilon_{2} - \varepsilon_{1}]\varepsilon + \varepsilon_{1}\varepsilon_{2} = 0$$

In the DP2 software the number of components per layer is limited to three to avoid complicated calculi and eventual correlations between the different parameters.

This model is very useful for modelling surface roughness consisting of 50% void and 50% material of the layer below.

EMAMG

The EMAMG model is a combination of the Maxwell-Garnett model and the Bruggeman model.

SSC model

Sen & al presented a self similar model based on the following equation:

$$\frac{\varepsilon - \varepsilon_0}{\varepsilon + 2\varepsilon_0} = f \frac{\varepsilon_1 - \varepsilon_0}{\varepsilon_1 + 2\varepsilon_0}$$

The derivation provided by Sen & al results in:

$$\left(\frac{\varepsilon_1 - \varepsilon}{\varepsilon_1 - \varepsilon_0}\right) \cdot \left(\frac{\varepsilon_0}{\varepsilon}\right)^{1/3} = \phi$$

with ϕ the proportion of the host medium

This model is applied to a fractal medium of infinitely wide particle size distribution. For example it is useful for modelling real porous systems, which show a wide size distribution of the void inclusions.

7.1.8.2 Effective medium approximations in DP2

To create an effective medium in DP2 a layer has to be created that contains the host medium and one or two additional materials. The materials can be assigned to the layer simply by "drag and drop" from the material folders in the User or Application library:

- User Library > Materials if materials are created by the user
- *Application Library > Materials* if already existing materials are used

If there are more than one material assigned to a layer there will be the additional fitting parameter "%", which is the proportion of the different materials. By checking the proportion fit box this parameter will be adjusted during the fitting process.

By default, a mixture of two materials is set to 50% of material A and 50% of material B. To adjust the proportion click on any % symbol displayed on the layer and the following window appears *Fitting choice Properties for layer # >Materials* :

Fitting Choice Properties for layer #2	×					
Thickness Materials Special Routine Checks						
Fit volume fraction						
Fit Material name	Volume fraction					
Param. TiO2.dsp	50.00 %					
Param. void.dsp	50.00 %					
Effective medium method						
EMA Screening factor :	2.0000					
Volume fraction fitting mode for first material						
• Nominal value (above value)						
C Multiguess absolute :						
O Multiguess relative :						
O Multistart absolute :						
O Multistart relative :						
C Correlated :						
Ok Cancel						

Volume fraction: the proportions can be adjusted; for a three component mixture the same window with three values of proportion will appear.

Material name: the different materials that compose the layer are shown with their respective fit box.

Volume fraction fitting mode: the different fitting functions on the proportion can be chosen.

Effective medium method: the method for the effective medium approximation can be selected.

In the following sections some examples for the application of EMAs are shown.



7.1.8.3 Examples for EMA

Roughness layer

In the case of TiO_2 layer on a glass substrate we start with a simple model using a New Amorphous dispersion for the TiO_2 layer:



The acquired spectra (dots) and the fit (lines) are shown in the graph below:



The results of the fit are:

d = 1321.9Å for the thickness and

 χ^2 = 12.88 for the goodness of fit

Now we add a roughness layer on top of the TiO_2 layer. This roughness layer is represented by a mixture of 50% TiO_2 and 50% void with a starting thickness of 30Å:

2	F	36.3 🗌 %	F TiO2.dsp	50.00 % × Void.dsp	50.00 % × ×
1	F	1306.2	F TiO2.dsp		××
S			Glass.spe		×

The acquired spectra (dots) and the fit (lines) are shown in the graph below:





The results are now:

d = 1306.2Å for the thickness of the TiO₂ layer,

 $d=36.3 \text{\AA}$ for the thickness of the roughness layer and

 χ^2 =1.23 for the goodness of fit (ten times better than without roughness layer!).

Interface layer

An interface layer may form between two layers for example by diffusion (mixture) or the given surface structure (hooking):

- material composite with portion of material B < portion of material A



The layer model would in this case look like the following:

3	F	1000.0	F B.dsp		×	×
2	F	30.0 🗌 %	F A.dsp	80.00 % × FB.dsp	20.00 % ×	×
1	F	1000.0	F A.dsp		×	×

Once a good fit is obtained with a fixed proportion, one can try to fit on the proportion to further reduce the value of χ^2 .

- hooking layer with portion of material B = portion of material A


3	F	1000.0	FB.dsp	2	< X
2	F	30.0 🗌 %	FA.dsp	50.00 % × FB.dsp 50.00 % ×	< X
1	F	1000.0	FA.dsp	Σ	< X

Also in this case one can try to fit on the proportion to further reduce the value of χ^2 .

Porous layer

If the porous material A is embedded between the substrate and material B the model would look like:

2	F	1000.0	FB.dsp			×	×
1	F	1000.0 F %	FA.dsp	50.00 % ×	Void_asp.ref	50.00 % ×	×
S			Substrate.spe			×	

The layer with the porous material A is modelled by: material A & Void. It is recommended to set the proportion as fitting parameter.

The situation is similar for a material where the optical properties depend on thickness. A rigorous treatment of this case is done by choosing a "gradient layer" but mostly a 2 layer model is enough. For example if the optical index increase from bottom to top the model looks like:

2	F	1000.0	FA.dsp			×	×
1	F	1000.0 F %	FA.dsp	50.00 % ×	Void_asp.ref	50.00 % ×	×
S			Substrate.spe			×	

The material is modelled by two layers: the normal layer and an effective medium layer with material A & Void. A fit on the proportion is also recommended. This model can equally be used for the inverse case by exchanging layer 1 and 2.

7.2 Back side reflections

7.2.1 Transparent Substrate

In many cases the analyzed sample including the substrate is transparent for at least a part of the spectrum. If the backside of the sample is in addition polished then back side reflection phenomena can occur.





Assuming the optical properties of the transparent material are isotropic we can distinguish two cases:



Case 1: The substrate was measured with its backside reflections

Use the "backside reflection" model which allows the optical treatment of the backside reflections collected during the measurement:

Use void as the substrate and then add a layer that represents the transparent substrate (assign the correct thickness to this layer!). Then add the layer structure as usual. An example is shown below:

3 F 28.0 🗌 %	F sinb.dsp	50.00 % × 🔽 🗌 void.dsp	50.00 % ×
2 F 1782.6	F sinb.dsp		××
1 500000.0	Bk7.ref		×>
2	Void.ref		×

The backside reflection model can be optimized by setting the total number of backside reflected beams and the percentage of the first and the last collected beam with the *limited backside reflection* function.

This function can be accessed by clicking on the thickness of layer 1 and then from the fitting choice properties tab by clicking on L.B.R. Check the box "use limited backside reflection".

Fitting Choice	e Properties	for layer #1			×
Thickness Limited ba I Use lim	Materials ckside refle ited backsi	Special functions ection de reflection	L.B.R.	Checks	
Number of	reflections	:			
☐ Fit	 Nomi Multi Multi 	inal value : guess : start :			2
First beam	number :				
⊢ Fit	 Nomi Multi Multi 	inal value : guess : start :			0
First beam	1%:	100.00			
Last beam	%:	75.00			

We advise to use the limited backside reflection function for thin transparent substrates of ca. 1mm thickness.

Reasonable values for the settings are:

- Number of reflections: nominal values = 2
- First beam number: nominal value = 0
- This means that the beam reflected from the surface and the first beam reflected from the backside are collected. In general do not fit the number of reflections.
- First beam % = 100
- Last beam % = 75 (can be adjusted manually from ca. 70% to 80% to improve the fit)

Case 2: The substrate was measured without backside reflections

This can be achieved by roughening the backside surface or if the backside reflected beam is spatially separated from the first reflection by masking it.



Sample where the bottom of the substrate is roughened

In this case a simple substrate containing the reference or dispersion file of the transparent material can be used as the substrate, for example:

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HORIBA

F 500.0	znonew.dsp	××
S	Bk7.ref	x

7.3 Advanced layer structures

Until now only isotropic layers with constant optical dispersion through the layer were treated. These layers are called "simple layers". However, layers can be anisotropic or can show a dependency of optical parameters on depth.

A layer can be added to an existing layer model by clicking with the right mouse bottom above the existing layers. The DP2 software allows to choose between the following types of layers:

Thickness unit : A Image: Clear results before fit Image: Clear graph before simulation Image: Show all fitting steps A01 : T0.000 * Image: Exp. File : Image: Show all fitting steps						
Add Paste material from clipboa Paste layer from clipboard	 Simple layer Gradient layer Anisotropic Uniaxial layer Anisotropic Biaxial layer Grad.Anisotropic Uniaxial layer Grad.Anisotropic Biaxial layer 					
S C-si_	isa.ref					
Add The Edit Add The Edit Add	Image: Bold in the second s					

- Simple layer: isotropic layer with homogeneous optical dispersion
- Gradient layer: isotropic layer with depth dependency of the optical dispersion
- Anisotropic Uniaxial layer: layer with uniaxial anisotropy and homogeneous optical dispersion
- Anisotropic Biaxial layer: layer with biaxial anisotropy and homogeneous optical dispersion
- Gradient Anisotropic Uniaxial layer: layer with uniaxial anisotropy and depth dependency of the optical dispersion
- Gradient Anisotropic Biaxial layer: layer with biaxial anisotropy and depth dependency of the optical dispersion

7.3.1 Graded Optical Constants

Graded optical constants are used to describe a gradual variation of optical constants (n and k) through a film. The refractive index at the top of the layer can thereby be larger or smaller than the index at the bottom of the layer. The index profile can be linear or exponential.

To add a gradient layer click right in the model area and then *add Gradient layer*.

The example below describes a TiO_2 layer as a gradient layer.

First we use "void" inside one of the two layers (top & bottom) to vary the index of refraction. It can be used with reference or dispersion materials.

The index at the top of the layer is higher than the one at the bottom:



The index at the bottom of the layer is larger than the one at the bottom:

	tio2.dsp	80.00 % × Void.ref	20.00 % × ×
Lin B	tio2.dsp		X
S	Csi_wor.ref		×

Alternatively the same dispersion formula can be used as a starting point but with two different names allowing the fitting process to provide two different dispersions, one at the top and the other one at the bottom of the layer:

	Т	tio2-top.dsp	×	x
Lin	В	tio2-bottom.dsp	x	
S		Csi_wor.ref	X	

A gradient layer can be used if a homogeneous layer model provides a large χ^2 and adding a roughness layer is not enough to provide a satisfactory χ^2 .

The example below (TiO₂ layer on a glass substrate) shows how χ^2 can be successively improved by adding a gradient and a roughness layer:

Results with a single layer model:

 $\chi^2 = 13.24; d= 1317 \text{ Å}$

Results with a single layer + roughness layer:

 $\chi^2 = 2.31$; thickness (TiO₂) = 1297Å; thickness (roughness) = 34Å

Results with a graded layer:

 $\chi^2 = 4.1$; thickness = 1293Å

Results with a graded layer + roughness layer:

 $\chi^2 = 0.94$; thickness (graded layer) = 1304Å; thickness (roughness) = 34Å

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The final model leads to a nearly perfect fit as can be seen in the graph below:



With the following "graded" optical constants for TiO₂:



7.3.2 Optical Anisotropy

A material that is anisotropic has different optical properties in different directions. At the opposite an isotropic material has the same optical properties in all directions.

In the most general case the dielectric tensor is a rank-2 tensor (3 x 3 matrix) in eigenaxes coordinates. This tensor is diagonal. The three distinct refractive indices are n_x , n_y and n_z .

$$\varepsilon = \begin{bmatrix} n_x^2 & 0 & 0 \\ 0 & n_y^2 & 0 \\ 0 & 0 & n_z^2 \end{bmatrix}$$

Anisotropy is an intrinsic material property. Reasons are non-symmetric atomic or molecular structures of long range order throughout the material. Common anisotropic materials are non-cubic crystals such as sapphire, rutile, calcite, lithium niobate and ordered polymers or organic materials like liquid crystal films.

A sign of anisotropy is asymmetric interference fringes in the spectra.

In DP2 we define the laboratory coordinates by (N, S, P): N describing the direction of anisotropy perpendicular to the sample surface, S perpendicular to the plane of incidence and P parallel to the plane of incidence.

The *Euler angles*, and are used to describe the relative orientation of the two coordinate systems (n_x, n_y, n_z) and (N, S, P).



An *axis of anisotropy* is defined as the axis along which isotropy is broken (or an axis of symmetry such as normal to crystalline layers). Anisotropic materials can have one or two such optical axes.

An *uniaxial material* has 2 different optical constants ($n_x = n_y n_z$) which are also called extraordinary (E) or ordinary (O). The birefringence magnitude is then defined by $n = n_e - n_o$.

A *biaxial material* has 2 axes of anisotropy and thus 3 different optical constants $(n_x n_y n_z)$.

There are three simple cases of uniaxial anisotropy in DP2 when the optical axis (i.e. n_e) is parallel to N, S or P. In all other cases (ne not aligned with one laboratory axis N, S or P or for biaxial anisotropy) the Euler angles have to be set.

To add an anisotropic layer into your model click right in the model area, and then *add Anisotropic uniaxial* or *biaxial layer*.





Note that the anisotropic material can be used also as a substrate:

-	Fitting parameters				
e	Copy substrate to clipboard		escription	Simulation	Save
j.	Paste substrate from clipboard		👘 Edit	Fit	Save As
	Change to	•	Anisotropi	ic Uniaxial sub	strate
			Anisotropi	c Biaxial subs	trate

A uniaxial anisotropic layer model would look like this then:



Modeling an uniaxial anisotropic layer

Modeling an anisotropic layer or substrate mostly relies on the identification of the direction of the axis of anisotropy from "pre-measurements":

- if two measurements at the same point on the sample rotated by 90° around its normal provide the same results the axis of anisotropy is N.
- if two measurements at the same point on the sample rotated by 90° provide different results the axis of anisotropy perpendicular to N (in-plane anisotropy).

There are 3 methods to identify the direction of the axis of anisotropy in the case of inplane anisotropy in order to uncouple S and P directions.

1. using the MM-16 rotate the sample to annul the off-diagonal elements of the Mueller Matrix: mij = 0

 $\begin{bmatrix} M_{11} & M_{12} & m_{13} & m_{14} \\ M_{21} & M_{22} & m_{23} & m_{24} \\ m_{31} & m_{32} & M_{33} & M_{34} \\ m_{41} & m_{42} & M_{43} & M_{44} \end{bmatrix}$

- 2. using the UVISEL in the configuration M = 0 and A = 0 rotate the sample to annul R and R₂ (R = R₂ = 0).
- 3. if the sample is transparent a birefringent sample will affect the light extinction between crossed polarizers.

Modeling an biaxial anisotropic layer

In DP2 an biaxial anisotropic layer can be described by 3 orthogonal optical properties:

2	Ζ	Flayer z.dsp	×	×
F 1000.0	X	Flayer x.dsp	×	
,⊢P	Y	Flayer y.dsp	×	
1 F 1000.0		Sio2_isa.ref	×	×
S		C-si_isa.ref	×	



The orientation of (nx, ny, nz) in the laboratory coordinates (N, S, P) is expressed by the Euler angles. Click on the thickness of the biaxial anisotropic layer. In the fitting choice properties tab select Euler angles:



For simplicity try to describe the material with the first two Euler angles Φ and Θ .







 Φ rotates the in-plane orientation around the sample normal:

 Θ tips the Z-direction away from the sample normal:



7.3.3 Periodic structure

A periodic structure is a repetition of one to x layers.

For example if there is 20 times the layer stack (SiO_2/TiO_2) on Si it is not necessary to fit the thickness of the 40 individual layers. One can assume that the all SiO_2 and all TiO_2 have the same thickness and thus reduce the number of unknown parameters drastically.

The most important applications are mirrors, anti-reflective coatings (ARC) and multiquantum wells.

To model a periodic structure click on the blue number of the layer and select *Create rep-etition*. A default repetition value of 2 appears in the selected layer. You can change it by clicking twice on it. By using the horizontal lines of the repetition feature you are able to select the layers involved in the periodic structure.

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Fitting parameters	
Delete layer	
Copy layer to clipboard Paste layer from clipboard	
Create Repetition	
Change to	•

A typical periodic structure would look like:

	2	1000.0	F tio2.dsp	×	×
Ĩ	1 E	1000.0	Sio2_isa.ref	×	×
	S		C-si_isa.ref	×	

7.4 Bound Multimodels

A bound multimodel is useful to bind several experimental data types from the same sample or different samples that exhibit at least one common physical property. Combining data provides generally the advantage to reduce strong parameter correlation in the optical modeling process.

To create a bound multimodel click on «Create a new model...» tab. Make sure «User Library» is selected and then click on the «Bound MultiModel» button.



7.4.1 Multiple data for the same sample

It is generally advantageous to collect as much independent experimental data as possible. Reflectance, transmittance and ellipsometric data all provide complementary information.

It is important to make sure that data were acquired from the same location on the sample under analysis.

As an example we show the combined analysis of transmittance and ellipsometric data:

Usually first a measurement is performed in the ellipsometric mode. Then both optical heads are moved to 90° to perform the second measurement in the transmittance mode.

Build one model for each measurement configuration and set the fitting parameters for each model.

The model for the ellipsometric mode has to contain the corresponding experimental file and the right angle of incidence. Make also sure that the right spectral range for the fitting and the right fitting parameters are chosen (Is, Ic) or (Ψ, Δ) . By default, the experimental file full spectral range and (Is, Ic) data are selected.



Note that in this example the ellipsometric measurement of the sample has been performed with a mask avoiding the measurements of backside reflections.

The model for the transmittance mode has to contain again the corresponding experimental file and the angle of incidence set to 0° (do not confuse with the 90° of the ellipsometer heads, here this is the angle of the sample in the plane of incidence!). Make also sure that the right spectral range for the fitting is chosen and that the fitting parameter is T.



Then create a bound multimodel and drag and drop the two model files in the blank area in the right of the window. The first model is called M1 and the second M2.

🖉 Bound HultiHodel - Noname2, been	KD.	
Bound Multimodel	Peruks	
Drop model files in the list below . 6 🐐		
M1 - T_ZnP(Nc22).mdl M2 - ZnP(Nc22).mdl	Left: X: BigH: Y	_
Experimental lile (Flexultr Vacquisition Data: LBV2rPNs:2214 protect or on_mask sp		
2 Zr/No22.udf Void ref 1 Zr/No22.udf		
\$ Sic2 hand		_
Save Save As	Uptate moder	

Click on bind. For this example the layer of ZnP and the roughness are coupled for the two models. It means that layer 2 and 3 of model M2 measured in transmission and layer 1 and 2 of model M1 measured in standard ellipsometry are coupled.

<mark>Bind Para</mark> M1 ↔ M2	meters					Ł
	1		M2 - Z	nP(Nc2	22).mdl	
		AOI	L1d	L2d	L2 % ZnPNc22.udf	
=	AOI					[
E	L1d					
5).1	L2d		V			[
5	L3d			v		
Z,	L3 % ZnPNc22.udf				✓	
ZnF						
5						
ž						
_						
Cancel						

Now the fit can be launched. The thickness and optical constants for ZnP as well as for the thickness of the roughness layer are forced to be identical for both models. The results are shown below:



This method is used for layers on transparent substrates (like thin layers on glass) when strong parameter correlation is present.

7.4.2 Multiple Sample Analysis

► Example

Any time two or more samples are known to have some physical property in common, the analysis of both (or all) samples may be performed simultaneously, with the common physical property coupled across the models for the different samples.

Simply perform the ellipsometric measurements of your the samples. Then build a model for each sample and set the right fitting parameters. The experimental variables are (Is, Ic) or (Ψ, Δ) for ellipsometric measurements.

Combined analysis of two samples

•Measurement strategy

- Simply perform the ellipsometric measurements of your two samples.

Modelling features

- Build each model and register them. Make sure to set the fitting parameters for each model. The experimental variables are (Is, Ic) or (ψ, Δ) .
- The two samples are composed of a GeSbTe alloys deposited onto a SiO₂ layer, the substrate is silicon. The GeSbTe films are nominally identical but of differing thickness.

75.000 * ▼ Exp. F	▼ 75.000 * ▼ Exp. File : GST_Ful_#02.15h 11mm 39s.spe					
E 314.7	FGST-1_tl.dsp	×				
3 1028.0	sio2.dsp	×				
2 🔲 3000000.0	C-si_isa.ref	×				
1 1028.0	sio2.dsp	×				
S	□void.dsp	×				
75.000 * Exp. F	75.000 * ¥ Exp. File : GST_Ful_#01.15h 18mm 21s.spe					
F 71.8	FGST-1_tl.dsp	×				
3	sio2.dsp	×				
2 🔲 3000000.0	C-si_isa.ref	×				
1008.0	sio2.dsp	×				
5	multiple and	×				

- A dispersion is used to represent the optical properties of the GeSbTe alloy. Note that you have to use the same name for both, «GST-1_tl.dsp» for this example. As the name is the same, the optical constants are bound by default.
- Then create a bound multimodel and drag and drop the two model files in the blank area in the left of the window. The first model is called M1 and the second M2.
- Launch your fit. The GeSbTe film thickness will vary on each sample and a single set of GeSbTe optical constants are forced to be identical for both films.



Fitting results of combined analysis of two samples

When to use it ?

- Anytime, two or more samples have some physical property in common.
- It is very useful to characterize ultra-thin films like thin metal or organic films on opaque or glass substrate. These examples are well known to provide strong correlation between thickness and optical constants.

Results Accuracy

- The parameter correlations should be reduced.
- A unique solution should be obtained.

8 Fitting model to data

8.1 The fitting process

A spectroscopic ellipsometry measurements yields experimental data that have to be fitted to a model to determine the unknown sample parameters.

The goodness of fit (GOF) is determined by the χ^2 value which should be as low as possible. If we measure for example spectra of Ψ and Δ then the χ^2 value is a comparison of n theoretically calculated pairs (Ψ_{th} , Δ_{th}) and n experimental determined pairs (Ψ_{exp} , Δ_{exp}):

$$\chi^{2} = \min \sum_{i=1}^{n} \left[\frac{\left(\Psi_{th} - \Psi_{exp}\right)_{i}^{2}}{\Gamma_{\Psi,i}} + \frac{\left(\Delta_{th} - \Delta_{exp}\right)_{i}^{2}}{\Gamma_{\Delta,i}} \right]$$

with Γ_i the standard deviation of each data point (generally set to 1).

One difficulty in such a process is to set optimal starting values for the unknown parameters in the model from which the theoretical pairs are calculated. The simulation option of the DP2 software is a fast way to check whether a model is approximately correct or not.

Another common problem of the fitting process is the correlation of different parameters of a model. In this case the solution might not be unique but might dependent on the initial set of starting parameters. The solution might then correspond to a local minimum and not as wanted to the absolute global minimum.

8.2 The iterative algorithms

In the bottom of the model window, the user can go to *Modeling description > edit > fitting params* and select the minimization algorithm: Levenberg-Marquardt, Simplex or Thick Film.

Modeling description					
Modeling description name : Def	Modeling description name : Default				
Modeling conditions Fitting pare	Modeling conditions Fitting params Simulation params Optical calculations params				
Name of fit curves :					
🔽 Clear fit results before fit	Minimization Observables				
	User selection :				
	C Mueller matrix only				
	Mueller matrix and user selection : Error				
Maximum Iterations	Is • 0.0100				
100	Ic 0.0100				
Fitting stop criteria	Nothing N/A				
X ² : 1E-6	Nothing N/A				
ΔX* : 1E-30	Recalculate from Ψ A				
	Elli Equivalent Calculations				
	Minimization algorithm				
	Marquardt				
	Marquardt Simplex				
	Thick Film_ 15				
	Ok Cancel Save as default				

8.2.1 The Levenberg-Marquardt algorithm

This algorithm is a combination of the Inverse Hessian method and the steepest descent method. **It uses the calculus of partial derivatives**. The choice of starting parameters is here critical because the algorithm can converge easily to a local minimum. Therefore the starting values for the fitting parameters have to be set very cautiously to avoid this. However, if the starting fitting parameters are already close to the ideal solution, the Levenberg-Marquardt algorithm converges very rapidly.

Another advantage is that the algorithm provides a correlation matrix and an estimate of the uncertainty for each fitted parameter.

8.2.2 The Simplex algorithm

The Simplex algorithm is a **geometric minimization method** that does not use the calculus of partial derivatives. The big advantage is that it is able to "crawl out" of local minima to find the global minimum. The simplex is a geometrical figure in a n-dimension space that has n+1 vertices: n is the number of unknown parameters (adjustable parameters). The n+1 points in the n-dimension space form a triangle in 2D, and tetrahedron in 3D. This minimization method evaluates a function at each vertex and attempts to move it into a new minimum. Each new minimum of the function becomes a vertex of a new polyhedron. This is repeated until the absolute minimum is encountered. The simplex formed from the points should thereby be non-degenerate, thus it should have a non-zero volume.

With this method **the choice of initial values is not critical**, which is important when little information on the studied material is available. However, this algorithm is slow and fitting many parameters can be really time-consuming. Moreover, this algorithm is not always sensible to the global minimum we search: two consecutive fits do not lead necessarily to the same solution.

To conclude, the simplex algorithm is good for opening a path to the final solution. However, it is not the best method to find the final solution. Another disadvantage is that the user cannot get access to the measurement precision or a parameter correlation matrix with the simplex method.

8.2.3 The thick film algorithm

This algorithm is based on a **fast Fourier Transformation** and can therefore be applied only to spectra of thick layers that show at least some interference pattern.

8.2.4 Conclusion

The Simplex method and the Levenberg-Marquardt algorithm are complementary. However, with a high number of unknowns (more than 6), some difficulties might appear:

- the initial parameters may be critical for the solution
- the uniqueness of the solution

Keep in mind that the Simplex method is used to approach the solution and has to be used when little information about the sample is provided. The Levenberg-Marquardt algorithm is faster but it can be used only when the starting values are already close to the solution.



Most of the time, for simple samples (thin monolayer or thin bilayer) of well defined materials (reference or simple dispersion) the Levenberg-Marquardt algorithm will yield the best solution.

8.3 The fitting functions

Nominal value: this option is used to perform the fit for a given value of the parameter

Multiguess: this method allows the user to perform a fitting process with different initial/ end values of the parameter. The retained fitted value is directly related to the best χ^2 .

Multistart: this method allows the user to perform a fitting process with different constant values of the parameter. χ^2 is calculated for these fixed values. The retained value is directly related to the best χ^2 .

Multiguess and Multistart have two options: absolute and relative:

The option *absolute* allows the user to choose the guess range and the increment value; the defined interval of values is independent from the nominal value.

The option *relative* allows the user to define the guess range as a variation in percentage around the nominal value and not as an interval of values.

Correlated: this option allows to correlate a parameter from the current layer to a corresponding parameter of second layer. This ratio can be constant or variable. The fitting process will be performed only on the current layer and the retained value is the one which will reach the best χ^2 .

8.4 How to perform a fit

8.4.1 1st STEP : Define and set the fitting parameters

Once the structure of the model is defined (type of layer, materials, succession of layers...), the user has to define the fitting parameters and set their values. The fitting parameters can be thickness, angle of incidence, proportion of an effective medium layer and parameters of an optical dispersion.

To define the fitting parameters, just check the appropriate fit box. Note: when the user checks the fit box of the material, a window appears: Dispersion Fit Parameters, Alloy Fit Parameters or PPC Fit Parameters in which the optical fitting parameters of the materials can be defined.

To set a value for the thickness, the angle of incidence or the proportion respectively click on thickness value, AOI value and proportion value. The Fitting Choice Properties window appears in which you can set the appropriate value. The only way to set the value of the different optical fitting parameters, open the material by a double click on the material name (or by a right click on the material name, then select Open material). Then, set a value for each parameter and save your material; the displayed window is the same you obtain when you create a material.

An alternative way of defining the fitting parameters and their values is to right click on the layer and select Fitting parameters; the Fitting Choice Properties window appears. In this window, several tabs are displayed: Thickness, Materials...in which you can define and/or set the different fitting parameters.

8.4.2 2nd STEP : Fitting description

Once the parameters are defined and their values are set, the next step of the fitting process is to determine the way the fit will be performed. The user has to determine:

- the fitting functions
- the iterative algorithm
- the spectral range on which the fit will be performed
- the observables on the graph
- the ellipsometric configuration used to acquire the spectrum

The fitting functions apply for angle of incidence, thickness and proportion. A left click on these values (or right click + *Fitting parameters* on these values) allows to access to the *Fitting Choice Properties* window where the fitting functions are displayed:

- multiguess absolute
- multiguess relative
- multistart absolute
- multistart relative

The iterative algorithms can be determined in *Modeling description* by clicking *edit*; these option are available in the *Fitting params* tab. The user has the choice between:

- the Levenberg-Marquardt algorithm
- the Simplex algorithm
- the Thick Film algorithm

and can also determine the maximum number of iterations. We suggest 100 iterations for simple cases of model, and 1000 iterations for complicated cases where the χ^2 does not cease to decrease.

The spectral range, ellipsometric configuration and observables can be determined in *Modeling description* by clicking *edit*; these option are available in the *Modeling conditions* tab. By default, the DP2 software uses the experimental spectral range and the experimental ellipsometric configuration. If you wish to modify the spectral range for the fitting process, we recommend you to take the same increment of wavelength or photon energies.

8.4.3 3rd STEP : To launch the fit

Once all the fitting requirements and fitting choices are done, the user can perform a simulation to rapidly see whether the starting parameters are reasonable. The other option is to launch the fit. Here are the simulation and fit buttons:

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Simulation	
Fit	

When a simulation is launched, the simulation appear together with the experimental results in the fit/simulation graph window:



When a fit is launched, the fit window is displayed:



Which shows the fit result (thickness in this case), the goodness of fit (χ^2) and the number of iteration needed to get the result.

In the fit window one can:

- Abort the fit: it stops immediately any fitting process and only the last χ^2 value displayed on screen is retained
- Jump to the next start value (available for Multiguess or Multistart fitting): this is used to pass to the next guess when the current guess seems to be not reasonable
- Stop the fit: this stops the fitting process, but the best χ^2 obtained so far is recalculated and retained
- Close: this is available when the fitting process is terminated and allows to return to the main window of DP2

8.5 Fitting procedures

Sometimes it is necessary to perform a fit in separated steps.

For example polymers can have a very complicated optical dispersion in the UV but are mostly transparent in the visible and the IR. It can be advantageous to determine in a first fitting step the thickness in the visible-IR part of the spectrum and then determine in a second step the dispersion in the UV with the fixed thickness from the first fitting step:

First we define the model and fitting conditions for step 1:

In the *model description* we set the fitting range from 0.7 to 2eV and name it step 1.

Modeling description						
Modeling description name : step 1						
Modeling conditions Fitting params Simulation params Optical calculations params						
Spectral Range C Experimental file full spectral range	Ellipsometric Configuration Experimental file configuration					
 Experimental file limited spectral range Start : 0.7000 eV • End : 2 eV User defined spectral range : Start : 1.5000 eV • End : 6.0000 eV Increment : 0.0500 eV 	 User defined configuration : Modulator angle : 0.0 Analyser angle : 45.0 Measurement angles Angle of incidence V.A.S.E. Theta angle V.A.S.E. 					
Increment : 0.0500 eV © Theta alige V.A.S.E. © Use spectrum number : 1 © Display specific observables after fit or simulation : Bottom axis observable : Photon Energy v Left axis observable : Is Right axis observable : Ic v						
Ok	Cancel Save as default					

Then we define the fitting parameters thickness and Cauchy coefficients for the polymer in this spectral region:

Image:					
S	C-si_isa.ref				
- Eiting skaipe					
step 1	Polymer 2 steps ✓ Step 1 ✓ Default	Simulation	Save		
Add 💼	Edit Add 🕋 Edit Add 🏦 Edit Add 🏦	Edit Fit	Save As		

We name it step 1 in the *Fitting choice* menu.

Now we define the model and fitting conditions for step 2:

In the *model description* we set the fitting range from 4 to 6.5eV and name it step 2.

Modeling description name : step 2 Modeling conditions Fitting params Simulation Spectral Range © Experimental file full spectral range © Experimental file limited spectral range Start : 4.0000 eV End : 6.5 eV © User defined spectral range : Start : 1.5000 eV End : 6.0000 eV Increment : 0.0500 eV	 params Optical calculations params Ellipsometric Configuration Experimental file configuration User defined configuration : Modulator angle : 0.0 Analyser angle : 45.0 				
Modeling conditions Fitting params Simulation Spectral Range Experimental file full spectral range Experimental file limited spectral range Start : 4.0000 eV End : 6.5 eV User defined spectral range : Start : 1.5000 eV End : 6.0000 eV Increment : 0.0500 eV	 params Optical calculations params Ellipsometric Configuration Experimental file configuration User defined configuration : Modulator angle : 0.0 Analyser angle : 45.0 				
	Measurement angles Angle of incidence V.A.S.E. Theta angle V.A.S.E. Use spectrum number : 1 				
Display specific observables after fit or simulation : Bottom axis observable : Photon Energy Left axis observable : Is Right axis observable : Ic Ok Cancel Sources default					

Then we define the fitting parameters polymer in the given spectral region: now it is only point-by-point calculation of the dispersion – the thickness is fixed to the value found in the first step:

1 1000.00	₩ E polymer.ppc				
S	C-si_isa.ref				
Eiting choice					
step 2 polymer 2 steps	▼ step 1 ▼ Default ▼ Simulation) Save			
Add a Edit	dit Add 👘 Edit Fit	Save As			

Now we can set-up the fitting procedure that we call for example "polymer 2 steps" by assigning the fitting choice and corresponding model description (MD):

User defined Fitting Proce	edure 🗙
Procedure name : polymer	2 steps
Drag Fitting choices from here to the fitting steps list Default fitting choice 1 fitting choice 2 step 1 step 2	Fitting steps
	Check X ^z < 100 Modeling description : step 2 ✓
	✓ OK X Cancel



Now the complete fitting procedure can be launched.

8.6 Quality of Data - Goodness of fit (GOF)

When a model is used to describe experimental data the software always provides some sort of result. The user then has to decide whether the result is good or not. There are some techniques that help the user to evaluate the quality of the results. The first steps for an evaluation are:

- check the goodness of fit (GOF) value represented by χ^2
- compare visually the measured optical data and the final best-fit calculated data
- check whether results are physically reasonable boundaries % output check criteria

These issues are discussed in detail below.

8.6.1 χ^2 value

The χ^2 value is a direct measure of discrepancies between measured optical data and the final best-fit calculated data. High χ^2 values are usually indicative of a poor fit to the experimental data.

Note that what is a low and what is a high χ^2 value depends on the layer model and the fitting conditions. A high χ^2 value in one case can be a low value in another. Generally speaking, the more simple the model is the lower the χ^2 value should be.

For example, the χ^2 values for a spectroscopic ellipsometer measurement in the range 1.5eV-5eV should give the following values:

Samples	χ2
SiO ₂ 20Å on Si	$\chi^2 < 0.20$
SiO ₂ 2000Å on Si	$\chi^2 < 2$
PZT 5500Å on Pt	$\chi^2 < 10$
TiO ₂ 5500Å on Si	$\chi^2 < 10$
AlN 25000Å on Si	$\chi^2 < 15$

The following points have to be kept in mind when evaluating χ^2 values:

- the thicker the layer the more structure is in the spectra the higher χ^2 will be
- optical properties that enhance structural features in the spectra will lead to higher χ^2 values
- by reducing the fitting range, the χ^2 value generally decreases because the spectrum shape appears simpler on small wavelength intervals. However, reducing the fitting range reduces the precision because information is suppressed, so that generally the confidence limits increase.
- reducing the number of measurement points will lead to a lower χ^2 value because the spectra will have less features

8.6.2 Physically reasonable parameters

The determined parameters should of course make physically sense. Signs of failure are for example:

- negative thickness values
- negative dispersion values (peak energies, band gaps, damping,...)
- transparent materials: an index of refraction that increases with increasing wavelength

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HORIBA

Unphysical optical constants are a sign for an inappropriate optical model. The optical constants may become unphysical in order to compensate the deficiencies of the model.

One way to avoid this problem is to fix the unphysical optical constants to reasonable values and to fit only the other parameters. If this does not work a more sophisticated model has to be chosen.

8.6.3 Confidence limits

The Marquardt Levenberg fitting algorithm provides confidence limits for each parameter.

If a confidence limit of the same order of magnitude as the fitting parameter itself there is very likely a problem with the model. Even if the χ^2 value is good, the optical constants make physically sense the fit can still be bad if the confidence limits are high. The reasons can be:

- the experimental data are not sensitive to the fitted parameter of interest
- the model is incorrect and the final χ^2 value is large and thus bad. In this case, all the fitted parameters are affected by large confidence limits: the larger the χ^2 value the larger the confidence limits.
- parameters are strongly correlated. In this case their confidence limits will be very large.

8.7 Uniqueness of fit

The final thing the user has to do in the fitting process is to determine whether the best-fit result is unique or not.

Non-unique results can occur if:

- there are parameters that only weakly affect the fit: the best thing to do is to fix these parameters to reasonable values and to not fit them.
- there is strong correlation between parameters: reduce the parameter set in order to eliminate the correlations.

8.7.1 Parameter correlation

Parameter correlation occurs when:

- there are too many unknown parameters in the model
- the effects of one parameter compensates exactly the effects of another parameter

Correlations between parameters can be identified by the following symptoms:

- the Marquardt Levenberg fitting algorithm provides a correlation matrix. If the magnitude of any of the non-diagonal elements of the correlation matrix is close to unity then strong parameter correlation exists.
- the fitting process needs a lot of iterations and the χ^2 value is only slightly reduced during each iterations: the iterative algorithms are not capable of determining which of the correlated parameters needs to be adjusted.
- the fitting parameters take drastically different values at each iteration and the χ^2 value varies only slightly. In this case at least two parameters are correlated and compensate each other and/or the χ^2 value is not sensitive to the given parameter(s).
- static repeatability measurements: if the repeatability value of parameters is bad although χ^2 is reasonable then parameters are correlated
- profile scans across the sample: physically reasonable parameters should vary steadily across the sample. If there are "jumps" then there is most probably parameter correlation

Parameter correlation can be of course also avoided by combining different optical data types like data from spectroscopic ellipsometry (at different angles of incidence), reflectometry and transmission measurements. The DP2 software can handle all these data types and all data can be fitted simultaneously.



9 Routine Measurements

There are methods that allow the user to automate the measurements and the out-putting of the results. These methods will be described only shortly here because there is a detailed description in the DP2 user manual.

9.1 Recipes

With the help of recipes you can automate your measurements. Click on the recipe button $\frac{1}{Rec.}$ and you see that there are two types of recipes:



The main application of the *immediate recipe* is to measure repeatedly at the same spot with the same acquisition routine & model. However by defining groups you can also apply different acquisition routines & models repeatedly at the same spot:

🙆 Imm	🕅 Immediate Analysis Recipe - Noname1.rci					
Definition Sample Judgement Criteria Execution Criteria						
Create	group Delete group					
Group	Repetition count	Acq. Routine	Model	Sampling Interval		
1	1 🔺]		0.0 × s		
2	1 🔺 🔻			0.0 • s		

The **mapping recipe** allows the application of acquisition routines & models to a series of different points on the sample (grid):



The grid is set-up in the grid-menu that can be opened with the state button.

9.2 Reports

With DP2 reports can be created automatically after the measurements. You can set-up a template for a report by clicking the $\frac{1}{100}$ button.

Create new report template		
User Library		
Acquisition Results Report Template		
Data Manipulation Report Template		
(mdt) Model Report Template		
Recipe Report Template		
X Cancel		

In the report templates you can activate features like headers, pictures of the layer model, acquisition data, results, etc.

As an example the options for the Acquisition Results Report Template are shown below:





10 QuickStart

10.1 SiO₂/Si: HJY certified sample

In this chapter the basic steps are shown to perform the complete analysis of a very simple sample, the system test sample: native SiO_2/Si substrate.

The idea is to begin with the analysis of a simple sample to get the automatisms that will be needed for analysis of more difficult samples. As already discussed, the main steps of any analysis are:

- system start
- sample positioning
- data acquisition
- modelling
- data fitting

System start

- Switch on the controller unit and the computer where the DeltaPsi2 software is installed.
- Turn on the *light source* (usually a Xe arc lamp) and wait 15min to let it stabilize.
- Open the *DeltaPsi2* software and click on the "software options" tab:

The following window will appear:

Software Options					
User Library		User Library file fold	ers		
coor cibrary		Current User Library	profile : 🔎	Sort by name 🛛 🎧 Sort by creation order	
Application Library		Default			
		👷 Create a new j	profile	Delete the current profile	
File name formats		Folders :			
		Acquisition Routines	C:\DeltaPsi2\U:	ser Library\Default\Acquisition Routines\	
File explorer		Data Manipulation	C:\DeltaPsi2\U:	ser Library\Default\Data Manipulation\	
		Substrates	C:\DeltaPsi2\U:	ser Library/Default/Substrates/	
Tool bar		Materials	C:\DeltaPsi2\U:	ser Library\Default\Materials\	
		Samples	C:\DeltaPsi2\U:	ser Library\Default\Samples\	
Progress screens		Models	C:\DeltaPsi2\U:	ser Library/Default/Models/	
riogress sereens		Multimodels	C:\DeltaPsi2\U:	ser Library\Default\Multimodels\	
		Modeling Scenarios	C:\DeltaPsi2\U:	ser Library\Default\Modeling Scenario\	
calculated		Grids	C:\DeltaPsi2\U:	ser Library/Default/Grids/	
observables	-	Recipes	C:\DeltaPsi2\U:	ser Library\Default\Recipes\	
Acquisition		Report Templates	C:\DeltaPsi2\U:	ser Library/Default/Report Templates/	
parameters		Acquisition Data	C:\DeltaPsi2\U:	ser Library/Default/Acquisition Data/	
Observables		Model Results	C:\DeltaPsi2\U:	ser Library/Default/Model Results/	
precision		Multimodel Results	C:\DeltaPsi2\U:	ser Library\Default\Multimodel Results\	
Modeling output		Modeling Scenario Results	C:\DeltaPsi2\U:	ser Library\Default\Modeling Scenario Results\	
modering output		Recipe Results	C:\DeltaPsi2\U:	ser Library\Default\Recipe Results\	
Close		Model Reports	C:\DeltaPsi2\U:	ser Library/Default/Model Reports/	

- Choose "user library" and choose an existing user library profile in the drop-down menu or "Create a new profile...".
- Close the "software options" window and click the "manual measurement" tab to open the "Views" screen:

B 📶 💏 at. Åcq. Mod. Grd. Rec. Tem. 🖹 🔂 (F) 0 E B -💌 🞾 - Ellipse neter View - 8 × Vser Library + 5 困 8 Acquisition Routines Signal Data Manipulation 250 Substrates 110.808 mV 200 • S0 Materials Samples 150 49.295 mV C S1 0.42846 ls Models 100 C S2 9.864 mV 0.10308 Multimodels lc: 50 Modeling Scenarios 0.42846 C Bw Grids 0 Recipes 5 10 15 20 25 30 35 40 45 C R2w S0 Adjustment 0 0.10308 Report Templates Ellipsometric data **Results** Param 🗄 🕨 Acquisition Data Δ 76.472 * 13.074 * Ψ Model Results 200 ms Integration Time eters Multimodel Results 0 V **High Voltage** Modeling Scenario Res Semi-infinite model Recipe Results • XYZ stage cm⁻ User unit: Reports 4.291 εį 4.214 εr Model Reports Micro-Spot II: M=0° A=+45° Multimodel Reports • Configuration Modeling Scenario Rep 2.270 0.928 • n k Background off Background Recipe Reports Application Library Acauisition Routines 🗄 🕨 Data Manipulation Motors Substrates 🔹 🗉 Stop ▶ Start 4000.00 cm⁻¹ 0.10 cm⁻¹ • • Monochromator By Materials Samples 南 Close 🔞 New spectroscopic 🛛 🔞 New multiwavelength Pause Models • AOI = 70* 8d Ellipsomete Views Current profile : Default

Views screen:

Sample positioning

- Place your sample on the stage and switch on the autocollimator illumination.
- Look through the eyepiece of the autocollimator and align the crosshairs so that the centers overlap by adjusting the two knobs underneath the stage.
- In the views screen set the monochromator to 2.75eV (or 450nm if you prefer to work with wavelength).
- Adjust now the height of the stage by turning the corresponding knob underneath the stage. Try to find the position where S0 is at its maximum
- In the views screen click "S0 adjustment" or manually set the high voltage so that S0 is between 60 and 80mV.

Data acquisition

For data acquisition click on the Acq. button:



The following window will appear:



Create new acquisition routine				
User Library	O Application Library			
Ellipsometer Refle	ctometer LCE			
	Ellipsometer Calibration			
	Spectroscopic Mono			
	Spectroscopic MWL			
	Kinetic Mono			
	R&T Mono			
(poo)	R&T MWL			
	X Cancel			

Choose Spectroscopic Mono:

Acquisition simulation Configuration choice No merge I: M=0° A=+45° Start: 1.500 eV End: 6.000 eV Increment: 0.100 eV	Spectroscopic m	ono acquisition setup Gra	ph setup Graph	
Configuration choice No merge I: M=0° A=+45° I: M=0° A=+45° Calibration index: No data Micro-Spot: Background : Background : Background : Protection UV : Off Incidence angle : 70.00 * Integration time : 1 x 200 ms SO Max :	Acquisition si	mulation	Spectrum range	9
No merge ▼ II: M=0° A=+45° ▼ End: 6.000 eV End: 0.100 eV Increment: 0.100 eV Mode: Standard Micro-Spot: ▼ Background : Background on Protection UV: Off High Voltage: Automatic ▼ 250 V SO Min : 60 mV SO Max : 80 mV Incidence angle: 70.00 * Integration time: 1 x 200 ms Accumulation: •	Configuration ch	oice	User unit :	eV ▼
II: M=0° A=+45° End: 6.000 eV Increment: 0.100 eV Mode: Standard Micro-Spot: Background : Background on Protection UV: Off High Voltage: Automatic V 250 V SO Min : 60 mV SO Max : 80 mV	No merge	•	Start :	1.500 eV
Lalibration index: No data Micro-Spot : Image: Standard Background :: Background on Image: Standard Protection UV : Off High Voltage : Automatic Image: Standard So Min : So Max : 80 mV So Max : 1 X 200 ms Accumulation :	II: M=0° A=+	45° 🔹	End :	6.000 eV
Calibration index: No data Micro-Spot : Image: Standard Background :: Background on Image: Standard Protection UV : Off High Voltage : Automatic Image: Standard Kode : So Min : So Max			Increment :	0.100 eV
Micro-Spot : Background : Background on Protection UV : Off High Voltage : Automatic V250 V SO Min : 60 mV SO Max : 80 mV Accumulation :	alibration index	No data 💌	Mode : Standa	ırd 💌
High Voltage : Automatic V 250 V SO Min : 60 mV SO Max : 80 mV	Micro-Snot			
Protection UV : Off High Voltage : Automatic 250 V SO Min : 60 mV SO Max : 80 mV Accumulation :	Background :	Background on 🔻		
High Voltage : Automatic v 250 V SO Min : 60 mV SO Max : 80 mV Accumulation :	Protection UV :	Off		
SO Min : 60 mV SO Max : 80 mV Accumulation :	High Voltage :	Automatic 💌 250 V	Incidence angle	e: 70.00 •
SO Max : 80 mV Accumulation :	SO Min :	60 mV	Integration time	: 1 X 200 ms
	SO Max :	80 mV	Accumulation :	1

For a thin SiO₂/Si sample the acquisition parameters should be the following:

- AOI: 70° (close to the Brewster angle of a pure Si substrate)
- spectral range: in order to test the system the complete spectral range should be used
- step size: 0.1eV: this allows the resolution of all spectral features of a thin SiO₂/Si sample
- configuration II: you want to determine very accurately and do not expect to be close to 45°.
- integration time: 200ms
- Since the sample is already in the right position you can *run* the acquisition. Enter the sample name any comments if needed. Click "o.k.".
- After the measurement is run the acquired spectra are stored automatically in the folder *Results**Acquisition data*. The spectra should look like the following:



Different parameters can be displayed as functions of the wavelength or photon energy:

- Ψ and Δ : the ellipsometric angles known to a wide community and therefore used for publications, reports, etc.
- I_s and I_c : these are the measured parameters of a phase modulated ellipsometer: they are usually used for the fitting process
- *n* and *k* or ε_r and ε_i : optical constants: useful to see immediately the properties of a substrate or a "pseudo"-substrate

For example the graph from above looks in the $I_s(E)$ and $I_c(E)$ representation like this:



Modelling

To create a new model click on the *Mod.* tab:



The following window appears:

Create new mode	l	×
User Library	O Application Library	
mal	Spectroscopic Model	
(bmm)	Bound Multimodel	
	🗙 Cancel	

You can choose the folder where you want to put the model file (User library or Application library). Click on *Spectroscopic Model* to access the model window:

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*** *** * * ** **	And Brd Rec. 9am 1 🔂 💕 🔑 🔟	🞽 🛛 👌
User Library -	Model - Noname2.mdl	-18 ×
User Library	Recall A % Q 11 B % A % Q 11 A % A % Q 11 A %	Servicedu file Doder Stowieson 3 vo Heo
Materials Manples Models Models Modeling Scenarios Grids Recipes Report Templates	Pring modelare Modeling decogrition Dodgad decogrition Partial Image: Strape Stop Image: Partial Strape Stop Image: Partial Strape Stop Add Edit Add Edit Strape Stop	ndefon FR Save As
Noname2 mdl		
Current profile : Default		
1 Sample Str	ucture area	Modeling screen layout
2 Results Gr	aphics display area	
3 Fitting Res	ults area	

Within **Results** Acquisition data in the *TreeView*, select the acquired file *.spe and drag and drop it to the corresponding area (*Exp.File*).

Go to the *Application Library**Materials* folder and drag and drop the file *C-si_isa.ref* into the pink bar marked *S* for substrate. Then drag and drop the file *Sio2_isa.ref* from *Application Library**Materials* above the substrate:

¹ 1000.0	Sio2_isa.ref	××	×
S	C-si_isa.ref	×	

The next step is to change the default value for the thickness (1000Å) to the approximate value of your sample (in this case 20Å): click on the displayed thickness value to open the window *Fitting choice Properties for layer #1* and adjust the thickness. Keep thereby the option *nominative value*:

¹ F 20.0	Sio2_isa.ref	××
S	C-si_isa.ref	×

Now a simulation can be launched by clicking on the corresponding button to check whether the thickness start value is reasonable.

Data fitting

Now that the model is set-up the experimental data can be fitted. For that click on *Edit* in *Modeling description* on the bottom of the model window:

 Modeling description 		
Default		
Add	Ť	Edit

Select *fitting params* and verify that I_s and I_c are the minimization observables. Choose *Marquardt* as *Minimization algorithm*.

Modeling description				
Modeling description name : Def	ault			
Modeling conditions Fitting para	ms Simulation params Optical calculations params			
Name of fit curves :				
✓ Clear fit results before fit	Minimization Observables © User selection : © Mueller matrix only			
	\bigcirc Mueller matrix and user selection :			
Maximum Iterations 100 Fitting stop criteria X² : 1E-6 ΔX² : 1E-30	Error Is			
	✓ Recalculate from Ψ, Δ ► Elli Equivalent Calculations Minimization algorithm Marquardt ↓ 1E-16 tol : 1E-15			
	Dk Cancel Save as default			

Click "Ok", save the model and launch the fit with the appropriate button.

Results

The page displayed after the fit will be:





where you can see that:

thickness = 18.434Å ± 0.274 (confidence limits)

 $\chi^2 = 0.13$ (and thus < 0.20 as required)

You can try now to change the spectral range for this specific case. Click on *Edit* in *Modeling description* on the bottom of the model window, and check the box: *User defined spectral range* in *Modeling conditions*. Try different settings. The result should stay more or less the same as shown in the table below:

	Thickness (Å)	2
From: 1.5eV		
To: 5eV	18.434	0.13
Increment: 0.1eV		
From: 1.5eV		
To: 4.5eV	18.638	0.14
Increment: 0.1eV		
From: 1.5eV		
To: 4.0eV	18.341	0.11
Increment: 0.1eV		
From: 1.5eV		
To: 3.5eV	17.840	0.09
Increment: 0.1eV		
From: 1.5eV		
To: 3eV	18.350	0.09
Increment: 0.1eV		

This is due to the fact that the result for the thickness is nearly independent of the spectral range in the case of a transparent material. This is not necessarily the case if the deposited layer shows a complicated dispersion with absorption bands (examples in chapter 10).



11 Application Examples

11.1 Overview

Chapter	Task	Sample	Topic
11.2	Thickness of thin film	SiO ₂ /Si	UVISEL configurations
11.3	Thickness and dispersion	Si3N4/Si	Dispersion files
11.4.1	Transparent substrate	glass	Back side reflection
11.4.2	Film on transparent substrate	a-Si/glass	Back side reflection Amorphous dispersion
11.5	Composed materials	OPSL	EMA
11.6	Multilayers	TiO ₂ /SiO ₂ /Si	Layer repetition
11.7	Complicated dispersions	Organic layer	Nk-fit
11.8	Inhomogeneous layer	ITO/glass	Graded index layer
11.9	Correlated parameters	TiO2/glass	Bound multimodel
11.10	Non-cubic crystals	Al ₂ O ₃	Uniaxial anisotropy
11.11	Strained materials	polymers	Biaxial anisotropy

The following table summarizes the examples discussed in this chapter:

11.2 Thickness of a single film

11.2.1 Silicon Oxide on Silicon

11.2.1.1 Key points

- Influence of the measurement configurations
- How to use of the merge function

11.2.1.2 Typical settings of acquisition parameters:

Sample	Thermal oxide layer
Spectral range	1.5 – 5eV
Increment	0.050eV


AOI	70°
Spot size	1mm x 3mm
Integration time	200ms
configuration	II & III in standard merge

11.2.1.3 Experiment

The standard merge (STM) acquisition function generates automatically 3 files:

one for the measurement in configuration II: xxx-CII.spe, one for the measurement in configuration III: xxx-CIII.spe and the "merged" file: xxx-STM.spe. The two next figures show Ψ and Δ spectra for configuration II (blue) and configuration III (red). Remember that from the definition of the ellipsometric angles (ρ =tan $\Psi e^{i\Delta}$) the domains are [0, 90[for Ψ and [0, 360[for Δ .



There are obviously mirror effects on Ψ around 45° and on Δ around 90° and 270°. Explanation:

 Ψ , Δ are calculated from Is and Ic in the last step of the acquisition flowchart, which starts by digital processing in the acquisition board (details in appendix C):

$$\begin{pmatrix} S_0 \\ S_1 \\ S_2 \end{pmatrix} \longrightarrow \begin{pmatrix} S_{\omega} \\ S_{2\omega} \end{pmatrix} \longrightarrow \begin{pmatrix} Is \\ Ic \end{pmatrix} \longrightarrow \begin{pmatrix} \Psi \\ \Delta \end{pmatrix}$$

The deduction of Ψ and Δ from Is and Ic is according to the following relations: Configuration II, for M = 0 and A = 45°: Is = sin2 Ψ sin Δ and Ic = sin2 Ψ cos Δ Configuration II, for M = 45° and A = 45°: Is = sin2 Ψ sin Δ and Ic = cos2 Ψ

The next graph shows Is and Icfor the two configurations (note that Is does not depend on the configuration):



A closer look at the trigonometric functions shows that:

	Ψ	Δ
Config. II	sin2 Ψ defines Ψ in the interval [0, 45] only. Values above are flipped down around the line $\Psi = 45^{\circ}$	knowing $\sin\Delta$ and $\cos\Delta$ allows determination of Δ over the complete domain
Config. III	$\cos 2\Psi$ allows the determination of Ψ over the complete domain	$\sin\Delta$ defines Δ in the interval [0, 90] only. Values above are flipped down around the line $\Delta = 90^{\circ}$

The resulting standard merge file contains therefore Ψ from configuration III and Δ from configuration II.

HORIBA

Explore the future

File	Thickness	χ^2	Fitting spec
Config. II	859	2	(Is, Ic)
Config. II	855	400	(Ψ, Δ)
Config. III	859	1000	(Is, Ic)
Config. III	860	2000	(Ψ, Δ)
STM	859	2	(Is, Ic)
STM	859	2	(Ψ, Δ)

The next table summarizes fitting results for the two configurations:

The high χ^2 or a wrong thickness result from:

configuration II file has the wrong and cannot be fitted using (Ψ, Δ) configuration III file has wrong and cannot be fitted using (Ψ, Δ) configuration III file has wrong Ic and cannot be fitted using (Is, Ic)

Conclusion:

we recommend to restrict generally to configuration II and use (Is, Ic) as minimization observables. The only interest of STM is to extract exact values for Ψ and Δ but it does not improve the fitting process.

11.3 Thickness and dispersion of a single film

11.3.1 Silicon Nitride on Si

11.3.1.1 Key points

- Choice of dispersion
- Illustration of local minimum
- Fitting functions and algorithms

This example illustrates the determination of optical properties of the film by two different dispersion models.

It also emphasizes the behaviour of fitting algorithms and wrong mathematical solutions (local minima).

11.3.1.2 Typical settings of acquisition parameters:

Sample	Si3N4 layer
Spectral range	1.5 - 5 eV
Increment	0.050eV
AOI	70°

Spot size	1mm x 3mm
Integration time	200ms
Configuration	II & III in standard merge

11.3.1.3 Experiment

The standard .spe file obtained in configuration II should look like that:



Note that Ψ is flipped for some parts of the curve. This is not important for modelling which is done on Is and Ic and not on Ψ and Δ .

11.3.1.4 Interpretation

Optical properties of a material can be described either with the dielectric function or with the complex index of refraction. The relation between the two complex values is:

 $re(\varepsilon) + i im(\varepsilon) = (n + ik)^2$

The first dispersion under study is called *classical*. Classical, because it is not based on quantum mechanics.



Dispersion formula :	
Classical	Ŧ
Adachi-3 Forouhi	~
Adachi-Forouhi	
Adachi-New Forouhi	
Afromovitz	
Amorphous	
Cauchy Absorbent	
Cauchy Transparent	
Classical	
Conrady	
Double Amorphous	
Double New Amorphous	
Excitonic	
Fixed Index	
Haltman	
-Kato-Adachi	
Kato-Adachi IR	
New Amorphous	
Quatre Amorphous	
Quatre Oscillators	
Schott (Briot)	
Sellmeier Absorbent	
-Sellmeier Transparent	
languy	
languy Extended	
Tauc Lorentz	
- Tauc Lorentzz	
Tiele Anne Lorentza	
Triple Amorphous	~
Triple New Amorphous	

The complete expression is:

$$\mathcal{E} = \mathcal{E}_r + i \mathcal{E}_i = \mathcal{E}_{\infty} + \frac{\left(\mathcal{E}_s - \mathcal{E}_{\infty}\right)\omega_t^2}{\omega_t^2 - \omega^2 + i\Gamma_0\omega} + \frac{\omega_p^2}{-\omega^2 + i\Gamma_D\omega} + \sum_j \frac{f_j \omega_j^2}{\omega_j^2 - \omega^2 + i\Gamma_j\omega}$$

For $\mathrm{Si}_3\mathrm{N}_4$ we use only one Lorentzian oscillator, so that the dispersion can be written as:

$$\mathcal{E} = \mathcal{E}_{\infty} + \frac{\left(\mathcal{E}_{s} - \mathcal{E}_{\infty}\right) \mathcal{O}_{t}^{2}}{\mathcal{O}_{t}^{2} - \mathcal{O}^{2} + i \Gamma_{0} \mathcal{O}}$$

The graphical representation is:



Note that:

- if $\Gamma_0 = 0$: im(ε) = 0: the material is transparent.
- if $\Gamma_0 \ge 0$: im(ε) ≥ 0 : the material shows absorption.
- im(ε) is small except around ω_t . The FWHM of this peak is equal to Γ_0 .
- for $\omega \to 0$, re(ε) $\to \varepsilon_s$
- for $\omega \to \infty$, re(ϵ) $\to \epsilon_{\infty}$
- ϵ_∞ is usually equal to 1 unless otherwise specified.

If we enter the parameters $\epsilon_s{=}4$ and $\omega_t{=}10$ the following dispersion is generated:



We recognize $\varepsilon_s = 4$ in low energies (n²= ε_s for $\omega=0$).

We use this material SiN osc.dsp into the mdl environment:

¹ F 2000.0	F SiN osc.dsp	×	×
S	Csi_uv.ref	×	

Simulations with varying thickness show that 2000 Å will be a good guess :



Fitting thickness and two dispersion parameters leads to following results :

thickness : 2506 Å ± 26 $\varepsilon_s = 3.24 \pm 0.05$ $\omega_t = 9.28 \pm 0.06$ $\chi^2 = 11$

and an excellent graphic agreement :



We repeat fitting with different initial values for thickness in the range 500 to 5000Å:

Initial thickness	χ^2	thickness result - comment
500	4100	169Å - Bad
1000	3300	550Å - Bad
1500	2400	639Å - Bad
2000	11	2500Å - Good
2500	3300	2500Å - Bad
3000	11	2500Å - Good

3500	5000	1327Å - Bad
4000	4300	2000Å - Bad
4500	5400	2000Å - Bad
5000	5200	2200Å - Bad

Obviously, only two thickness start values, 2000 and 3000Å give good solutions (lowest χ^2). For the other start values the Marquardt minimization algorithm falls into a local minimum where χ^2 is much higher.

Note that starting from the right thickness (2500Å) leads to a bad solution. This is due to the fact that we are fitting in addition to thickness also 2 dispersion parameters. This shows that permanent care has to be taken with multi-dimensional minimization routines.

The next figure shows the fitting function and local/global minima:



The red trajectory starting from the right thickness does not lead to the best (global) minimum. The minimization algorithm falls into a local minimum with correct thickness but a much too high χ value caused by the incorrect final dispersion parameters.

Two green trajectories starting from wrong thicknesses (2000 and 3000Å) follow the high slope of the χ surface, thus avoiding the local minima met above, and fall into the correct solution, which corresponds to the global minimum.

DP2 can handle these potential difficulties in two ways:

1. multiguess

several fits are performed with different start values:

Fitting Cho	ice Propertie	es for la	yer #1			X
Thickness	Materials S	oecial fu	nctions L.B.	R. Checks		
✓ Fit						
O Nomina	al value :			2000.0	Å	
MultiguMultigu	ess absolute ess relative :		Start : End :	0.1	A A	
⊖ Multista ⊖ Multista	art absolute : art relative :		Increment :	100.0	A	
C Correla	ited to : Lay	er numb	er :	1		
	Rati	o factor	:			
	• F	ixed :		1.000		
	0 \	'ariable	Start :	1.000		
			End :	1.000		
			Increment :	0.000		
		Ok	Cance	el		

In the present case we apply *multiguess absolute*, starting from different thicknesses between 0 and 5000Å (increment100Å). We find the correct solution:

```
thickness: 2506\text{\AA} \pm 26\text{\AA}
\varepsilon_s = 3.24 \pm 0.05
\omega_t = 9.28\text{eV} \pm 0.06\text{eV}
\chi^2 = 11
```

The multiguess method was applied here only for one variable (thickness), but of course it can be applied to every adjustable parameter.

The method is very reliable but can be very time consuming. For example if we choose in the case of m adjustable parameters 10 guesses for each parameter the number of fits will be 10^{n} !

2. Simplex minimization method

At the beginning the user should again set start values for the adjustable parameters. For example if m parameters are adjustable the Simplex method will generate randomly m additional sets of these m variables. The result is a polyhedron in a m-dimensional space. The Simplex method will then twist and scrunch this polyhedron around a neighborhood of "small" χ^2 . This method is less sensitive to local minima than the Marquardt-Levenberg algorithm. Its drawbacks are speed and the fact that two successive attempts will never give the same solution. Therefore one should repeat the fit several times to make sure the method has really reached the best solution. Although DP2 allows it, we suggest not to combine this method with the multiguess option.

In the following table the results of 10 successive attempts (same initial values) are summarized for the present example.

Simplex	χ^2	Solution found, comment
1	8000	2500 Å – Bad



2	4200	1500 Å – Bad
3	4700	1100 Å – Bad
4	4200	1300 Å – Bad
5	3300	2500 Å – Bad
6	4000	2600 Å – Bad
7	64	2200 Å – Bad
8	4000	1400 Å – Bad
9	10	2500 Å – Good
10	2600	1600 Å – Bad

Unfortunately only attempt number 9 leads to the correct solution. This means that one has to repeat the fit many times to be sure about the correct solution.

Now we return to our initial problem, the properties of the nitride layer. The best solution for the moment gives $\chi^2 \approx 11$. If we set additional Γ_0 as unknown parameter we get:

thickness: 2564Å ± 31Å $\varepsilon_s = 3.13 \pm 0.05$ $\omega_t = 9.25 \text{eV} \pm 0.0 \text{eV}$ $\Gamma_0 = 0.06 \text{eV} \pm 0.02 \text{eV}$ $\chi^2 \approx 8$

So χ^2 has decreased from 11 to 8 (thus 30 %) while the number of fitted parameters has increased from 3 to 4 (thus 33 %). This means that the introduction of one more parameter does not improve the fit significantly. The model using a Lorentz dispersion formula has reached its limit in this case. An alternative is the *new amorphous* formula, which corresponds to a rewriting of the original Forouhi-Bloomer formula:

$$n = n_{\infty} + \frac{B_{i} (\omega - \omega_{i}) + C_{i}}{(\omega - \omega_{i})^{2} + \Gamma_{i}^{2}}$$

and

$$\mathbf{k} (>\mathbf{g}) = \frac{f_i(\omega - \omega_g)^2}{(\omega - \omega_i)^2 + \Gamma_i^2}; \mathbf{k} (\omega \mathbf{g}) = \mathbf{0}$$

where $B_i = \frac{f_i}{\Gamma_i} [\Gamma_i^2 - (\omega_i - \omega_g)^2]$ and $C_i = 2f_i \Gamma_i(\omega_i - \omega_g)$

The following graph shows an example of this dispersion:





Note that for $\omega \le \omega_g$: k = 0 that means the material is transparent below the band gap. We try the following start parameters for Si₃N₄:

> $n_{\infty} = 1.75$ $\omega_{g} = 3.42eV$ $f_{j} = 0.16$ $\omega_{j} = 7.31eV$ $\Gamma_{j} = 1.7eV$



We use now the dispersion SiN new in the model instead of SiN osc:

¹ F 2000.0	F SiN new.dsp	×	×
S	Csi_uv.ref	×	

and get: $\chi^2 = 4.95$

thickness = $2530\text{\AA} \pm 19\text{\AA}$

Thus χ^2 has decreased from 11 to 4 while the number of fitted parameters has increased from 3 to 6. This means that the New Amorphous formula should be used instead of the classical formula.

An additional reduction of χ^2 can often be achieved by adding a *top layer* in the layer model. This top layer is defined as an *EMA* (effective medium approximation) of 50% of the underlying material and 50% of void. A good start value for the thickness is in most cases 30Å. The volume fraction is usually kept fixed but thickness becomes an additional fitting parameter.

The new layer model would then look like:

2	F	26.5 🗌 %	FSiN formation.dsp	50.00 % ×	Void.ref	50.00 % × ×	×
1	F	2512.1	F SiN formation.dsp			×	×
S			Csi_uv.ref			×	

In the present case this overlayer improves the fit to = 2.8. The total thickness (layer + overlayer) is very close to the one layer model result.



The results of the dispersion parameters are then:



 $f_j = 0.081$ $\omega_j = 7.074 eV$ $\Gamma_i = 0.887 eV$

11.4 Transparent substrates

11.4.1 Pure glass without film

11.4.1.1 Key points

- Back side reflection
- Influence of substrate thickness
- Reflection & Transmission experiments

11.4.1.2 Typical settings of acquisition parameters:

Sample	Glass substrate
Spectral range	1.5 – 6.5eV (825 – 190nm)
Increment	0.050eV
AOI	55°- 60°
Spot size	1mm x 3mm
Integration time	1000ms
Configuration	II

For transparent materials that are polished on both sides we get effects due to the backside (BS) refection of the light beams:



In the case of a 1mm thick glass sample that is polished on both sides the detection head will collect several incoherent from the BS reflected beams. The exact number of collected incoherent beams is not critical as long as they are at least 2.



When the substrate thickness is bigger than 2mm the BS reflected beams are sufficiently separated from the front side reflected beams so that they can be masked and do not reach the detection head. This separation can be optimized by adjusting the spot size of the beam on the sample (beam diameter & angle of incidence).

The following table suggests solutions for different substrate thickness ranges:

thickness/mm	Solution
< 0.1	Measure as it is. Model calculations assume infinite number of coherent BS reflected beams.
0.1 – 1	Measure as it is. Model calculations assume infinite number of incoherent BS reflected beams.
1 – 2	Mask the BS reflected beams with appropriate spot size (beam diameter & angle of incidence).
> 2	Mask the BS reflected beams.

Note that BS reflection effects can be observed for all materials that are transparent in a part of the light spectrum. For glass the transparent region is in the visible, for other materials such as Si, InP, GaAs the transparent region is in the NIR.

The following graph shows calculated spectra for a 1mm thick glass sample, with (dots) and without (lines) BS reflected beams:



In the low energy range the two spectra are very different. That is the spectral range where the BS reflection affects the measurements. In the high energy range the glass is absorbing so that no light reflected from the BS reaches the detector. The absorption edge of the glass is around 4.2eV.

11.4.1.3 Experiment

Since glass has a reflectivity of only ca. 5% the integration time is increased to 1000ms to improve signal-to-noise ratio (S/N).

The BS reflected beams will be included in the model: we have a real substrate (void) underneath the 1mm thick glass layer:

1 1000000.0	F glass.dsp	×	×
S	void.dsp	×	

The optical properties of the void substrate are of fixed index type dispersion with n = 1 and k = 0 over the whole spectral range.

The optical properties of glass can be modelled by a New Amorphous dispersion. Start values are for example:

Parameter value	Comment	
$n_{\infty} = 1.45$ Refractive index of glass at long wavelength		
$\omega_{\rm g} = 3.8$	Absorption edge, optical bandgap	
$f_{j} = 0.01$	Small value for small k	
$\omega_j = 12$	Absorption peak, in far UV	
$\Gamma j = 1.5$ Damping parameter: not critical as long as it is positi		





The following graph shows measurement (dots) and fit (lines):

The model described above did not given an excellent fit. For the transparent region of a material we would always expect Is = 0 (= 0) with or without BS reflection. The measurement does not give Is = 0. This is due to surface roughness on the glass.

Assuming that the sample is symmetrical, we therefore add a standard roughness layer $(30\text{\AA}, \text{EMA: }50\% \text{ glass } \& 50\% \text{ void})$ on both sides of the glass.

The final layer model would therefore be the following:

3	F 50.2 🗌 %	F glass.dsp	50.00 % ×	Void.ref	50.00 % × ×
2		F glass.dsp			××
1	F 35.0 🗌 %	F glass.dsp	50.00 % ×	Void.ref	50.00 % × ×
S		Void.ref			×

The fit is now much better:





We can derive the final dispersion parameters:

$$n_{\infty} = 1.373$$
$$\omega_{g} = 3.496 \text{eV}$$
$$f_{j} = 0.002$$
$$\omega_{j} = 7.987 \text{eV}$$
$$\Gamma_{j} = 0.063$$

Leading to the following dispersion curves:



11.4.2 a-Si on glass

11.4.2.1 Key points

- New Amorphous and Tauc Lorentz dispersion
- spectral range for thickness determination
- importance of overlayer

11.4.2.2 Typical settings of acquisition parameters:

Sample	a-Si on glass
Spectral range	1.5 – 6.5eV (825 – 190nm)
Increment	0.050eV
AOI	55° - 60°
Spot size	1mm x 3mm



Explore the future



Integration time	1000ms		
Configuration	II		

11.4.2.3 Experiment

The following graph shows a measurement of a-Si on a glass substrate:



The spectrum can be divided into two parts:

For energies < 2.5eV there are interference fringes indicating that the material a-Si is transparent in this spectral region. The thickness of the a-Si layer is basically determined in that spectral range.

For energies > 2.5eV the fringes disappear, a-Si is absorbing in this spectral range.

11.4.2.4 Analysis

A first attempt is to model a-Si as a single layer with two different dispersion formula: a single layer model with New Amorphous dispersion gives a fit value of $\chi^2 = 87$ while using a Tauc-Lorentz dispersion leads to $\chi^2 = 32$.

The fit result is better with the Tauc Lorentz formula. However, the fit can further be improved to $\chi^2 \approx 8$ when an overlayer is introduced to the layer model:

3	40.9	%	asi formation.dsp	50.00 % ×	Void.ref	50.00 % ×	×
2	2370.4		asi formation.dsp			×	×
1	1000000.0		1737.dsp			×	×
S			Void.ref			×	

This overlayer is an EMA of 50% a-Si and 50% void. Physically it corresponds to a roughness layer + a native oxide layer that forms on a-Si (as it does on c-Si and p-Si).

The following graph shows the experimental data (dots) and the final fit (lines):



The deduced dispersion parameters for a-Si are then:

$$E_g = 1.672 eV$$

 $\varepsilon_{\infty} = 0.764$
 $A = 222.674$
 $E_0 = 3.609$
 $\chi^2 = 2.272$

The corresponding dispersion is displayed in the graph below:



11.5 Composed Materials

11.5.1 Oxidized porous Silicon Layer (OPSL)

11.5.1.1 Key points

- inhomogeneity of OPSL
- EMA: composition of SiO2 and c-Si (crystalline Si)
- Model improvement step by step
- Total thickness of OPSL layer

11.5.1.2 Typical settings of acquisition parameters:

Sample	10µm OPSL	Reference (0.5µm SiO ₂)		
Spectral range	0.75-4.5eV (1655-275nm)	0.75-4.5eV (1655-275nm)		
Increment	0.005eV	0.050eV		
AOI	70°	70°		
Spot size	1mm x 3mm	1mm x 3mm		
Integration time	200ms	200ms		

11.5.1.3 Measurement of the reference sample:

In this example we analyze a sample coated with a layer of silicon and OPSL (approximately 10 μ m thick) and a reference sample coated with a SiO₂ layer (approximately 0.5 μ m thick).

The reference is analyzed to determine the optical properties of the material in a simple layer structure. As a first step we use the reference file to model SiO_2 and introduce a roughness layer on top (EMA: 50% $SiO_2 \& 50\%$ void):

2	F	53.0 🗌 %	Sio2_isa.ref	50.00 % ×	Void.ref	50.00 % × ×
1	F	6576.7	Sio2_isa.ref			××
S			C-si_isa.ref			×





Below the experimental (dots) and fit (lines) data:

As a result we get:

$$\chi^2 = 3.82$$

thickness SiO₂ = 6576.7Å
thickness roughness layer = 53Å

The fit is excellent so that the optical properties of SiO_2 correspond to the used reference file.

The dispersion of SiO_2 is displayed below:





11.5.1.4 Measurement of the OPSL sample:

Below the acquired spectrum:





HORIBA

The spectrum is acquired in the range 0.750 - 4.5eV (275-1700nm) which corresponds to the transparent region of SiO₂.

The amplitude of the interference fringes decreases fast around 1.5eV. We can ascribe this unexpected behaviour to the fact that the OPSL layer is a mixture of transparent SiO₂, void and c-Si. c-Si is already absorbent around $2 \sim 3eV$.

Because of the absorption we better fit in the spectral range 0.75 - 1.40eV (885 - 1700nm) to determine the thickness of the OPSL layer.

11.5.1.5 **OPSL** results and comments

The OPSL layer should be a mixture of SiO₂, c-Si and void. Thus the layer is a mixture of a high index component (C-si) and two low index components (SiO₂ and void). The index of void and SiO₂ are close compared to the index of c-Si so that we actually cannot distinguish between SiO₂ and SiO₂ + void. There would be strong correlation between these two components.

Therefore we can start with the following model: one layer which is composed of c-Si with inclusions of SiO₂:

¹ F 100000.0 F %	Sio2_isa.ref	20.00 % × C-si_isa.ref	80.00 % × ×
S	C-si_isa.ref		×

This first attempt provides a very bad fit with $\chi^2 \approx 2500$ so that we have to improve the model.

The process of OPSL deposition suggests a dependency of SiO_2 proportion on depth within the film. So that we create a succession of EMA layers with increasing SiO_2 content from bottom to top:

2 F	90000.0 F %	Sio2_isa.ref	60.00 % ×	C-si_isa.ref	40.00 % × ×
1 F	1000.0 F %	Sio2_isa.ref	20.00 % ×	C-si_isa.ref	80.00 % × ×
S		C-si_isa.ref			×

We choose 90000Å for the top layer and 1000 Å for the bottom layer. The total thickness of the layers should be around 100000Å. Then there are two possible approaches: the top layer has a high index (higher c-Si content than SiO₂) or a low index (lower c-Si content than SiO₂). Both approaches lead to bad fit results with χ^2 values around 500 ~ 570. However, fits are already better than in the previous model.

We continue adding layers:

³ F	90000.0 F %	Sio2_isa.ref	80.00 % ×	C-si_isa.ref	20.00 % × ×
2 F	1000.0 F %	Sio2_isa.ref	60.00 % ×	C-si_isa.ref	40.00 % × ×
1 F	1000.0 F %	Sio2_isa.ref	20.00 % ×	C-si_isa.ref	80.00 % × ×
S		C-si_isa.ref			×

The χ^2 value drops now sharply to 80 ~ 90. We continue adding layers:

4	F	90000.0 F %	Sio2_isa.ref	90.00 % ×	C-si_isa.ref	10.00 % × ×
3	F	1000.0 F %	Sio2_isa.ref	80.00 % ×	C-si_isa.ref	20.00 % × ×
2	F	1000.0 F %	Sio2_isa.ref	60.00 % ×	C-si_isa.ref	40.00 % × ×
1	F	1000.0 F %	Sio2_isa.ref	20.00 % ×	C-si_isa.ref	80.00 % × ×
S			C-si_isa.ref			×

The χ^2 value drops to ~10.

We add now a final layer close to 100% SiO₂ on top. We suppose that this layer is the thickest, however, we try now different thickness start values by using the *multiguess absolute* function. We try start values in the interval 80000 - 110000Å (increment 5000Å) and get the following final result:

⁵ F	90000.0 F %	Sio2_isa.ref	97.00 % ×	C-si_isa.ref	3.00 % × ×
4 F	1000.0 F %	Sio2_isa.ref	90.00 % ×	C-si_isa.ref	10.00 % × ×
³ F	1000.0 F %	Sio2_isa.ref	80.00 % ×	C-si_isa.ref	20.00 % × ×
2 F	1000.0 F %	Sio2_isa.ref	60.00 % ×	C-si_isa.ref	40.00 % × ×
1 F	1000.0 F %	Sio2_isa.ref	20.00 % ×	C-si_isa.ref	80.00 % × ×
S		C-si_isa.ref			×

We get a χ^2 value of 4.73.

Physically the top layer should be 100% SiO2. So we add a "thin" layer of pure SiO2 on top:

6	F	1000.0	Sio2_isa.ref			××
5	F	90000.0 F %	Sio2_isa.ref	97.00 % ×	C-si_isa.ref	3.00 % × ×
4	F	1000.0 F %	Sio2_isa.ref	90.00 % ×	C-si_isa.ref	10.00 % × ×
3	F	1000.0 F %	Sio2_isa.ref	80.00 % ×	C-si_isa.ref	20.00 % × ×
2	F	1000.0 F %	Sio2_isa.ref	60.00 % ×	C-si_isa.ref	40.00 % × ×
1	F	1000.0 F %	Sio2_isa.ref	20.00 % ×	C-si_isa.ref	80.00 % × ×
S			C-si_isa.ref			×

The experimental (dots) and fit (lines) data are shown below:





The results for thickness and composition of each individual layer can be seen in the next figure:

6	F	1220.1	Sio2_isa.ref			×	×
5	F	88381.6 F %	Sio2_isa.ref	98.18 % ×	C-si_isa.ref	1.82 % ×	×
4	F	2328.6 F %	Sio2_isa.ref	92.88 % ×	C-si_isa.ref	7.12 % ×	×
3	F	1714.8 F %	Sio2_isa.ref	75.77 % ×	C-si_isa.ref	24.23 % ×	×
2	F	1038.9 F %	Sio2_isa.ref	44.54 % 🗡	C-si_isa.ref	55.46 % ×	×
1	F	799.3 F %	Sio2_isa.ref	10.76 % ×	C-si_isa.ref	89.24 % ×	×
S			C-si_isa.ref			×	

The best χ^2 value is 2.60 with a total OPSL layer thickness of 95483.3Å.

11.6 Multilayers

11.6.1 Eight layer High-Low Stack on Silicon

11.6.1.1 Key points

- layer repetition
- layer correlation

11.6.1.2 Typical settings of acquisition parameters:

Sample	4 repetitions of TiO_2/SiO_2 on Si
Spectral range	1.5 – 6.5eV (825 – 190nm)
Increment	0.050eV
AOI	70°



Spot size	1mm x 3mm
Integration time	200ms
Configuration	II

In this example we will fit ellipsometric data from a eight layer high-low index stack deposited on silicon substrate to determine the refractive indexes and thicknesses of all eight layers. The sample structure is approximately 4 x (1150Å TiO_2 / 710 Å SiO_2) / Si:

TiO ₂
SiO ₂
TiO ₂
SiO ₂
Si substrate

11.6.1.3 Experiment

The corresponding Layer model is the following::

3 F	30.0 🗆 %	E tio2.dsp	0.00 % × Void.ref	0.00 % × ×
1 2 F 7	/10.0	Ftio2.dsp		××
1 E 11	50.0	Sio2_isa.ref		××
S		C-si_isa.ref		×

It consists of a repetition (4x) of the SiO_2/TiO_2 stack and a roughness layer on top. The measurements and the fit with this model is shown below.



We get the average thickness of the SiO_2 and the TiO_2 layer as well as the average dispersion of the TiO_2 layer. Additionally we get of course the thickness of the roughness layer:

Average thickness SiO_2 : 1162Å Average thickness TiO_2 : Å Thickness roughness: Dispersion of TiO_2 :



11.7 Complicated dispersions

11.7.1 NK-fit for an organic layer

11.7.1.1 Key points

- Performing a NK-fit
- Creation of calculated files *.clc
- Refractive index /dielectric constant extension to a higher range

11.7.1.2 Typical settings of acquisition parameters:

Sample	Organic layer
Spectral range	1.5 – 6.5eV
Increment	0.050eV



AOI	70°
Spot size	1mm x 3mm
Integration time	200ms
Configuration	II

This analysis is performed on an unknown organic layer of ca. 400Å thickness. The difficulty here is that none of the dispersion models is able to provide a good fit over the entire measurement range of 1.5 - 6.5eV. A good fit can only be obtained in the reduced spectral range of 1.5 - 5eV.

Thickness and optical properties can be determined from 1.5 - 5eV. To get the optical constant also from 5eV to 6.5eV with the same precision that we have for the reduced range 1.5-5eV we can use the *NK-fit* function. This function allows direct calculation of n and k in the spectral region of interest if all other parameters including thickness are well known and fixed.

The model for the reduced range of 1.5 - 5eV is a single layer model with a multiple oscillator dispersion. In this case we use *Triple New Amorphous* (classical dispersion would provide similar results). For the substrate we use the material file Csi_wor.ref that provides data points in the UV region, which we need later:

¹ F 419.2	FL1 organic.dsp	××
S	Csi_wor.ref	×

It is not straight forward to find the set of parameters for this kind of dispersion.

The method we suggest is to begin with simple or double oscillators and to launch a fit on the interval 1.5 - 3.5eV or 1.5 - 4eV. Try different start values for the dispersion parameters until the χ^2 value cannot be improved anymore.

Select now a dispersion of the same type with a higher number of oscillators (for example New Amorphous \rightarrow Double New Amorphous) and launch the fit on the same interval (using the previously found parameters as start values).

If there is an improvement of the fit try to extend the range, for example: 1.5 - 4eV or 1.5 - 4.5eV.

Carry on adding oscillators and extending the range as long as the χ^2 value can be improved.

With Triple New Amorphous we get a χ^2 value of 2 and a thickness of 419.2Å in the range 1.5 - 5eV. The dispersion parameters are displayed below. Once we exceed 5eV the fit starts to deteriorate and the χ^2 value increases. That is the point where we better use the NK-fit function to obtain *n* and *k* until 6.5eV.

The dispersion parameters (triple new amorphous) for the reduced range 1.5 - 5eV are:

n _∞	1.6658216
ω _g	2.7930357
F ₁	-0.0048616
ω ₁	4.6567426
Γ_1	0.3770423



F ₂	0.1749506
ω ₂	3.0110691
Γ_2	0.1551019
F ₃	-0.0077110
ω ₃	3.8020499
Γ ₃	0.2434855

To use the NK-fit function click on *Mat*. and select *Point by Point Calculated*. Drag and drop the dispersion that was found in the reduced range to the material 1 box and activate the box:



Then replace the dispersion layer of your model by the PPC layer you just created:

1	419.2	₩ <mark>F</mark> L1 organic PPC.ppc	×	×
S		Csi_wor.ref	×	

The dispersion of the layer is now fitted at a fixed layer thickness (obtained by fitting in the reduced range).

Launch the fit in the spectral region 1.5 - 6.5eV (using the Levenberg-Marquardt algorithm). The fit should give a χ^2 value of 0. Update the model and the PPC file to get the dispersion of the layer in the complete spectral range.

Opening the PPC file in the material folder to get access to the optical constants:





You can also create a *.clc file which can be used as a reference file of the organic layer by activating the box *Table values* and clicking on the icon $\boxed{100}$.

Now we can use the PPC file in the model:

1 F 4	419.1	L1 organic CLC.clc	×	>	<
S		Csi_wor.ref	×		

and fit the thickness over the entire spectral range:



The result is:

thickness =
$$419.125$$
Å
 $\chi^2 = 0.019893$

Note that χ^2 value is not exactly 0. This is due to the fact that for the PPC calculation we used start values (thickness of 419.2Å) that are not 100% exact. Hence, the calculated material contains little imperfections but is still very close to reality.

11.7.1.3 Interpretation

The starting point of the NK-fit is to obtain the best fit possible in a reduced spectral range (here for example 1.5 - 5eV instead of 1.5 - 6.5eV) to determine the thickness and dispersion of the layer.

Then these values are fixed and the optical constants n and k are calculated in the complete spectral range.

However, keep in mind the NK-fit provides results from a calculation and not from a measurement. The quality of the results depends directly on the quality of the model. Important is to obtain a very good fit in the reduced spectral range.

11.7.2 Quaternary semiconductor material

11.7.2.1 Key points

- Alloy function

11.7.2.2 Typical settings of acquisition parameters:

Sample	Photodiode structure
Spectral range	1.5 – 6.5eV
Increment	0.050eV
AOI	70°
Spot size	1mm x 3mm
Integration time	200ms
Configuration	II

Sample structure:

Native oxide
InP
In _{0.76} Ga _{0.24} As _{0.57} P _{0.43}
InGaAs
c-InP

11.7.2.3 Experiment

The corresponding layer model is the following:

4 F 30.0 □%	F InP.dsp	0.00 % × Void.ref	0.00 % × ×
³ F 20000.0	E InP.dsp		x x
2 F 1200.0	InGaAsP.aly (0.430)		x x
¹ F 13000.0	E InGaAs.dsp		××
S	F InP.dsp		×

It consists of a InP substrate, a InGaAs, InGaAsP and InP layer with an overlayer on top. The dispersion of InP and InGaAs was modeled with the Kato-Adachi dispersion formula. For the quaternary material the alloy function was used.

The measurements and the fit with this model is shown below.





As a result we get the thickness of each individual layer and the dispersion of the materials. For the quaternary material we also can derive the composition:

```
Thickness InGaAs: 13095Å
Thickness In<sub>0.76</sub>Ga<sub>0.24</sub>As<sub>0.57</sub>P<sub>0.43</sub>: 1192Å
Thickness InP: 20525Å
Thickness overlayer: 21Å
```

Dispersions:





11.8 Inhomogeneous layers

11.8.1 Polymer / ITO on glass substrate

11.8.1.1 Key points

- graded index layer

11.8.1.2 Typical settings of acquisition parameters:

Sample	Polymer / ITO on glass
Spectral range	1.5-6.5eV (825-190nm)
Increment	0.050eV
A.O.I	58°
Spot size	1mm
Integration time	200ms

The measurements and the fit with this model is shown below.

Polymer
ITO
Glass

11.8.1.3 Experiment

ITO is a semi-transparent conducting material that exhibits absorption in the FUV and NIR. It is known to be inhomogeneous due to deposition methods and post-treatments. A graded layer model has therefore to be taken into account:

2 F 200.0		Fpolymer.dsp	×	×
1 E 1200.0	Т	FITO_top.dsp	x	x
	В	FITO_bottom.dsp	×	
S		Glass.dsp	x	

For both the polymer and the ITO we use the classical oscillator model for the dispersion.

The experimental data and the fit are shown below:





As a result we get the thickness of both layers and the optical constants:



Dispersions:





11.9 Parameter correlation

11.9.1 TiO₂ on Glass: Bound Multimodel

11.9.1.1 Key points

- Bound Multimodel
- Transparent substrate
- Reflection and Transmission data

11.9.1.2 Typical settings of acquisition parameters:

Sample	TiO ₂ on glass
Spectral range	1.5 – 6.5eV (190 – 825nm)
Increment	0.050eV
AOI (SE)	70°
Spot size	1mm x 3mm
Integration time	200ms
Configuration	II

Sample structure:

TiO ₂		
Glass substrate		

In this example we analyze a TiO_2 layer on a glass substrate. The TiO_2 layer is not homogeneous in depth. Therefore we model the TiO_2 layer with two layers:

A top layer of TiO_2 modelled by an appropriate dispersion formula and a bottom layer described by an EMA of TiO_2 and Void.

The sample is equally coated on both sides. Measurements are performed in transmission and in reflection.

The model for the reflection measurement is the following:

2 F	796.3	FTiO2_Di52T2-1_mm16.dsp	××	
F	776.4 F %	FTiO2_Di52T2-1_mm1 92.68 % × Void.ref	7.32 % × ×	2.4
S		glassnew.dsp	×	




On layer 1 the thickness is fitted in the multiguess absolute mode (range: 0.1 to $1000\text{\AA}/$ increment: 200\AA).

The model for the transmission measurement is the following:

⁵ F 796.3	FTiO2_Di52T2-1_mm16.dsp	××
⁴ F 776.4 F	% FTiO2_Di52T2-1_mm1 92.68 % × Void.ref	7.32 % × ×
³ 🔲 1000000.0	glassnew.dsp	××
² L4*1.000	% FTiO2_Di52T2-1_mm1 92.68 % × Void.ref	7.32 % × ×
1 L5*1.000	FTiO2_Di52T2-1_mm16.dsp	××
S	Void.ref	×

Here the multiguess absolute mode (range: 0.1 to 1000\AA / increment: 200\AA) for fitting the thickness is applied to layer 4.

We set a correlation between layer 2 and 4, between layer 1 and 5 and between the proportions of layer 2 and 4. The "*Correlated to*" feature can be selected by left-clicking on the thickness/proportion value of the layer of interest.

In transmission the light beam goes through the entire sample. Therefore we set the Void.ref file as the substrate in the layer model.

Now we can combine the two different measurements and two different models for the sample. The dispersion that is used for TiO_2 must be of course the same for the two models. Also, the thickness of the corresponding layers must be the same.

To meet these requirements we use the Bound Multimodel feature of the DP2 software. This feature allows to bind parameters of different models.

Click on Mod. and choose Bound Multimodel. The following window appears:

Bound Multimodel Drop model files in the list below M1 - Transmission TiO2.mdl M2 - Reflection TiO2.mdl Experimental file: Results Image: Contract of the list below Image: Contract of the list of the list below Image: Contract of the list of the	- 8 ×
Drop model files in the list below M1 - Transmission TiO2.mdl M2 - Reflection TiO2.mdl Experimental file : <results\acquisition data=""> T_Di52T2.spe 5 TiO2_Di52T121_mm16.dsp</results\acquisition>	
M1 - Transmission TiO2.mdl M2 - Reflection TiO2.mdl Experimental file : <results\acquisition data=""> T_Di52T2.spe S Ti02_Di52T21_mm16.dsp</results\acquisition>	
M2 - Reflection 1:02.mdl Experimental file : </th <td></td>	
Experimental file : <results\acquisition data=""> T_Di52T2.spe</results\acquisition>	
Experimental file : <results\acquisition data=""> T_Di52T2.spe</results\acquisition>	
Experimental file : <results\acquisition data=""> T_Di52T2.spe</results\acquisition>	
Kesults\Acquisition Data> T_Di52T2.spe 5 Ti02_Di52T2:1_mm16.dsp	
5 Ti02_D62T2:1_mm16.dsp	
5 <u>Ti02_Di52T21_mm16.dsp</u>	
5 <u>Ti02_Di52T21_mm16.dsp</u>	
5 Ti02_Di52T2:1_mm16.dsp	
5 TiO2_Di52T2-1_mm16.dsp	
5 Ti02_Di52T2-1_mm16.dsp	
5 TiO2_Di52T2-1_mm16.dsp	
4 TiO2_Di52T2-1_mm16.dsp	
2 TiO2_Di52T2-1_mm16.dsp	
1 TiO2_Di52T2-1_mm16.dsp	
S Void.ref	
Save Save As Signal Fit Update models	

Drop the model files of the transmission and reflection measurements in the upper left box of the Bound Multimodel window. To display one of the models, just click on it.

Binding the parameters:

Click on bind at the bottom of the window. The following window appears:

Bind Parameters								
		Model2						
		AOI	L1d	L1 % TiO2_Di52T2-1_mm16.dsp	L2d			
	AOI							
	L1d							
	L2d							
-	L2 % TiO2_Di52T2-1_mm16.dsp							
del	L3d							
Mo	L4d		V					
	L4 % TiO2_Di52T2-1_mm16.dsp							
	L5d							

In this case model 1 is the model for the transmission measurement and model 2 is the model for the reflection measurement.

Now we activate the boxes to bind the following parameters:

- thickness of layer 4 of model 1 with thickness of layer 1 of model 2
- thickness of layer 5 of model 1 with thickness of layer 2 of model 2
- composition of layer 4 of model 1 with the composition of layer 1 of model 2.

Now we can launch the fit.

11.9.1.3 Results and Comments

The following two graphs show the experimental results and fits. We obtain the following set of parameters with a χ^2 value of 0.644768:

χ' minimization on Transmission							
$\chi^2 = 0.644768$							
Iterations Number = 13							
Parameters							
1) M1 - L1 Thickness [Å]	= 788.861 ± 0.000						
2) M1 - L2 Thickness [Å]	= 783.071 ± 0.000						
3) M1 - L4 Thickness [Å]	= 783.071 ± 9.631						
4) M1 - L5 Thickness [Å]	$=$ 788.861 \pm 10.274						
5) M2 - L1 Thickness [Ă]	$= 783.071 \pm 0.000$						
6) M2 - L2 Thickness [Ă]	$= 788.861 \pm 0.000$						
7) M1 - L2 % TiO2_Di52T2-1_mm16.dsp	= 92.70 ± 0.00						
8) M1 - L4 % TiO2_Di52T2-1_mm16.dsp	$=$ 92.70 \pm 0.15						
9) M2 - L1 % TiO2_Di52T2-1_mm16.dsp	= 92.70 ± 0.00						
10) TiO2_Di52T2-1_mm16 n∞	$=$ 1.5105450 \pm 0.0278058						
11)TiO2_Di52T2-1_mm16 ωg	$=$ 1.3667940 \pm 0.1395556						
12) TiO2_Di52T2-1_mm16 fj	$=$ 0.0527662 \pm 0.0133738						
13)TiO2_Di52T2-1_mm16 ωj	$=$ 6.6452490 \pm 0.2127505						
14) TiO2_Di52T2-1_mm16 Гј	$=$ 0.6177589 \pm 0.1276324						

The next step is to update the models by clicking on *update*. The models with the final parameters are then:

For reflection:

2	F	788.9	FTiO2_Di52T2-1_mm16.dsp	××
1	F	783.1 F %	FTiO2_Di52T2-1_mm1 92.70 % × Void.ref	7.30 % × ×
S			glassnew.dsp	×

For transmission:

5	F 788.9	FTiO2_Di52T2-1_mm16.dsp		××
4	F 783.1 F %	F TiO2_Di52T2-1_mm1 92.70 % ×	Void.ref	7.30 % × ×
3		glassnew.dsp		××
2	783.1 🗆 %	F TiO2_Di52T2-1_mm1 92.70 % ×	Void.ref	7.30 % × ×
1	788.9	FTiO2_Di52T2-1_mm16.dsp		××
S		Void.ref		×

The thickness results are now written in the model. The dispersion of TiO_2 is shown below:





11.10Non-cubic Crystals

11.10.1 hexagonal Al_2O_3 / Al

11.10.1.1 Key points

- uniaxial anisotropy

11.10.1.2 Typical settings of acquisition parameters:

Sample Al ₂ O ₃ / Al			
Sportrol range	UVISEL: 1.5 – 6.5eV (825 – 190nm)		
spectral range	(MM-16: 430 – 850nm)		
Increment	UVISEL: 0.050eV		
AOI	70°		
Spot size 1mm x 3mm			
Integration time 200ms			
Configuration	II		

Sample structure:

Anisotropy Aluminium oxide	
Al substrate	







Measurements with the UVISEL at the same spot but under different rotation around the sample normal are equal thus indicating uniaxial anisotropy of N-type. The corresponding layer model is therefore:

	Е	FAI2O3-E.dsp	x x
	0	FAI2O3-O.dsp	×
S		Al_asp.ref	×

It consists of one uniaxial anisotropic layer (N polarization) with two different dispersions for the ordinary (O) and the extra-ordinary direction (E). The dispersion model is the classical oscillator model:

$$\varepsilon = 1 + \frac{(\varepsilon_{s} - 1)\omega_{t}^{2}}{\omega_{t}^{2} - \omega^{2} + i\Gamma_{0}\omega} \quad \text{with } \Gamma_{0} = 0$$

The experimental data and the fit are shown below:



As a result we get the thickness of the Al₂O₃ layer and the anisotropic optical constants:

Thickness Al₂O₃: 19940Å Dispersions Al₂O₃:



One can observe a birefringence of $n = 0.01 = n_0 - n_E$. Note that the Al₂O₃ layer is porous and has therefore a rather low refractive index.



11.10.2 Strained materials

11.10.3 stretched polymer foil

11.10.3.1 Key points

- biaxial anisotropy

11.10.3.2 Typical settings of acquisition parameters:

Sample	PET foil
Spectral range	UVISEL: 1.5 – 6.5eV (825 – 190nm)
Increment	UVISEL: 0.050eV
AOI	70°
Spot size	1mm x 3mm
Integration time	200ms
Configuration	II

11.10.3.3 Experiment

Mueller Matrix measurements with the MM-16 show that the PET foil exhibits anisotropy because the off-diagonal elements are not 0. Before the actual measurements were performed the optical axes of the sample have to be found. This is achieved by looking at the sample between crossed polarizers and by using a microscope in conoscopic mode.

With the crossed polarizers we can find the trace of the optical axes:





The microscope in conoscopic mode yields the type of anisotropy (uniaxial or biaxial) and the direction of the optical axes by comparing the images with references from literature.

The microscope in conoscopic configuration:



Below we see the image compared with theoretical shapes:





From there we get the information that the anisotropy is biaxial and the direction of the optical axes.

We now can align the sample for subsequent ellipsometer and transmission measurements.

The layer model is:



containing a biaxial layer with Euler angles and taking into account backside reflections. Euler angles:



The dispersions for x, y, z are described with a classical oscillator model:

$$\epsilon = \epsilon_{\infty} + \frac{(\epsilon_{s} - \epsilon_{\infty})\omega_{t}^{2}}{\omega_{t}^{2} - \omega^{2} + i\Gamma_{0}\omega}$$

First we fitted measured ellipsometric transmission data (three AOI: 5°, 20°, 35°):



The results could be confirmed by fitting the ellipsometric data in reflection (AOI = 70°):





As a result we get the thickness of the foil:

```
        Parameters

        1) L1 Thickness [μm] = 188.988900

        2) L1 Φ Rotation = 0.2

        3) L1 Θ Rotation = -0.2

        4) L1 ψ Rotation = -0.4
```

and the optical constants:



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 $n_x \text{ at } 633 \text{ nm} = 1.684$ $n_x - n_y = -0.021$ $n_x - n_z = 0.168$



Next pages show some Application Models sum up on Data Sheets.





- Fit parameters
 - Thickness : layers 1 and 2
- Materials
 - c-Si, SiO₂, void: Reference library



General Remarks

• Adding a surface roughness on top of the SiO₂ film provides an improvement of the χ^2 by a factor 2. Hence the final model takes into account the roughness.

• Using a dispersion formula for SiO₂ materials, eg adjusting the refractive index of SiO₂ doesn't provide an improvement of the χ^2 .

x² minimization on Is, Ic

 $Is = Sin(2\Psi) \times Sin(\Delta), Ic = Sin(2\Psi) \times Cos(\Delta)$ S' = 3.852280Iterations Number = 12

Parameters

1) L1 Thickness [Å] = 5227.530 ± 4.931 2) L2 Thickness [Å] = 73.290 ± 9.848

Correlation matrix

=1=	=2=
1.000	-0.980
	1.000











General Remarks

• Adding a roughness layer on top doesn't improve significantly the results. Others techniques may help to state on the final model.

Spectroscopic Ellipsometry Data Sheets

AIN/ Glass Substrate BR

Model

³ F	56.3 🗌 %	F AIN.dsp	50.00 % × Void.ref 50.00 %	% ×	×	
2 F	6728.4	F AIN.dsp		×	×	
1 🗆 1	0000000.0	1737.dsp	Backside reflection	×	×	
S		Void.ref	model	×		

Fit Conditions

Fit Results

- Spectral range : 0.75 4.75 eV
- Step size: 0.05 eV
- Angle of incidence : 70°
- Model Information: Backside reflection model
- Fit parameters
 - Thickness : layers 2 and 3
 - Optical constants: AIN
- Materials
 - 1737, Void: Reference library
 - AIN: Classical formula, transparent Lorentz oscillator with $\epsilon_{\rm \infty}=1.0$



Optical constants of AIN



3

FAIN.dsp 1737.dsp Void.ref	x x x
1737.dsp Void.ref	×
Void.ref	×
in(2Ψ) ×Cos(Δ) 474 ± 62.200 2920 ± 0.0627 9543 ± 0.0849 1852 ± 0.0369	0.4 0.3 0.3 0.2 0.2 0.1 15 0.1 0.0 0.0 0.0 -0.1 -0.1 -0.2
	<i>in(29) ×Cos(∆)</i> 474 ± 62.200 2920 ± 0.0627 9543 ± 0.0849 1852 ± 0.0369

General Remarks

• Adding a rough surface layer provides a decrease of the χ^2 from 28 to 6, hence the final model chosen includes the rough overlayer.

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Corre	lation	matri	x		
=1=	=2=	=3=	=4=	=.5=	=6=
1.000	-0.497	-0.318	-0.179	0.210	0.129
	1.000	0.612	0.024	-0.458	-0.704
		1.000	0.653	-0.828	-0.528
			1.000	-0.850	0.282
				1.000	0.123
					1.000

25

3

Photon Energy (eV)

35

4

2.4

2.2 80,

1.5

2



201



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2

2.5

3

Photon Energy (eV)

3.5

4

45



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Spectroscopic Ellipsometry	Data Sheets		
SiO ₂ /p	o-Si / SiO ₂ / c-Si Substrate		
Model			
³ F 530.0	SiO2_wor.ref	×	×
² F 3978.5	P-si-as_jel.ref	×	×
1 F 227.0	SiO2_wor.ref	×	×
S	Csi_wor.ref	×	

Fit Conditions

- Spectral range : 1.5-5.0 eV
- Step size: 0.05 eV
- Angle of incidence : 70°
- Fit parameters
 Thickness : layers 1, 2 and 3
- Materials
 - SiO₂, p-Si (as doped), c-Si: Reference library

Fit Results



General Remarks

• The fit results is quite good by using the reference materials of the library.

• It may be improved by improving the p-Si optical constants.

 χ^2 minimization on Is, Ic

 $I_{S} = \frac{Sin(2\Psi) \times Sin(\Delta)}{4}, I_{C} = \frac{Sin(2\Psi) \times Cos(\Delta)}{4}$ $\Psi^{2} = 4.082222$ Iterations Number = 8

Parameters

L1 Thickness [Å] = 226.951 ± 7.109
 L2 Thickness [Å] = 3978.528 ± 4.299
 L3 Thickness [Å] = 530.018 ± 0.637

Correlation matrix

=T=	=2=	=3=
1.000	-0.358	0.029
	1.000	-0.018
		1.000

8

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General Remarks

• Adding a rough overlayer strongly decreases the χ^2 value.

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Explore the future

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Model without roughness



Fit Results

χ minimization on Is,Ic
$Is = Sin(2\Psi) \times Sin(\Delta), Ic = Sin(2\Psi) \times Cos(\Delta)$
52.387940
Iterations Number = 1
Parameters
1) L1 Thickness [Å] = 8513.500 ± 112.703

2)	aSi_TL	Eg	=	2.0708	±	0.0256
3)	aSi TL	œ3	=	3.6028	±	0.1968
4)	aSi TL	A	=	124.7316	±	10.1517
5)	aSi ^T L	E_0	=	3.2131	±	0.0537
6)	aSi TL	C	=	1.5282	±	0.1037

Correlation matrix

=1=	=2=	=3=	=4=	=5=	=6=
1.000	-0.004	-0.597	-0.352	0.300	-0.197
	1.000	-0.293	0.887	-0.575	0.337
		1.000	-0.191	-0.474	-0.246
			1.000	-0.583	0.601
				1.000	-0.167
					1.000

General Remarks

• Amorphous silicon is very often a rough material, and adding a rough overlayer is of major importance to increase the fit results.



Explore the future



HORIBA

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Explore the future

Model without roughness

² F 364.9	Fa-Si.dsp	×	×
1 500000.0	1737.dsp	×	×
S	void.dsp	×	

Fit Results

x²	mi	nin	niza	tic	o a	n	Is,	, I	C			
Is		11 m	2(2)	×Sii	a (A)		Ic .	=	Sin	29)	*Cos	F(A)
X ²	= 3	.03	2675									
Ite	erat	101	IS NI	umbe	er =	6						
Pa	ram	ete	rs									
1)	L2	Thi	ckne	235	[Å]	=	364	4.	929	±	3.8	337
2)	a-S	i	Eg			=		1.	4718	± ±	0.0	655
3)	a-S	i	200			=		1.	0871	. ±	0.1	241
4)	a-S	i	A			=	14'	7.	7662	±	11.2	2022
5)	a-S	i	E ₀			=	-	з.	5311	. ±	0.0	427
6)	a-S	i	c			=	2	2.	1975	ŧ	0.0	619
Co	rre	lat	ion	ma	tri	x						
=1=		=2	=	=3	=		4=		=5	=	=6	=
1.0	00	0.	358	-0.	225	0	.21	5	-0.	076	-0.	208
		1.	000	-0.	800	0	.96	8	-0.	829	-0.	236
				1.	000	-0	.08	1	ο.	231	0.	048
						1	00	0	-0.	910	-0.	093
									1.	000	0.	120
											1.	000

0.2 0.8 0.6 0.0 0.4 -0.2 Is 0.2 lc -0.4 0 -0.2 -0.6 -0.4 -0.8 1 2 3 4 5 6 Photon Energy (eV)

General Remarks

• The rough overlayer on top of amorphous silicon has a strong impact on the decrease of the χ^2 . Hence the final model includes the rough overlayer on top.







χ²	minim	izatio	on on	Is,I	С			
Is	= Sint	2\$)×Si.	n(Δ),	Ic =	Sin(2	\$P) /	< <i>Сов (Д)</i>	
X	= 2.999	9347	>					
lte	ration	Nambe	er = 4					
Par	amete	rs						
1)	L2 Thio	ckness	[Ă] =	1731	.508	±	9.112	
2)	L3 Thio	ckness	[Ă] =	30	.789	±	6.262	
3)	ITO oso	cdr ε	з =	3	.8893	±	0.0262	
4)	ITO oso	cdr ω	t =	5	.2784	±	0.0500	
5)	ITO oso	cdr ωj	p =	1	.9024	±	0.0098	
6)	ITO oso	$cdr = \Gamma_0$	=	0	.2824	±	0.0355	
7)	ITO_os(edr Fo	= £	0	.0939	±	0.0089	
								· · · ·
Сот	rrelat	ion ma	atrix					
=1=	=2=	=3)= =	=4=	=5=		=6=	=7=
1.00	00 -0.2	63 -0.	547 ().109	-0.42	4	0.192	-0.003
	1.0	00 0.	022 -0	0.002	0.05	1	-0.604	-0.019
		1.	000 0	0.701	0.78	7	0.172	-0.102
			1	000	0.53	3	0.250	-0.105
					1.00	0	0.108	-0.217
							1.000	-0.095
								1.000

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Model without roughness



Fit Results



General Remarks

• The presence of the rough overlayer improves the goodness of fit.

• ITO films are often known as inhomogeneous, and / or rough. We advise to test different models including single layer + roughness, single graded layer, 2 layers with 2 different ITO optical constants. To choose between those different models, weight the benefit gained on the χ^2 against the added complication (more regression parameters, more complicated model, etc).

Explore the future



1.000

-0.866

1.000

-0.616

0.636

1.000

0.368

-0.511

1.000

-0.798

0.232

0.399

-0.269

-0.281

1.000

-0.556

0.568

0.040 0.269

1.000

*In deltapsi2, build a new spectroscopic model, and put in the substrate She results found for layer 2, eg 72.37% c-Si + 19.24% a-Si + 8.39% void. Then under modeling description, click on edit, and in the modeling conditions tab, choose "User defined spectral range" and put 0.6-5 eV. Then, click on simulation, and in the results display area, select the (n,k) axis.

Explore the future

F 2966.5 F %	c-{ 61.93 % ×	Fa-: 19.48 % × vo 1	8.59 % × ×
F 5011.6	Sio2_wor.ref		××
	c-Si_Jy.clc		×
it Results			
<pre>minimization on Is,Ic = Sin(2Ψ) ×Sin(Δ), Ic = Sin</pre>	а (2¥) ×Cos (Д)		
<pre>minimization on Is,Ic = Sin(2Ψ)×Sin(Δ), Ic = Sin = 33.004960 eracions Number = 12</pre>	1 (2Ψ) ×Cos (Δ)	0.8	
<pre>minimization on Is,Ic = Sin(29)×Sin(Δ), Ic = Sin = 33.004960 eracions Number = 12 rameters L1 Thickness [Å] = 5011.62 L2 Thickness [Å] = 2966.53 L2 & c=Si Jy c c = 61.93</pre>	4 ± 11.051 4 ± 19.423 + 1.49	0.8 0.4 Is 0	
<pre>minimization on Is,Ic = Sin(29)×Sin(4), Ic = Sin = 33.004960 celacions Number = 12 arameters L1 Thickness [Å] = 5011.62 L2 Thickness [Å] = 2966.53 L2 % c-Si_Jy.clc = 61.93 L2 % a-Si.dsp = 19.48 a-Si Eq = 1.20</pre>	$(29) \times Cos(\Delta)$ 4 ± 11.051 7 ± 19.423 ± 1.49 3 ± 1.34 82 ± 0.0674	0.8 0.4 Is 0	
<pre>minimization on Is,Ic s = Sin(29)×Sin(A), Ic = Sin = 33.004960 eracions Number = 12 Arameters L1 Thickness [Å] = 5011.62 L2 Thickness [Å] = 2966.53 L2 % c-Si_Jy.clc = 61.93 L2 % a-Si.dsp = 19.48 a-Si Eg = 1.20 a-Si z∞ = 12.14 a-Si Å = 170.33</pre>	4 ± 11.051 7 ± 19.423 ± 1.49 3 ± 1.34 82 ± 0.0674 03 ± 1.1853 11 ± 14.6276	0.8 0.4 Is 0 -0.4	
<pre>minimization on Is,Ic = Sin(29) *Sin(A), Ic = Sin = 33.004960 terations Number = 12 arameters) L1 Thickness [Å] = 5011.62) L2 Thickness [Å] = 2966.53) L2 % c-Si_Jy.clc = 61.93) L2 % a-Si.dsp = 19.48) a-Si Eg = 1.20) a-Si Eg = 1.20) a-Si & = 12.14) a-Si & = 170.33) a-Si Eg = 3.07) a-Si C = 0.80</pre>	4 ± 11.051 7 ± 19.423 ± 1.49 4 ± 1.34 1 ± 1.1853 11 ± 14.6276 59 ± 0.0403 31 ± 0.0469	0.8 0.4 Is 0 -0.4 -0.8	
$\begin{array}{rcl} & \text{minimization on Is,Ic} \\ & = Sin(29) \times Sin(\Delta), & Ic = Sin \\ & = 33.004960 \\ & \text{teracions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ \hline & \text{arameters} \\ & \text{otractions Number} = 12 \\ & \text{arameters} \\ & aram$	4 ± 11.051 7 ± 19.423 ± 1.49 3 ± 1.34 82 ± 0.0674 03 ± 1.1853 11 ± 14.6276 59 ± 0.0403 31 ± 0.0469	0.8 0.4 IS 0 -0.4 -0.8 1 2 3 Photon Energy	4

General Remarks

• The rough overlayer on top of layer 2 decreases the χ^2 by a factor 10 !. Hence the final model includes the rough overlayer on top.



Explore the future



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General Remarks

• Adding a rough overlayer on top of the film stack decreases the χ^2 by a factor 3. Hence the final model takes into account the roughness.



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Explore the future




1000	.2	F ZnO	.dsp			××
		glas	snew_jy.ds	р		×
esults						
inimization	on Ta	Ic	1			
		A. L.		14		
Sin/29)×Sin	(Δ), IC =	Sin(29) ×Cos	(4)	· · ·	••	
5in/29)×5in 19.196420	(Δ) , $Ic =$		(4)	0.4	and the state	
= <u>Sin(29)</u> ×Sin = 19.196420 	(Δ) , $IC =$ c = 26		(4)	0.4	······································	*****
ameters	(<i>d</i>), <i>Ic</i> =		(Δ)	0.4 0.3 0.3	······································	*****
meters 1 Thickness	$(\Delta), IC = 26$ $[\bar{A}] = 1803$	5in(29)×Cos 8.248 ± 42.1	214	0.4 0.3 0.3	······································	
ations Number ameters 1 Thickness	$(\Delta), IC = 26$ $[\tilde{\lambda}] = 1800$	Sin(29) *Cos 8.248 ± 42. 0.6937 ± 2.	214 1827	0.4 0.3 0.3 0.2	······································	
Sie (29) \times Sin 19.196420 actions Number meters 1 Thickness n0 ε^{∞} n0 ε_{0}	$(\Delta), IC =$ $(\Delta), IC =$ $(\Delta) = 26$ $[\tilde{\lambda}] = 1800$ = -1 =	Sin(29) *Cos 8.248 ± 42.3 0.6937 ± 2.3 0.1145 ± 1.3	214 1827 2347	0.4 0.3 0.3 0.2 0.2		*****
$\frac{Sie(2\Psi) \times Sin}{19.196420}$ $\frac{meters}{10}$ $10 E_{g}$ $10 E_{g}$	(A), IC = c = 26 [Å] = 180 = - = -	Sin(29) *Cos 8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3	214 1827 2347 8993 1470	0.4 0.3 0.3 0.2 0.2 1s 0.1	······································	*****
meters 19.196420 meters 10 Fickness 10	(A), IC = c = 26 [Å] = 180 = - = - = -	8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1.	214 1827 2347 8993 1470	0.4 0.3 0.3 0.2 0.2 Is 0.1	······································	*****
$\begin{array}{c} \text{sin}(2\Psi) \times 5in\\ \text{19.196420}\\ \text{tions Number}\\ \hline \\ \text{Thickness}\\ 0 & \epsilon^{\infty}\\ 0 & \epsilon_{0}\\ 0 & \xi_{0}\\ 0 & \Gamma_{0}\\ 0 & \Gamma_{0}\\ 0 & \xi_{1}\\ 0 & \xi_{1}\\ 0 & \xi_{1}\\ \end{array}$	(Δ), Ic = c = 26 [Å] = 180 = - = - = -	8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.3	(<i>d</i>) 214 1827 2347 8993 1470 0159 0518	0.4 0.3 0.3 0.2 0.2 15 0.1 0.1		*****
Sim (2Ψ) ×Sin 19.196420 cions Number meters Thickness 0 20 <td>(Δ), Ic = c = 26 [Å] = 180 = - = - = - = -</td> <td>Sin(29) ×Cos 8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.3 0.5991 ± 0.3 0.0106 ± 0.3</td> <td>(<i>d</i>) 214 1827 2347 8993 1470 0159 0618 0090</td> <td>0.4 0.3 0.3 0.2 0.2 1s 0.1 0.1</td> <td>· · · · · · · · · · · · · · · · · · ·</td> <td>*****</td>	(Δ), Ic = c = 26 [Å] = 180 = - = - = - = -	Sin(29) ×Cos 8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.3 0.5991 ± 0.3 0.0106 ± 0.3	(<i>d</i>) 214 1827 2347 8993 1470 0159 0618 0090	0.4 0.3 0.3 0.2 0.2 1s 0.1 0.1	· · · · · · · · · · · · · · · · · · ·	*****
(2 ¹) × Sin (2 ¹) × Sin	$[\tilde{A}] = 180$ $[\tilde{A}] = 180$ = -1 = -1 = -1 = -1 = -1	Sin(29) ×Cos 8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.3 0.5991 ± 0.3 0.0106 ± 0.3 0.0006 ± 0.3	214 1827 2347 8993 1470 0159 0618 0090	0.4 0.3 0.3 0.2 0.2 1s 0.1 0.1 0.0		*****
meters Thickness $0 = E_0^{\infty}$ $0 = E_0^{\infty}$ $0 = E_0^{\infty}$ $0 = E_0^{\infty}$ $0 = E_0^{\infty}$ $0 = E_0^{\infty}$ $0 = E_1^{\infty}$ $0 = E_1^{\infty}$	$[\tilde{A}] = 180$ $[\tilde{A}] = 180$ = -1 = -1 = -1 = -1 = -1	Sin(29) ×Cos 8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.3 0.5991 ± 0.3 0.0106 ± 0.3 0.0006 ± 0.3 0.00	214 1827 2347 8993 1470 0159 0618 0090 0094 8814	0.4 0.3 0.3 0.2 0.2 15 0.1 0.1 0.1 0.0 0.0		*****
$\frac{\text{Sig}(2\Psi) \times \text{Sig}}{19.196420}$ $\frac{\text{ameters}}{100000000000000000000000000000000000$	$[\tilde{A}] = 180$ $[\tilde{A}] = 180$ = -1 = -1	Sin(29) ×Cos 8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.3 0.5991 ± 0.3 0.0106 ± 0.3 4.1684 ± 29.3 0.024 ± 0.3 0.024 ± 0.3 0.0006 ± 0.3 0.0000 ± 0.3 0.0006 ± 0.3 0.0000 ± 0.3 0.00000 ± 0.3 0.0000000 ± 0.3 0	214 1827 2347 8993 1470 0159 0618 0090 0094 8814 0540	0.4 0.3 0.3 0.2 0.2 0.2 0.2 0.1 0.1 0.1 0.0 0.0		*****
$\begin{array}{c} \text{Sig(29)} \times \text{Sig}\\ \text{Sig(29)} \times \text{Sig}\\ 19, 196420\\ \text{where}\\ \text{meters}\\ \text{total}\\ \text{meters}\\ met$	$[\tilde{A}] = 180$ $[\tilde{A}] = 180$ = -1 = -1	8.248 ± 42.3 0.6937 ± 2. 0.1145 ± 1.3 0.7744 ± 0.3 1.1582 ± 1. 2.8104 ± 0.4 0.5991 ± 0.4 0.0106 ± 0.4 4.1684 ± 29.3 3.8394 ± 2.4	214 1827 2347 8993 1470 0159 0618 0090 0094 8814 0540 0540	0.4 0.3 0.3 0.2 0.2 0.2 0.2 0.2 0.1 0.1 0.1 0.0 0.0 0.0 -0.1		*****

General Remarks

• The final model includes a rough overlayer on top that provides an excellent fit results.





χ²	minimization on Is, Ic								
Is	$Is = Sin(2\Psi) \times Sin(\Delta), Ic = Sin(2\Psi) \times Cos(\Delta)$								
X	= 0.071609								
Ite	erations Number = 3								
Pa	rameters								
1)	L1 Thickness [Å] = 49.738 ± 0.685								
2)	gaas-oxide εs = 2.3914 ± 0.0603								
3)	gaas-oxide $\omega t = 8.2930 \pm 0.2324$								
Co	rrelation matrix								
=1=	=2= =3=								
1.0	00 -0.913 -0.230								
	1.000 0.561								
	1.000								





0.5

5

4.5

Corre	elation	matri	x
=1=	=2=	=3=	=4=
1.000	-0.233	-0.020	-0.354
	1.000	0.049	0.032
		1.000	0.023
			1.000

9/2/0

2

2

2.5

3

3.5

Photon Energy (eV)

4

1.5



Explore the future



General Remarks

• The MgO exhibits a strong rough overlayer that must be taken into account in the final model.

9/2/08





0

5

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0.585

1.000

1.000

0.498

1.000

0.603

0.783

1.000

-0.399

-0.842

-0.821

-0.892

1.000

0.186

0.362

-0.361

0.253

0.036

1.000

1.95

9/2/

2

3

Photon Energy (eV)

Model without roughness



Fit Results

X2	minimization on Is,Ic
Is	<u>= Sin(2Ψ)×Sin(Δ),</u> Ic = Sin(2Ψ)×Cos(Δ)
χ^2	= 5.033619
Ite	erations Number = 4

Parameters

1)	L2 T	hickness	[Ă]	=	652.856	±	6.217
2)	BST	n∞		=	1.7459	±	0.0169
3)	BST	ωg		=	2.7834	±	0.1777
4)	BST	fj		=	0.1123	±	0.0201
5)	BST	ωj		=	5.1446	±	0.0613
6)	BST	Гј		=	0.9491	±	0.0560

Correlation matrix

=1=	=2=	=3=	=4=	=5=	=6=
1.000	-0.092	-0.307	-0.304	0.274	-0.016
	1.000	0.347	-0.166	-0.267	-0.699
		1.000	0.802	-0.894	-0.045
			1.000	-0.845	0.549
				1.000	-0.163
					1 000





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1.000

$Pt.spe \times \\$ it Results $Pt.spe \times \\$ it Results $Pt.spe \times \\$ $Pt.spe \times \\$ it results $Pt.spe \times \\$ $Pt.spe \times $ $Pt.spe \times \\$ $Pt.spe \times \\$ $Pt.spe \times \\$ $Pt.spe \times $	F 5679.2	FPZT.dsp	××
it Results $\frac{2 \text{ minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization minimization on IS, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization on Is, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization minimization minimization on IS, IC}}{1 = 12.861830}$ $\frac{2 \text{ minimization on IS, IC}}{1 = 12.861830}$ $2 \text{ minimiza$	5	Pt.spe	×
Results minimization on Is, IC controls, $rsin(d)$, $Ic = Sin(29) \times Cos(d)$ = 12.861830 rameters Li Thickness [Å] = 5679.248 ± 36.064 PZT by = 2.0560 ± 0.0114 PZT bg = 2.6188 ± 0.0378 PZT fj = 0.0866 ± 0.0071 PZT wj = 4.4906 ± 0.0139 PZT fj = 0.3590 ± 0.0169 PZT A, = -0.4376 ± 0.1278 PZT F, = 3.5172 ± 0.2746 relation matrix $\frac{relation matrix}{relation 0.000 - 0.082 - 0.082 - 0.053}$			
$\frac{\text{minimization on Is, Ic}}{\text{minimization on Is, Ic}} = \frac{12.861830}{12.861830} \times \frac{1}{10} \times $	Results		
$\frac{\text{minimization on Is, Ic}}{\text{Sin(37), Sin(3), Ic = Sin(29) \times Cos(4)}} = 12.861830$ = 12.861830 = 12.861830 = 12.861830 = 12.861830 = 12.861830 = 12.861830 = 12.861830 = 12.861830 = 2.0560 \pm 0.0114 P2T Eg = 2.6188 \pm 0.0378 P2T fj = 0.0866 \pm 0.0071 P2T g = 2.6188 \pm 0.0378 P2T fj = 0.0866 \pm 0.0071 P2T Mj = 4.4906 \pm 0.0139 P2T fj = 0.3590 \pm 0.0169 P2T A_0 = -0.4376 \pm 0.1278 P2T F_0 = 5.0960 \pm 0.4244 P2T A_0 = 3.5172 \pm 0.2746 rrelation matrix $\frac{\pi^2}{\pi^2} = \frac{\pi^3}{\pi^3} = \frac{\pi^4}{\pi^3} = \frac{\pi^5}{\pi^5} = \frac{\pi^6}{\pi^5} = \frac{\pi^7}{\pi^3}$			
$\frac{\text{Birlibrization on 13, 16}{\text{Firelation matrix}} = \frac{5}{5} = \frac{5}{6} = $	nininination on To To	I	
$\frac{rameters}{Li1 \text{ Thickness } [Å] = 5679.248 \pm 36.064 \\ P2T n^{\infty} = 2.0560 \pm 0.0114 \\ P2T Eg = 2.6188 \pm 0.0378 \\ P2T fj = 0.0866 \pm 0.0071 \\ P2T j = 0.0866 \pm 0.0071 \\ P2T j = 0.3590 \pm 0.0169 \\ P2T \Lambda_{0} = -0.4376 \pm 0.1278 \\ P2T \Lambda_{0} = 3.5172 \pm 0.2746 \\ \hline relation matrix \\ = \frac{2}{00} - 0.922 = 0.163 & 0.030 - 0.082 & 0.062 & 0.053 \\ = \frac{3}{00} - 0.922 & 0.163 & 0.030 - 0.082 & 0.062 & 0.053 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 - 0.082 & 0.062 & 0.053 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 - 0.082 & 0.062 & 0.053 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 - 0.082 & 0.062 & 0.053 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 - 0.082 & 0.062 & 0.053 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 & 0.082 & 0.063 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 & 0.082 & 0.063 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 & 0.082 & 0.062 & 0.053 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.163 & 0.030 & 0.082 & 0.063 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.030 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.040 & 0.040 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.040 & 0.040 & 0.040 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.040 & 0.040 & 0.040 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.040 & 0.040 & 0.040 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.040 & 0.040 & 0.040 \\ \hline matrix = \frac{1}{00} - 0.922 & 0.062 & 0.063 & 0.040 $	= 12.861830	л (29) ×Cos (Δ)	
$\begin{array}{rcl} \hline cameters \\ L1 \ Thickness \ [$] = 5679.248 \pm 36.064 \\ PZT \ m^{\circ} & = 2.0560 \pm 0.0114 \\ PZT \ Eg & = 2.6188 \pm 0.0378 \\ PZT \ Eg & = 2.6188 \pm 0.0071 \\ PZT \ Eg & = 2.6188 \pm 0.0071 \\ PZT \ J) & = 0.0866 \pm 0.0071 \\ PZT \ J) & = 0.3590 \pm 0.0169 \\ PZT \ A_{0} & = -0.4376 \pm 0.1278 \\ PZT \ A_{0} & = 3.5172 \pm 0.2746 \\ \hline \hline cameters \\ \hline crelation \ matrix \\ \hline cameters \\ \hline camete$	rations Number = 18		
$\begin{array}{rrr} PZT & n^{\infty} &= 2.0560 \pm 0.0114 \\ PZT & Eg &= 2.6188 \pm 0.0378 \\ PZT & fj &= 0.0866 \pm 0.0071 \\ PZT & uj &= 4.4906 \pm 0.0139 \\ PZT & rj &= 0.3590 \pm 0.0169 \\ PZT & rj &= 0.3590 \pm 0.0169 \\ PZT & r_{0} &= -0.4376 \pm 0.1278 \\ PZT & r_{0} &= 3.5172 \pm 0.2746 \\ \hline \hline rrelation matrix \\ \hline \begin{array}{c} =2e & =3e \\ =00 & -0.922 \\ 0.060 & 0.030 & -0.082 \\ 0.060 & 0.030 & -0.082 \\ 0.062 & 0.063 \\ 0.090 & 0.0169 \\ 0.050 \\$	rameters L1 Thickness [Å] = 5679.24	48 ± 36.064	0.3 1 1 1 1 1 1 1 1 A A
$\begin{array}{rrrr} pzr \ Eg & = \ 2.6188 \pm 0.0378 \\ pzr \ fj & = \ 0.0866 \pm 0.0071 \\ pzr \ \omega j & = \ 4.4906 \pm 0.0139 \\ pzr \ \Gamma j & = \ 0.3590 \pm 0.0169 \\ pzr \ \Lambda_0 & = \ -0.4376 \pm 0.1278 \\ pzr \ \Lambda_0 & = \ 3.5172 \pm 0.2746 \end{array}$	PZT noo = 2.03	560 ± 0.0114	
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	PZT Eg = 2.6	188 ± 0.0378 866 ± 0.0071	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	PZT wi = 4.49	906 ± 0.0139	-0.3 1 1 + 1 1 + 1 7 1 + 1 1
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	PZT Γj = 0.3	590 ± 0.0169	IIIIII_I_I_I_I_I_I_I_I
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	PZT Δ ₀ = -0.43	376 ± 0.1278	-0.6
$\begin{array}{cccccccccccccccccccccccccccccccccccc$		960 ± 0.4244	
orrelation matrix 1.5 2 2.5 3 3.5 = =2= =3= =4= =5= =6= =7= Photon Energy (eV) 000 -0.922 0.163 0.030 -0.082 0.040 0.040 000 -0.922 0.052 0.040 0.040 0.040	$PZT \Gamma_0 = 5.09$	172 + 0.2746	-0.9
= =2= =3= =4= =5= =6= =7= 000 -0.922 0.163 0.030 -0.082 0.062 0.053 -0.922 0.163 0.030 -0.082 0.040 -0.053	$PZT P_0 = 5.00$ $PZT A_0 = 3.5$		
000 -0.922 0.163 0.030 -0.082 0.062 0.053) PZT Γ_0 = 5.0:) PZT λ_0 = 3.5		1.5 2 2.5 3 3.5
	$PZT P_0 = 5.09$ $PZT A_0 = 3.5$ orrelation matrix = -22 = -32 = -42	=5= =6= =7=	1.5 2 2.5 3 3.5 Photon Energy (eV)
) PZT Γ_0 = 5.00) PZT A_0 = 3.5 correlation matrix 1= = 2= = 3= = 4= = 1 1.000 -0.922 0.163 0.030 -0 1.000 0.925 -0	=5= =6= =7= 0.082 0.062 0.053 0.269 0.040 -0.049 0.557 0.795 -0.058	1.5 2 2.5 3 3.5 Photon Energy (eV)
1.000 -0.461 0.959 -0.056) PZT Γ_0 = 5.09) PZT A_0 = 3.5 correlation matrix 1= =2= =3= =4= = 1.000 -0.922 0.163 0.030 -0 1.000 0.925 -0 1.000 -0.925 -0 1.000 -0 1.	=5= =6= =7= 0.082 0.062 0.053 0.269 0.040 -0.049 0.557 0.795 -0.058 0.461 0.959 -0.056 0.061 0.959 -0.056	1.5 2 2.5 3 3.5 Photon Energy (eV)
1.000 -0.461 0.959 -0.056 1.000 -0.378 0.084 1.000 -0.050	$\begin{array}{rrrr} P2T & \Gamma_0 & = & 5.09 \\ P2T & \lambda_0 & = & 3.5 \\ \hline \\ $	=5= =6= =7= 0.082 0.062 0.049 0.269 0.040 -0.049 0.557 0.795 -0.058 0.461 0.959 -0.056 1.000 -0.378 0.084 1.000 -0.050	1.5 2 2.5 3 3.5 Photon Energy (eV)

General Remarks

• Adding a rough overlayer on top of the film stack provides a significant decrease of the χ^2 . So the final model takes into account the roughness.

• The sample substrate was platinum. The material is listed in the application library of the software. Nevertheless, the optical properties (n,k) of metals vary strongly vs their composition. Thus as metals are strongly absorbing materials, the solution is to measure them and use the "spe" experimental file directly in the substrate.

Explore the future



Explore the future





General Remarks

 Adding a rough overlayer on top of the film stack decreases the χ^2 by a factor 2. Hence the final model takes into account the roughness.

- It is important noticing that the E_0 , E_1 , and E_2 values correspond to the energy at which the 3 extinction peaks (k) displayed on the graph "Optical constants of polymer" are maximum.

· We guess the presence of these absorption peaks directly on the raw (Is,Ic) data.

1.000

1.000

17/6

Applicat	ion Data She	25 29					232
Spectrosco	opic Ellipsometry	Data Sheets					25
	AlGal	N / AIN / Sap	phire S	Substrate)		
Model		-					
³ F	15.5 🗌 %	FAIGaN_T.ds 50	.00 % ×	Void.ref	50.00 %	××	
2 F	9032.2	FAIGaN_T.dsp			l	××	
1 F	4816.4	FAIN_T.dsp			l	××	
S	E	Al2O3-E_jy.ref				×	
s ^{LP}	0	Al2O3-O_jy.ref				×	
Fit Cond	litions	Fi	t Results				1
Spectral r	range: 0.6 - 4.0 eV						
 Step size: Angle of it 	: 0.01 ev ncidence: 70.07°				1 1	1	
Model Info	ormation:		0.2	s fit tit	1 104		
• Sapph	ire is an anisotropic m	aterial (in this		14.44.44.			
example • Multiqu	e N oriented).	ss varving from	化肥料				
7000 to	13 000 Å by step of 10	00 Å.	0 1 1 1		0.2	8	

· Fit parameters

- Thickness: Layers 1, 2 and 3
- Optical constants: AIN, AIGaN

Materials

• Al₂O₃-O, Al₂O₃-E, Void: Reference materials • AIN, AIGaN : Classical formula, respectively transparent and absorbing lorentz oscillator with $\mathcal{E}_{\scriptscriptstyle \infty}=1.0$

Optical constants of AlGaN



minimization on Is,Ic Jin(21) ~Sin(Δ), Ic = Sin(2Ψ) *Cos(Δ) 5.145501 Number = 8

2

Photon Energy (eV)

3

Parameters

-0.2

1

1)	L1 Thickness	[Ă]	=	4816.998	±	14.097
2)	L2 Thickness	[Ă]	=	9032.307	±	19.219
3)	L3 Thickness	[Ă]	=	15.545	±	1.687
4)	AlN_T ES		=	4.1243	±	0.0167
5)	AlN T wt		=	9.9398	±	0.2222
6)	AlGaN_T ES		=	4.6002	±	0.0157
7)	AlGaN Τ ωt		=	7.2200	±	0.0264
8)	$AlGaNT \Gamma_0$		=	0.1273	±	0.0081

Correlation matrix

=1=	=2=	=3=	=4=	=5=	=6=	=7=	=8=
1.000	0.231	0.035	-0.536	0.419	-0.489	-0.385	-0.250
	1.000	0.002	-0.500	-0.111	-0.868	0.158	-0.287
		1.000	-0.024	0.035	-0.044	-0.021	-0.162
			1.000	0.414	0.355	-0.328	0.152
				1.000	-0.224	-0.777	-0.224
					1.000	0.293	0.351
						1.000	0.199
							1.000

0.0

4

9/2/08

Explore the future



Model without roughness

2 F	9034.9		FAIGaN_T.dsp	×	×
¹ F	4812.4		FAIN_T.dsp	×	×
		Е	Al2O3-E_jy.ref	×	
s ^{LP}		0	Al2O3-O_jy.ref	×	

Fit Results



χ²	minimization o	n Is,I	C		
Is	= $Sin(2\Psi) \times Sin(\Delta)$,	, IC = .	Sin(29)	×Cos (Δ)	
χ^2	= 6.950035				
Ite	eracions wumber =	1			
Pa	rameters				
1)	L1 Thickness [Å]	= 4812	.400 ±	16.520	
21	L2 Thickness [Å]	= 9034	.900 +	22,433	
31	AIN T FS	= 4	.1257 +	0.019	7
41	AIN T OF	= 9	.8455 +	0.251	5
5	NIGON T PR	= 4	6057 +	0 0182	>
2,	AloaN T cs		.0057 1	0.010.	-
ь)	AIGAN_T wt	= 7	.2309 ±	0.0313	5
7)	AlGaN_T F.	= 0	.1391 ±	0.009	7
Co	rrelation matri	х			
=1=	=2= =3=	=4=	=5=	=6=	=7=
1.0	00 0.223 -0.539	0.414	-0.485	-0.381	-0.249
	1.000 -0.498	-0.111	-0.866	0.157	-0.288
	1.000	0.411	0.353	-0.324	0.154
		1.000	-0.224	-0.778	-0.221
			1.000	0.296	0.349
				1.000	0.207
					1.000

General Remarks

• Adding a rough overlayer on top of the film stack decreases the χ^2 from approximatively 7 to 5. Hence the final model takes into account the roughness.

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1.000

-0.031

1.000

-0.008 -0.320

1.000

-0.046

0.686

1.000

-0.340

-0.044

0.597

0.367

0.425 1.000

9/2/

1

2

Photon Energy (eV)

3

HORIBA



General Remarks

• Adding a rough overlayer on top of the film stack decreases strongly the χ^2 . Hence the final model takes into account the roughness.

9/2/09

Spectroscopic Ellipsometry Data Sheets

3x(SiO₂ / Ti) / Glass Substrate

Model

7 [F 1271.2	Sio2_wor.ref	×	×
⁶ [F 301.8	F Ti.dsp	×	×
5	F 323.9	Sio2_wor.ref	×	×
⁴ [F 305.1	F Ti.dsp	×	×
3	F 595.0	Sio2_wor.ref	×	×
2	F 71.5	F Ti.dsp	×	×
1	1000000.0	1737.dsp Backside reflection	x	×
S		void.dsp model	×	

Fit Conditions

- Spectral range: 1.2 6.5 eV
- Step size: 0.05 eV
- \bullet Angle of incidence: 70.0°
- Model Information: Backside reflections model is used as Glass is a transparent material.
- Fit parameters
 - Thickness: Layers 1, 2, 3, 4, 5, 6 and 7
 - Optical constants: Ti
- Materials
 - 1737, Void, SiO₂: Reference materials
 - Ti : Full classical formula including 3

absorbing lorentz oscillator + drude oscillator









Par	can	neters					
1)	L2	Thickness	[Ă]	=	71.521	±	21.070
2)	L3	Thickness	[Ă]	=	594.996	±	69.937
3)	L4	Thickness	[Ă]	=	305.074	±	39.898
4)	L5	Thickness	[Ă]	=	323.937	±	17.159
5)	L6	Thickness	[Ă]	=	301.774	±	20.320
6)	L7	Thickness	ΓĂΊ	=	1271.181	±	10.287
7)	Τi	500		=	4.8754	±	0.3816
8)	Τi	23		=	7.0492	±	0.4037
9)	Τi	ωt		=	6.4026	±	0.1069
10)	Τi	ωp		=	5.8581	±	0.7731
11)	Τi	Γ_0		=	4.0532	±	0.2911
12)	Τi	Γd		=	-4.6285	±	0.5853
13)	Τi	f ₁		=	0.3508	±	0.2443
14)	Τi	ω ₀₁		=	2.7640	±	0.1049
15)	Τi	Y1		=	1.7685	±	0.4375
16)	Τi	f_2		=	51.5694	±	10.7062
17)	Τi	ω ₀₂		=	0.9849	±	0.0790
18)	Τi	\vee_2		=	2.3151	±	0.2418

General Remarks

• Negative parameters have been found in the Titanium dispersion parameters. Nevertheless the optical constants found are physical.





Optical constants of YBaCuO

χ²	minimi	zatio	o n	n	Is,Ic		
Is	= Sin/2	\$) ×Si;	n(Δ),	, .	Ic = Sin(.	2 Y)	×Co.s (Δ)
χ^2	= 0.267	762					
Ite	erations	Numbe	er =	3			
Pa	rameter	3					
1)	L1 Thic	kness	[Ă]	=	997.761	±	7.757
2)	YBaCuO	500 200		=	1.1198	±	0.1461
3)	YBaCuO	23		=	3.2870	±	0.0487
4)	YBaCuO	ωt		=	9.9681	±	0.6498
5)	YBaCuO	ωp		=	2.3385	±	0.0172
61	YBaCuO	Г.		=	9.4044	+	1.3370

<i>~</i> ,	IDadao	+ Q		5.1011	÷	1.0010
7)	YBaCuO	Гd	=	0.4892	±	0.0241
8)	YBaCuO	f ₁	=	0.9990	±	0.0756
9)	YBaCuO	ω ₀₁	=	2.7010	±	0.0383
10	YBaCuO	Y ₁	=	1.9646	±	0.1178
11	YBaCuO	f_2	=	0.1399	±	0.0582
12	YBaCuO	$\omega_{0,2}$	=	3.7381	±	0.0408
13	YBaCuO	γ_2	=	1.4119	±	0.2653

Correlation matrix

=1=	=2=	=3=	=4=	=5=	=6=	
1.000	-0.108	0.532	0.216	-0.349	0.357	
	1.000	-0.546	-0.964	0.534	-0.879	
		1.000	0.731	-0.264	0.871	
			1.000	-0.497	0.969	
				1.000	-0.472	

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6)

7)

8)

9)

=1=

1.000

a-C

a-C

a-C

a-C

 Γ_1

 f_2

ω

 Γ_2

=2=

0.499

1.000

Correlation matrix

=3=

0.463

0.784

1.000

1.55

1.5

2

Explore the future

3

4

Photon Energy (eV)

03

0.25

6

5

 1.4684 ± 0.0448

 0.2054 ± 0.0103

 2.8885 ± 0.3476

=6=

-0.090

0.398

0.052

0.611

1.000

-0.514

 -0.8055 ± 0.1261

=5=

-0.141

-0.632

-0.657

-0.775

1.000

=

=

=4=

0.474

0.896

0.790

1.000





<u>x</u> ²	mi	ni	imi	zatio	n o	n	Is, Ic	100	- Coo (4)	
1.5			00	211	а(д),	<u> </u>	10 - 510(2)	r)	×соя(д)	
X	= 2		000	211		~				
Ite	Prau	-10	ms	Numbe	er =	Z				
-										
Pa	ram	et	er	3						
1)	L1	Tł	nic	kness	[Å]	=	1341.200	±	3.955	
2)	L2	Τł	nic	kness	ſĂſ	=	2111.801	±	2.764	
3)	TiO	02	в	n∞		=	2.0822	±	0.0202	
4)	TiO	02	в	ωg		=	3.1909	±	0.0430	
5)	TiO	02	в	fi		=	0.2555	±	0.0256	
6)	TiC	02	в	ωi		=	3.9848	±	0.0347	
7)	TiC	02	в	Гj		=	0.4769	±	0.0298	
8)	TiO	02	Т	n∞		=	1.9929	±	0.0170	1
9)	TiC)2	т	ωg		=	3.1925	±	0.0785	
10)	TiO	2	Т	fj		=	0.2522	±	0.0259	
11)	TiC)2	т	ωj		=	4.3735	±	0.0393	
12)	TiC)2	Т	Гј		=	0.6639	±	0.0286	

Optical constants of MgF₂

	Value
∞3	1.0000000
εs	1.8558800
ωt	-16.2837960

• In the physical formula of the Lorentz oscillator, the ω t parameter is squared so that the negative parameter found for ω t is good and provides physical optical constants (as shown on the right).



Model without gradient

² F 2104.7	MgF2.dsp	×	×
¹ F 1332.6	F TiO2.dsp	×	×
S	Csi_wor.ref	×	

Fit Results

χ²	mi	ni	mize	atio	n o	n	Is	, I	C			
Is	= (ĩin	12101	×Si1	α(Δ),	-	Ic	= .		?¥)	×Cos(Δ)	
X2	= 6	5.6	2696	3								
Ite	erau	.10	па М	umbe	er =	1:	1					
Pa	ram	et	ers									
1)	L1	Th	ickn	ess	[Ă]	=	13	32	.598	±	3.323	
2)	L2	Th	ickn	ess	[Ă]	=	21	04	.743	±	2.094	
3)	Ti	20	noo			=		2	.0673	±	0.0096	
4)	Ti	20	ωg			=		3	.2400) ±	0.0202	
5)	Ti	20	fj			=		0	.3527	' ±	0.0196	
6)	Ti	20	ωj			=		4	.1190) ±	0.0198	
7)	Ti	20	Гj			=		0	. 6887	' ±	0.0189	
Co	rre	la	tior	n ma	tri	x						
=1=	1	=2	2=	=3	=		4=		=.5=		=6=	=7=
1.0	00	-0.	.270	-0.	165	- 0).20)3	-0.1	55	-0.106	-0.450
		1.	. 000	ο.	315	- 0	0.49	50	0.6	46	-0.027	0.711
				1.	000	- 0).78	80	0.8	02	-0.845	0.350
						1	00	00	0.8	21	-0.605	0.231
									1.0	00	-0.633	0.704
											1.000	-0.138
												1.000



General Remarks

• The TiO₂ gradient improves strongly the fit results, hence the final model contains it.

 MgF₂ optical properties are not fitted as they were previously analyzed.

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-0.262

-0.981

0.988

1.000

1.000

0.378

1.000

-0.345

-0.996

1.000



Spectroscopic Ellipsometry Da	ta Sheets			3
4(SiO	₂ -TiO ₂) / GI	ass Substrate		
Model				
³ F 1118.1	Sio2_isa.ref		×	×
4 ² F 673.4	F TiO2.dsp		×	×
	Glass.dsp	Backside reflection	×	×
S	Void.ref	model	×	

Fit Results

Fit Conditions

- Spectral range: 430 850 nm
- Angle of incidence: 70°
- Model Information:
 - Backside reflection model

• The repeated layer function was used to repeat 4 times the TiO_2/SiO_2 layers. The results provide an average thickness of these 4 times repeated layers.

• It is worth noticing that the wavelength range is limited from 430 to 850 nm. Hence the TiO_2 dispersion used is a transparent Lorentz oscillator providing an extinction coefficient (k) equal to 0.

- Fit parameters
 - Thickness: Repeated layers 1 and 2
 - Optical constants: TiO₂
- Materials
 - Glass, Void: Reference materials
 - TiO_2 : Classical formula, Transparent Lorentz oscillator with $\mathcal{E}_{\infty}=1.0$



Optical constants of TiO₂



χ ² minimization on Is,Ic,Ic'
$Is = Sin(2\Psi) \times Sin(\Delta), Ic = Sin(2\Psi) \times Cos(\Delta)$
$IC' = Cos(2\Psi)$
$\chi^2 = 2.661902$
Iterations Number = 4
Parameters

1)	L2 Thickness []	i] =	673.362	±	1.451
2)	L3 Thickness [Å	i] =	1118.061	±	1.227
3)	TiO2 εs	=	4.7116	±	0.0100
4)	TiO2 ωt	=	-6.5152	±	0.0837
Co	rrelation matr	ix			

=1=	=2=	=3=	=4=
1.000	-0.639	-0.548	-0.283
	1.000	0.312	-0.225
		1.000	-0.447
			1.000

• Negative ωt parameter is possible as in Lorentz oscillator formula ωt is squared.

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Spectroscopic Ellipsometry	Data Sheets			34
Y ₃ Fe	e₅O ₁₂ / SiO₂ / c-Si Su	bstrate		
Model				
³ F 90.2 🗌 %	FY3Fe5O12.d 50.00 % ×	Void_asp.re 50.00 % ×	×	
² F 11256.1	FY3Fe5012.dsp	×	×	
¹ F 1568.7	Sio2_isa.ref	×	×	
S	C-si_isa.ref	×		
Fit Conditions	Fit Results			

- Spectral range: 0.75 2.50 eV
- Step size: 0.005 eV Angle of incidence: 70°
- Model Information:
 - The fit was performed on a restricted spectral range due to the strong absorption of Y₃Fe₅O₁₂ material above 2.5 eV. More investigation is necessary to extend the spectral range of fitting, eg. Testing others dispersion formulae.
- Fit parameters
 - Thickness: Layers 1, 2 and 3
 - Optical constants: Y₃Fe₅O₁₂
- Materials
 - c-Si, SiO₂: Reference materials
 - Y₃Fe₅O₁₂: Classical formula, Absorbing Lorentz + Drude oscillators





Model without roughness

2 F 11335.8	FY3Fe5O12.dsp	×	×
¹ F 1733.5	Sio2_isa.ref	×	×
S	C-si_isa.ref	×	

Fit Results

x	² m	inimiza	atic	o n	n	Is,1	C			
I	5 =	Sin (24)	*Si	a (Δ) .	, .	IC =	Sin(29) ×	Cos(A)	
X	2 =	34.2890	70							
1		tions N	umbe	er =	12	2				
P	ara	meters								
1) L1	Thickn	ess	[Å]	=	173	3.484	±	70.90	7
2) L2	Thickn	ess	[Å]	=	1133	5.840	±	86.03	6
3) Y3	Fe5012	200		=		3.4510	±	0.09	50
4) Y3	Fe5012	23		=		4.5885	±	0.05	50
5) Y3	Fe5012	wt		=		3.3257	±	0.06	00
6) Y3	Fe5012	ωp		-		0.3345	±	0.02	24
7) Y3	Fe5012	Γ.		=		0.3455	±	0.00	90
8) Y3	Fe5012	Гđ		=		0.2432	±	0.05	17
с	orr	elatio	n ma	tri	x					
=)	L=	=2=	= 3	=		4=	=5=		=6=	=7=
1.	000	-0.258	-0.	113	0	.023	0.265		0.556	-0.298
		1.000	-0.	185	-0	.939	-0.241		-0.324	-0.076
			1.	000	0	1.440	-0.879		0.597	-0.139
					1	000	-0.014		0.391	0.138
							1.000		-0.322	0.244
									1.000	-0.177
										1.000



General Remarks

- Adding a surface roughness improves the χ^2 by a factor 3.



=1=

1.000

=2=

-0.986

1.000

=3=

0.474

-0.336 1.000

-	-	-		-
		•	•)	
			•)	
		_		

1.68

1.66

1.64

9/2/

1

2

3

Photon Energy (eV)

4



Anodised Porous AI / Al₂O₃ / AI Substrate

Model



Fit Results

Fit Conditions

- Spectral range: 0.6 6.0 eV
- Step size: 0.05 eV
- Angle of incidence: 70°
- Model Information:

• The substrate is the measurement file of the Aluminium substrate itself. Cleaning the substrate is required as Aluminium surface always exhibits a native Al₂O₃ film.

• A remaining Al₂O₃ interface film was found (from previous results explaining the non-fitting of this film).

• A merge acquisition is necessary for metals.

• The conditions of elaboration of anodised Aluminium film introduces uniformly distributed pores on the surface. This is modeling by using an anisotropic film (N oriented as pores are perpendicular to the surface).

• Fit parameters

- Thickness: Layer 2
- Optical constants: Anodised Al

Materials

- Al₂O₃: Reference materials
- Anodised AI: Classical formula, Transparent

Lorentz oscillator with $\mathcal{E}_{\infty} = 1.0$

Optical constants of anisotropic anodised AI





X	minimization on	Is, Ic
10	- Sin(27, Sin(Δ),	$IC = Sin(2\Psi) \times Cos(\Delta)$
χ²	= 0.499575	
Ite	i = 1	1

Parameters

1)	L2 Thicks	ness [Å]	1			-	936.757	±	1.881
2)	anodised	porous	A1	0	83	-	1.7511	±	0.0042
3)	anodised	porous	A1	0	ωt	=	13.6688	±	0.3535
4)	anodised	porous	A1	E	23	=	1.8700	±	0.0073
5)	anodised	porous	A1	Ε	ωt	-	13.2305	±	0.2165

Correlation matrix

-1-	*2*	#3#	=4=	*5*	
1.000	-0.839	-0.356	-0.721	0.186	
	1.000	0.785	0.746	-0.189	
		1.000	0.454	-0.219	
			1.000	0.439	
				1.000	

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Model without anisotropy

2	F	963.5	Fanodised porous Al.dsp	××
1		70.9	Al2O3_pal.ref	××
S			nk Al.clc	×

Fit Results



General Remarks

• The anisotropic layer contained in the final model results a χ^2 =0.5 compared to 7.2 for a simple layer.





Para 1) L:	meters 1 Thickn	ess [Å]	=	954.	342	±	8.093
2) L2	2 Thickn	ess [Å]	=	198.	460	±	7.619
3) L2	2 % ZrO2	.dsp	=	91.	66	±	0.30
4) Zi	cO2 cs		=	з.	4848	±	0.0038
5) Zi	c02 ωt		=	8.	2982	±	0.0457
Corr	elation	n matri	x				
=1=	=2=	=3=	=	4=	=5=		
1.000	-0.993	-0.841	-0	.001	0.4	73	
	1.000	0.829	-0	.028	-0.4	47	
		1.000	0	.100	-0.1	78	
			1	.000	0.6	18	

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1.000

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General Remarks

• The anisotropic layer contained in the final model results a χ^2 =0.5 compared to 7.2 for a simple layer.

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PolySilicon / a-Si / SiO₂ / Glass Substrate

Model

5	F	35.2	Sio2_uv.ref		×	×
4	F	433.0 F %	Fa-: 9.54 % ×	c-{ 85.97 % × 🛛 vo 4.49 %	×	×
3	F	135.2	F a-Si.dsp		×	×
2		3000.0	Sio2_uv.ref		×	×
1		700000.0	1737.dsp	Backside reflection	×	×
s			void.dsp	model	×	

Fit Conditions

- Spectral range: 1.5 6.5 eV
- Step size: 0.1 eV
- Angle of incidence: 70°
- Model Information:
 - Backside reflection model
 - Polysilicon material is represented by a mixture of a-Si + c-Si + void
 - Native oxide on top surface of the sample
 - SiO₂ layer is not fitted as it comes from
 - previous results.
- Fit parameters
 - Thickness: Layers 1, 2, 3, 4 and 5
 - Optical constants: a-Si
 - Composition of layer 4: % a-Si + % c-Si + % void
- Materials
 - Void, Glass, SiO₂: Reference materials
 - a-Si: Tauc Lorentz formula

Optical constants of a-Si



Fit Results



x²	min	imize	tion o	n	Is, I	c			
7.00		1200	×Sin(Δ)	,	Ic =	Sin(.	29,) *Cos (4)	
x	- 0.4	85449	2						
	COP 1	one li	mmer =	9					
Par	ame	ters							
1)	L3 T	hickn	ess [Å]	-	135.	230	±	22.437	
2)	L4 T	hickn	ess [Å]	=	433.	020	±	17.727	
3)	LS T	hickn	ess [Å]	-	35.	239	±	1.667	
4)	L4 %	a-Si	.dsp	-	9.	54	±	1.71	
5)	L4 %	c-Si	Jy.clc		85.	97	±	2.22	
6)	a-Si	Eg		-	1.	1959	±	0.0715	
7)	a-Si	200		-	-3.	0059	±	1.7527	
8)	a-Si	A		-	184.	4456	±	37.4818	
9)	a-Si	Eo		=	4.	4428	±	0.3641	
10)	a-Si	C		-	4.	3307	±	0.9595	
Cor	rela	ation	matri	х					
=1=		*2=	=3=		-4=	=5=	i	=6=	=7=
1,00	00 -0	0,733	0.586	-0	0.110	0.5	20	0.358	0,29
		1.000	-0.260	- 5	0.314	-0.3	80	-0.223	0.02
			1.000	-0	1.314	0.6	98	0.341	-0.14
						-0.8	00	-0.246	0.50
						1.0	00	0.351	-0.12
								1.000	-0.22
									1.00

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Optical constants of p-Si of layer 4* Optical Constants of p-Si 6 4.0 5 3.0 n ⁴ k 2.0 3 1.0 2 1 3 4 5 Photon Energy (eV) 2 6

*In deltapsi2, build a new spectroscopic model, and put in the substrate the results found for layer 2, eg 9.54% c-Si + 85.97% a-Si + 4.49% void. Then under modeling description, click on edit, and in the modeling conditions tab, choose "User defined spectral range" and put 1.5-6.5 eV. Then, click on simulation, and in the results display area, select the (n,k) axis.

Appendix ADetailed Description of DispersionFormulae and Settings

Next pages detail each Dispersion Model available from DeltaPsi2 Software.



Appendix A1: Lorentz Dispersion Model

Spectroscopic Ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected polarized light in order to characterize thin film optical functions and other properties. The measured SE data is described by a model where layers refer to a given material. The model uses mathematical relations called dispersion formulas that help to evaluate the material's optical properties by adjusting specific fit parameters. This technical note deals with the Drude dispersion formula.

Note that the technical notes «Classical dispersion model» and «Lorentz dispersion model» are complementary to

Theoretical model

Drude's model (1900) is based on the kinetic theory of electrons in a metal which assumes that the material has motionless positive ions and a non-interacting electron gas. This simple model uses classical mechanical theory of free electron. It was constructed in order to explain the transport properties of conduction electrons in metals (due to intra-band transitions in a quantum-mechanical interpretation), conductive oxides and heavily doped semiconductors.

Since the conduction electrons are considered to be free, Drude oscillator is an extension of the single Lorentz oscillator to a case where the restoring force and the resonance frequency are null ($\Gamma_0=0$, $\omega_t=0$) - see «Lorentz dispersion model» technical note.

The equation of motion is:

$$m \cdot \frac{d\vec{v}}{dt} + m \cdot \Gamma_d \cdot \vec{v} = -e \cdot \vec{E}$$
(1)

where:

- m (dv/dt) is the acceleration force; m is the mass of the oscillator (here the electron) and v is the speed of the gas of electron;
- m Γ_d **v** is the friction force;
- the term -e.**E** is the electric force; it contains the term $E=E_0 e^{i\omega t}$ which is the electric field of pulsation ω and -e which is the electric charge.

The solution of the previous equation is given by the following expression for oscillation amplitude:

$$\vec{v}(\omega) = \vec{v}_0 \cdot e^{i\omega t} = \frac{-e}{m} \cdot \frac{E_0}{i\omega + \Gamma_d}$$
(2)

The conduction density of current J_c corresponding to the movement of the N electrons per unit volume is:

$$\vec{J}_{c}(\omega) = -N e \vec{v} = \frac{N e^{2}}{m \cdot (\Gamma_{d} + i \omega)} \vec{E}$$
(3)

The displacement density of current of vacuum is expressed by:

$$\vec{J}_{d} = \frac{\partial \vec{D}}{\partial t} = i\,\omega\,\varepsilon_{0}\,\vec{E}\,\,(4)$$

where **D** is the electric displacement of vacuum: $D = \varepsilon_0 E$.

The total density of current \mathbf{J} is given by:

$$\vec{J}(\omega) = \vec{J}_c + \vec{J}_d = \left[\frac{N \cdot e^2}{m \cdot (\Gamma_d + i\,\omega)} + i\,\omega\,\varepsilon_0\right]\vec{E} \quad (5)$$

Assuming that the plasma can also be characterized by identifying the total density of current to a global displacement current **Dtot** = $\varepsilon_0 \varepsilon_r \mathbf{E}$. A new expression for the total density of current is then given by:

$$\vec{I}(\omega) = \frac{\partial D_{tot}}{\partial t} = i\,\omega\,\varepsilon_0\,\,\widetilde{\varepsilon}\,\,\vec{E}\,\,(6)$$

The identification of the two last formulations of the total density of current ${\bf J}$ yields:

$$\vec{J}(\omega) = \left[\frac{N \cdot e^2}{m \cdot (\Gamma_d + i\,\omega)} + i\,\omega\,\varepsilon_0\right] \cdot \vec{E} = i\,\omega\,\varepsilon_0\,\widetilde{\varepsilon}(\omega) \cdot \vec{E} \quad (7)$$

By simplifying that last expression, it is possible to deduce Drude's dielectric function $\varepsilon(\omega)$ given by the following equation:

$$\widetilde{\varepsilon}(\omega) = 1 - \frac{N e^2}{m \varepsilon_0} \cdot \frac{1}{(\omega^2 - i \Gamma_d \omega)} = 1 - \frac{\omega_p^2}{-\omega^2 + i \cdot \Gamma_d \cdot \omega}$$
(8)

Note that for $\omega \to \infty$, $\varepsilon(\infty) \to 1$ then it is possible to rewrite the dielectric function as:

$$\widetilde{\varepsilon}(\omega) = \varepsilon(\infty) - \frac{\omega_p^2}{-\omega^2 + i \cdot \Gamma_d \cdot \omega}$$
(9)

like in DP2 software, where $\varepsilon(\infty)$ is the high-frequency contribution that is usually superior or equal to one.

The real $\epsilon_1(\omega)$ and imaginary $\epsilon_2(\omega)$ parts of the dielectric function are given by:

$$\varepsilon_1(\omega) = 1(\varepsilon(\infty)) - \frac{\omega_p^2}{\omega^2 + \Gamma^2}$$
 (10.1)

and

$$\varepsilon_2(\omega) = \frac{\omega_p^2 \cdot \Gamma}{\omega \cdot (\omega^2 + \Gamma^2)} \quad (10.2)$$

Behaviour of Drude dielectric function

- If $\omega < \omega_p$ then the real part of ε is negative: any electrical field cannot penetrate the metal that is totally reflective. The optical constants of the material are complex.
- If $\omega = \omega_p$ then the real part of the dielectric function is zero. This means that all electrons oscillate in phase throughout the material propagation length.
- If $\omega > \omega_p$ the reflectivity decreases and the metal becomes transparent. The refractive index of the material is almost real.
- When $\omega^2 >> \Gamma_d^2$ then $\varepsilon_r(\omega) = 1 \cdot (\omega_p^2 / \omega^2)$. The imaginary part of the dielectric function can be approximated by $\varepsilon_i(\omega) = (\omega_p^2 \Gamma / \omega^2)$ and there is no absorption at the high frequency limit.

The parameters of the dispersion formula

2 parameters are used in the Drude's model. Both ω_p and Γ_d influence the real and imaginary parts of the dielectric function.

• ω_p is the plasma frequency. It corresponds to the photon energy position where $\varepsilon_r(\omega)$ is approximately zero. As ω_p increases the amplitude of $\varepsilon_r(\omega)$ and $\varepsilon_i(\omega)$ increases too.

(I) —	N ⋅ e ²	
ω ^b _ 1	m·ε₀	

• Γ_d (in eV) is the collision frequency. Generally, for metals $0 < \Gamma_d < 4$. As Γ_d increases the broadening of the absorption tail increases too.

Limitation of the model

The Drude oscillator describes well the optical properties of metals but does not take into account the notion of optical energy band gap E_g semiconductors and the quantum effects.

Parameter setup

Note that:

- 1 The Drude's dielectric function is available in the Classical dispersion formula in the DeltaPsi2 software.
- 2 The sign \propto before a given parameter means that either the amplitude or the broadening of the peak is linked to that parameter.

Drude's model predicts monotonous decrease of $\epsilon_r(\omega)$ for decreasing photon energy. It is also characterized by monotonous increase of $\epsilon_i(\omega)$ (absorption tail) with decreasing photon energy towards the NIR region.



Starting values of a classical Drude function

Application to materials

This model fits well the optical properties of metallic samples and heavily doped semiconductors. The spectral range used for fitting depends on the material as shown in the array below.

The asterisk * refers to parameters that are negative and thus do not have any physical meaning but represent good starting values to perform the fit on the material. S.R. means Spectral Range.

> Materials Γ_{d} S. R. (eV) 8.8 $\omega_{\rm p}$ AI 1.0 15.702 0.696 1.5 - 6.5 3.694 12.175 5.799 0.75 - 4.75 Co -1.366 19.972 1.5 - 5 Мо 8.357 1.0 14.270 4.020 1.5 - 5 NiCr NiFe 1.0 14.790 4.780 1.5 - 5 Ti 1.0 10.9 2.53 1.0 - 6.0

References

- 1) J. M. Cazaux, Introduction a la physique du Solide, Masson (1996)
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Appendix A2: Drude Dispersion Model

Spectroscopic Ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected polarized light in order to characterize thin film optical functions and other properties. The measured SE data is described by a model where layers refer to a given material. The model uses mathematical relations called dispersion formulas that help to evaluate the material's optical properties by adjusting specific fit parameters. This technical note deals with the Drude dispersion formula.

Note that the technical notes «Classical dispersion model» and «Lorentz dispersion model» are complementary to this one.

Theoretical model

Drude's model (1900) is based on the kinetic theory of electrons in a metal which assumes that the material has motionless positive ions and a non-interacting electron gas. This simple model uses classical mechanical theory of free electron. It was constructed in order to explain the transport properties of conduction electrons in metals (due to intra-band transitions in a quantum-mechanical interpretation), conductive oxides and heavily doped semiconductors.

Since the conduction electrons are considered to be free, Drude oscillator is an extension of the single Lorentz oscillator to a case where the restoring force and the resonance frequency are null ($\Gamma_0=0$, $\omega_t=0$) - see «Lorentz dispersion model» technical note.

The equation of motion is:

$$m \cdot \frac{d\vec{v}}{dt} + m \cdot \Gamma_d \cdot \vec{v} = -e \cdot \vec{E} \ (1)$$

where:

- m (dv/dt) is the acceleration force; m is the mass of the oscillator (here the electron) and v is the speed of the gas of electron;
- m Γ_d **v** is the friction force;
- the term -e.E is the electric force; it contains the term $E=E_0 e^{i\omega t}$ which is the electric field of pulsation ω and -e which is the electric charge.

The solution of the previous equation is given by the following expression for oscillation amplitude:

$$\vec{v}(\omega) = \vec{v}_0 \cdot e^{i\omega t} = \frac{-e}{m} \cdot \frac{\vec{E}_0}{i\omega + \Gamma_d}$$
(2)

The conduction density of current J_c corresponding to the movement of the N electrons per unit volume is:

$$\vec{J}_{c}(\omega) = -N e \vec{v} = \frac{N e^{2}}{m \cdot (\Gamma_{d} + i \omega)} \vec{E}$$
(3)

The displacement density of current of vacuum is expressed by:

$$\vec{J}_{d} = \frac{\partial \vec{D}}{\partial t} = i \,\omega \,\varepsilon_{0} \,\vec{E} \,(4)$$

where **D** is the electric displacement of vacuum: $\mathbf{D} = \varepsilon_0 \mathbf{E}$.

The total density of current **J** is given by:

$$\vec{J}(\omega) = \vec{J}_c + \vec{J}_d = \left[\frac{N \cdot e^2}{m \cdot (\Gamma_d + i\,\omega)} + i\,\omega\,\varepsilon_0\right]\vec{E} \quad (5)$$

Assuming that the plasma can also be characterized by identifying the total density of current to a global displacement current **Dtot** = $\varepsilon_0 \varepsilon_r \mathbf{E}$. A new expression for the total density of current is then given by:

$$\vec{J}(\omega) = \frac{\partial D_{tot}}{\partial t} = i \,\omega \,\varepsilon_0 \,\widetilde{\varepsilon} \,\vec{E} \,(6)$$

The identification of the two last formulations of the total density of current ${\bf J}$ yields:

$$\vec{J}(\omega) = \left[\frac{N \cdot e^2}{m \cdot (\Gamma_d + i\,\omega)} + i\,\omega\,\varepsilon_0\right] \cdot \vec{E} = i\,\omega\,\varepsilon_0\,\widetilde{\varepsilon}(\omega) \cdot \vec{E} \quad (7)$$

By simplifying that last expression, it is possible to deduce Drude's dielectric function $\varepsilon(\omega)$ given by the following equation:

$$\widetilde{\varepsilon}(\omega) = 1 - \frac{N e^2}{m \varepsilon_0} \cdot \frac{1}{(\omega^2 - i \Gamma_d \omega)} = 1 - \frac{\omega_p^2}{-\omega^2 + i \cdot \Gamma_d \cdot \omega}$$
(8)

Note that for $\omega \to \infty$, $\varepsilon(\infty) \to 1$ then it is possible to rewrite the dielectric function as:

$$\widetilde{\varepsilon}(\omega) = \varepsilon(\infty) - \frac{\omega_p^2}{-\omega^2 + i \cdot \Gamma_d \cdot \omega}$$
(9)

like in DP2 software, where $\varepsilon(\infty)$ is the high-frequency contribution that is usually superior or equal to one.

The real $\varepsilon_1(\omega)$ and imaginary $\varepsilon_2(\omega)$ parts of the dielectric function are given by:

$$\varepsilon_1(\omega) = 1(\varepsilon(\infty)) - \frac{\omega_p^2}{\omega^2 + \Gamma^2}$$
 (10.1)

and

$$\varepsilon_2(\omega) = \frac{\omega_p^2 \cdot \Gamma}{\omega \cdot (\omega^2 + \Gamma^2)} \quad (10.2)$$

Behaviour of Drude dielectric function

- If $\omega < \omega_p$ then the real part of ε is negative: any electrical field cannot penetrate the metal that is totally reflective. The optical constants of the material are complex.
- If $\omega = \omega_p$ then the real part of the dielectric function is zero. This means that all electrons oscillate in phase throughout the material propagation length.
- If $\omega > \omega_p$ the reflectivity decreases and the metal becomes transparent. The refractive index of the material is almost real.
- When $\omega^2 >> \Gamma_d^2$ then $\varepsilon_r(\omega) = 1 (\omega_p^2 / \omega^2)$. The imaginary part of the dielectric function can be approximated by $\varepsilon_i(\omega) = (\omega_p^2 \Gamma/\omega^2)$ and there is no absorption at the high frequency limit.

The parameters of the dispersion formula

2 parameters are used in the Drude's model. Both $\omega_{\rm p}$ and $\Gamma_{\rm d}$ influence the real and imaginary parts of the dielectric function.

> • $\omega_{\rm p}$ is the plasma frequency. It corresponds to the photon energy position where $\varepsilon_r(\omega)$ is approximately zero. As ω_p increases the amplitude of $\epsilon_r(\omega)$ and $\epsilon_i(\omega)$ increases too.



• Γ_d (in eV) is the collision frequency. Generally, for metals $0 < \Gamma_d < 4$. As Γ_d increases the broadening of the absorption tail increases too.

Limitation of the model

The Drude oscillator describes well the optical properties of metals but does not take into account the notion of optical energy band gap E_a semiconductors and the quantum effects.

Parameter setup

Note that:

- 1 The Drude's dielectric function is available in the Classical dispersion formula in the DeltaPsi2 software.
- 2 The sign \propto before a given parameter means that either the amplitude or the broadening of the peak is linked to that parameter.

Drude's model predicts monotonous decrease of $\varepsilon_r(\omega)$ for decreasing photon energy. It is also characterized by mo-

notonous increase of $\varepsilon_i(\omega)$ (absorption tail) with decreasing photon energy towards the NIR region.



Starting values of a classical Drude function

Application to materials

This model fits well the optical properties of metallic samples and heavily doped semiconductors. The spectral range used for fitting depends on the material as shown in the array below.

The asterisk * refers to parameters that are negative and thus do not have any physical meaning but represent good starting values to perform the fit on the material. S.R. means Spectral Range.

Materials	Q 3	ω_{p}	Γ _d	S. R. (eV)
AI	1.0	15.702	0.696	1.5 - 6.5
Со	3.694	12.175	5.799	0.75 - 4.75
Мо	-1.366*	19.972	8.357	1.5 - 5
NiCr	1.0	14.270	4.020	1.5 - 5
NiFe	1.0	14.790	4.780	1.5 - 5
Ti	1.0	10.9	2.53	1.0 - 6.0

References

- J. M. Cazaux, Introduction a la physique du Solide, Masson 1) (1996)
- 2) H. M. Rosenberg, The Solid State, Oxford University Press (1988)
- 3) F. Wooten, Optical Properties of Solids, Academic Press (1972) 4)
- E. Hecht, Optics, Chap. 3, Hardcover (2001) www.ece.tamu.edu/People/bios/hemmer_files/15-dispersion.ppt 5)

Appendix A3: Classical Dispersion Model

Spectroscopic ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected and polarized light in order to characterize thin film optical functions and other properties. The measured data are used to describe a model where each layer refers to a given material. The model uses mathematical relations called dispersion formulae that help to evaluate the thickness and optical properties of the material by adjusting specific fit parameters.

This application note deals with the classical dispersion formula. Note that more detailed explanation are given in the technical notes: Drude's and Lorentz dispersion model.

Theoretical model

The classical dispersion model is based on the sum of the single and double Lorentz, and Drude oscillators.

$$\widetilde{\epsilon}(\omega) = \varepsilon_{\infty} + \frac{\left(\varepsilon_{s} - \varepsilon_{\infty}\right) \cdot \omega_{t}^{2}}{\omega_{t}^{2} - \omega^{2} + i \cdot \Gamma_{0} \cdot \omega} + \frac{\omega_{p}^{2}}{-\omega^{2} + i \cdot \Gamma_{d} \cdot \omega} + \sum_{j=1}^{2} \frac{f_{j} \cdot \omega_{0j}^{2}}{\omega_{0j}^{2} - \omega^{2} + i \cdot \gamma_{j} \cdot \omega}$$

• Lorentz dispersion model

The Lorentz classical theory (1878) is based on the classical theory of interaction between light and matter and is used to describe frequency dependent polarization due to bound charge.

The Lorentzian dispersion formula comes from the solution of the equation of an electron bound to a nucleus driven by an oscillating electric field E. The response is equivalent to the classical mass on a spring which has damping and an external driving force. It generates damped harmonic oscillations.

In the DeltaPsi2 software the classical dispersion formula contains three oscillators.

$$\widetilde{\varepsilon}(\omega) = \varepsilon_{\omega} + \frac{(\varepsilon_s - \varepsilon_{\omega}) \cdot \omega_t^2}{\omega_t^2 - \omega^2 + i \cdot \Gamma_0 \cdot \omega} + \sum_{j=1}^2 \frac{f_j \cdot \omega_{0j}^2}{\omega_{oj}^2 - \omega^2 + i \cdot \gamma_j \cdot \omega}$$



Parameters describing the real part of the dielectric function

- The constant ε_{∞} is the high frequency dielectric constant. Generally, $\varepsilon_{\infty} = 1$ but can be greater than 1 if oscillators in higher enegies exist and are not taken into account.
- The constant $\varepsilon_s \ (\varepsilon_s > \varepsilon_{\infty})$ gives the value of the static dielectric function at a zero frequency. The difference $\varepsilon_s \varepsilon_{\infty}$ represents the strength of the single Lorentz oscillator. The larger it is then the smaller the width Γ_0 of the peak of the single Lorentz oscillator.

Parameters describing the imaginary part of the single Lorentz oscillator dielectric function

- ω_t (in eV) is the resonant frequency of the oscillator whose energy corresponds to the absorption peak. When ω_t increases then the peak is shifted to higher photon energies. Generally, 1 ≤ ω_t ≤ 20.
- Γ_0 (in eV) is the broadening of each oscillator also known as the damping factor. The damping effect is due to the absorption process involving transitions between two states. On a graphic representing $\epsilon_i(\omega)$, Γ_0 is generally equal to the Full Width At Half Maximum (FWHM) of the peak. As Γ_0 increases the width of the peak increases, but the amplitude of that decreases. Generally, $0 \le \Gamma_0 \le$ 10.

Parameters describing the imaginary part of the multiple Lorentz oscillators dielectric function

- f_j (j = 1, 2 ... N) term is the oscillator strength present in the expression of the multiple Lorentz oscillator. As f_j increases then the peak amplitude increases, but the width of the peak γ_j decreases. Generally, $0 \le f_i \le 10$.
- ω_{0j} (in eV) ($j = 1, 2 \dots N$) is the resonant (peak) energy of an oscillator for a collection of several Lorentzian oscillators. It is similar to ω_t . Generally, $1 \le \omega_{0j} \le 8$.
- γ_j (in eV) (j = 1, 2 ... N) parameter is the broadening parameter corresponding to the peak energy of each oscillator. It behaves like Γ_0 . Generally, $0 \le \gamma_i \le 10$.



• Drude dispersion model

Drude's model (1900) is based on the kinetic theory of electrons in a metal which assumes that the material has motionless positive ions and a non-interacting electron gas. This simple model uses classical mechanical theory of free electron. It was constructed in order to explain the transport properties of conduction electrons in metals (due to intra-band transitions in a quantum-mechanical interpretation), conductive oxides and heavily doped semiconductors.

Since the conduction electrons are considered to be free, Drude oscillator is an extension of the single Lorentz oscillator to a case where the restoring force and the resonance frequency are null ($\Gamma_0=0$, $\omega_t=0$).

In the DeltaPsi2 software the classical dispersion formula contains the Drude model expressed as below:

$$\widetilde{\varepsilon}(\omega) = \varepsilon_{\infty} + \frac{\omega_{p}^{2}}{-\omega^{2} + j \cdot \Gamma \cdot \cdot \omega}$$



• ω_p is the plasma frequency corresponding to the photon energy position where $\varepsilon_r(\omega)$ is approximately zero. When ω_p increases then the amplitude of the $\varepsilon_r(\omega)$ and $\varepsilon_i(\omega)$ increases too.



• Γ_d (in eV) is the collision frequency. As Γ_d increases the broadening of the absorption tail increases too. Generally, for metals $0.4 < \Gamma_d < 4$.

Application to Materials

The Lorentz oscillator model works well for insulators and semiconductors above the band gap.

The Drude model describes well the optical properties of metals but does not take into account the notion of optical band gap energy $\rm E_q$.

The combination of both is often adequate when the material is a little conductive and has a metallic character (like conducting oxides ITO, RuO_2).

The graph above typically shows the influence of Drude's function characterized by an increasing absorption (imaginary part of the dielectric function) toward the IR region.



Dielectric function of BiSrCaCuO

List of materials following Drude and Lorentzian dispersion models

The asterisk ***** refers to parameters that are negative and thus do not have any physical meaning but represent good starting values to perform the fit on the material.

Materials	e ko	٤s	ω _t	ω _p	Г₀	Γd
Ag	-2.875	3.401	9.6310	8.707	1.783	0.073
A	0.959	16.787	1.637	13.320	0.705	0.119
Cr	3.9	4	0	20.000	0	100.000
Cu	1	1.65	1.80	4.00	3.00	0.85
IrO ₂	1.0	6.209	6.650	3.989	9.002	0.819
ITO/PET	1.0	4.077	6.841	1.244	2.365	0.107
ITO/glass	3.201	4.039	4.935	1.850	0.302	0.0854
MoSi	1	7.948	6.628	2.714	9.081	7.870
NiCr	1	0.890	0	14.270	0	4.020
NiFe	1	0.880	0	14.790	0	4.780
Si ₃ N ₄	1.0	5.377	3.186	0	0	0
SnO ₂	0.714	0.792	4.933	7.814	1.141	15.546
Та	1	1.790	0	21.530	0	19.799
TiN	1.348	3.610	5.067	4.769	6.339	4.343
YAG:Tb(10%)	1.0	2.545	10.342	0.793	0	0
Y ₃ Fe ₅ O ₁₂	3.410	4.636	3.369	0.305	0.346	0.165
ZnO	1	2.900	7.250	5.450	2.700	17.150

Materials	Q 3	٤s	ω	ω _p	Γo	Γd
AI	0.884	0	0	13.346	0	0.119
Au	-4.230*	6.956	8.481	7.971	7.449	0.078
BiSrCuO	2.059	4.403	5.048	2.023	3.388	0.239
C60	2.306	3.362	4.556	0.000	1.046	0.000
Cr	1.769	56.676	1.311	4.179	5.715	0.609
Cu	3.070	8.173	2.234	8.896	0.880	-0.047
CuPC	2.551	0	0	0	0	0
Mo - Pal	2.216	10.869	4.190	8.306	2.048	0.347
p-Si	7.000	0	0	0	0	0
SiC - pal	2.922	6.401	6.843	0.000	0.974	0.000
Та	1.129	2.617	3.140	7.683	1.527	0.045
Ti	-0.551	0	0	5.415	0	1.942
TiN	3.050	0	0	12.071	0	11.321

List of materials following Drude and double Lorentz oscillator models

	r					
Materials	f ₁	ω _{0,1}	Y 1	f ₂	ω _{0,2}	γ 2
Al	9.255	1.990	1.667	7.766	1.572	0.391
Au	-1.476	0.182	4.243	-3.011	6.619	2.692
BiSrCuO	0.104	3.058	0.285	0.154	3.816	0.514
C60	0.223	2.704	0.439	0.364	3.523	0.420
Cu	-6.330	-2.137	0.909	9.926	4.597	10.564
Cr	1.648	6.769	3.606	0	0	0
CuPC	0.425	2.022	0.210	0.149	3.574	0.334
Mo - Pal	3.093	1.712	0.557	11.119	2.426	1.475
p-Si	6.000	3.400	1.000	5.500	4.400	1.250
SiC - pal	0.154	2.403	2.188	0.080	4.042	1.203
Та	-1.935	-1.971	0.926	18.717	7.896	26.969
Ti	0.300	-5.789	-0.404	2.102	-3.328	2.203
TiN	-1.089	2.006	1.026	0.929	4.743	1.011

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Appendix A4: New Amorphous Dispersion Formula

Spectroscopic ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected and polarized light in order to characterize thin film optical functions and other properties. The measured data are used to describe a model where each layer refers to a given material. The model uses mathematical relations called dispersion formulae that help to evaluate the material's optical properties by adjusting specific fit parameters.

This application note deals with the new amorphous dispersion formula. For further information about the original theory derived from Forouhi-Bloomer, please refer to the technical note «Forouhi Bloomer alias Amorphous dispersion formula».

New Amorphous theoretical model

The «New Amorphous» dispersion formula was derived by Horiba Jobin Yvon on the basis of Forouhi-Bloomer formulation. This new formulation was established in order to give a Lorentzian shape to the expressions of the extinction coefficient and refractive index. The absorption coefficient is given by :

$$k(\omega) = \begin{cases} \frac{f_{j} \cdot (\omega - \omega_{g})^{2}}{(\omega - \omega_{j})^{2} + \Gamma_{j}^{2}} & ; \text{for } \omega > \omega_{g} \\ 0 & ; \text{for } \omega \le \omega_{g} \end{cases}$$
(1)

The refractive index is written through this formula :

$$n(\omega) = n_{\infty} + \frac{B \cdot (\omega - \omega_{j}) + C}{(\omega - \omega_{j})^{2} + \Gamma_{j}^{2}} (2)$$

where :

$$\begin{cases} \mathsf{B}_{j} = \frac{\mathsf{f}_{j}}{\Gamma_{j}} \cdot \left(\Gamma_{j}^{2} - (\omega_{j} - \omega_{g})^{2}\right) \\ \mathsf{C}_{j} = 2 \cdot \mathsf{f}_{j} \cdot \Gamma_{j} \cdot (\omega_{j} - \omega_{g}) \end{cases} (3)$$

Extension to multiple oscillators: «N -New Amorphous»

The New Amorphous formulation can be extended to the case where many oscillators are present in the material. Such dispersions are called « N – New Amorphous » and are given in the array below.

Formula	Number of oscillators	Number of parameters	
New Amorphous	N=1	5	

Double New Amorphous	N=2	8
Triple New Amorphous	N=3	11

The equations for $k(\omega)$ and $n(\omega)$ for N oscillators are given below :

- for the extinction coefficient :

$$k(\omega) = \begin{cases} \sum_{j=1}^{N} \frac{f_j \cdot (\omega - \omega_g)^2}{(\omega - \omega_j)^2 + \Gamma_j^2} & ; \text{ for } \omega > \omega_g \\ 0 & ; \text{ for } \omega \le \omega_g \end{cases}$$
(4)

- for the refractive index :

$$\mathbf{n}(\omega) = \mathbf{n}_{\infty} + \sum_{j=1}^{N} \frac{\mathbf{B}_{j} \cdot (\omega - \omega_{j}) + \mathbf{C}_{j}}{(\omega - \omega_{j})^{2} + \Gamma_{j}^{2}}$$
(5)

Increasing the number of oscillators leads to a shift of the peaks of absorption toward the ultraviolet region.

The parameters of the equation

Parameter describing the refractive index

The term $n(\infty)$ is an additional parameter, at least greater than one and equal to the value of the refractive index when $\omega \rightarrow \infty$.



4 parameters describe the extinction coefficient.

- f_j (*j*=1, 2, 3) (in eV) is related to the strength (amplitude) of the extinction coefficient peak. As the value of f_j increases both values of the refractive index and extinction coefficient increases. Generally, $0 \le f_j \le 1$.
- Γ_j (j=1, 2, 3) (in eV) is the broadening term of the peak of absorption. The larger Γ_j is the larger the absorption peak but the smaller its amplitude. Generally, 0.2<Γ_j<8.
- ω_j (in eV) is approximately the energy at which the extinction coefficient is maximum (peak of absorption). As the value of ω_j increases the absorption peak is shifted towards the UV region. Generally, $1.5 < \omega_j < 10$.
- ω_g (in eV) is the energy band gap. It is equal to the minimum of energy required for a transition from the valence band to the conduction band. It is the energy from which the absorption starts to be non-zero: $k(E \ge Eg) \ge 0$. Always, $\omega_g < \omega_f$.

Relations between «Amorphous» and «New Amorphous» parameters

The array below gives the relations between the parameters of Forouhi-Bloomer and New Amorphous dispersion formulae.

Forouhi-Bloomer	New Amorphous
А	f _i
В	2.ω _j
С	$\omega_{j}^{2}+\Gamma_{j}^{2}$
$\sqrt{\epsilon_{\infty}}$	n _∞
Eg	ω _g

N.B: In DeltaPsi2 software, the user is advised to perform fitting using the «New Amorphous» formula (and its extensions) instead of the «Amorphous» one because *B* and *C* parameters are often correlated.

Parameter setup

Note that :

- The graphs below show the different contributions (in red dashed lines) to the imaginary part of the Double New Amorphous dielectric function (in red bold line).
- The sign « α » before a given parameter means that either the amplitude or the broadening of the peak is linked to that parameter.

New Amorphous function



Double New Amorphous function



Optical properties of Alq3 given by the Double New Amorphous function

Applications to materials

The new amorphous model works particularly well for amorphous materials exhibiting an absorption in the visible and/or FUV range (absorbing dielectrics, semiconductors, polymers).

We advise the user to compare the results obtained between the new amorphous and Tauc-Lorentz dispersion formula. The Tauc-Lorentz model may fit better the absorption part of the experimental spectrum.



Materials following the New Amorphous model

The asterisk * refers to parameters that are negative and thus do not have any physical meaning but represent good starting values to perform the fit on the material.

Materials	nø	ω _g	fj	ω _j	Γ _j	S. R. (eV)
a-C	1.564	0.542	0.123	3.620	2.645	1.5 - 3.5
AlGaAs	1.123	4.170	2.192	5.569	0.510	0.6 - 3
Al ₂ O ₃	1.56	9.85	2.46	10.4	0.44	0.73 - 4.73
AIN	1.574	0.490	0.558	5.604	7.897	1.5 - 5
Alq3	1.526	0.422	0.061	8.465	2.278	0.75 - 4
Amino- Silane	1.298	1.342	0.181	4.659	3.166	1.5 - 6
a-Si	1.750	1.250	0.750	3.5	1.3	1.5 - 5
p-Si	2.626	0.599	0.123	3.908	0.510	1.5 - 5
AsSe	2.499	1.686	0.242	2.471	0.562	0.75 - 4.5
AZO	1.923	2.828	0.134	3.575	0.516	0.6 - 2
Ba _{0.5} Sr _{0.5} TiO ₃	1.843	3.702	0.256	4.892	0.940	1.5 - 5.5
BaTiO ₃	1.500	3.000	0.100	5.000	0.500	1.5 - 5.5
BaFeO	2.538	0.936	0.022	2.374	0.213	0.75 - 4.7
BK7	1.456	5.500	0.073	8.127	1.347	1.5 - 5.5
Ge doped CH	3.270	-3.336*	0.010	2.550	0.245	1.5 - 6
СН	1.650	0.488	0.084	3.443	1.839	0.7 - 3
CFx	1.361	2.108	0.023	4.299	1.186	0.7 - 4.75
DLC	1.501	0.420	0.093	2.783	2.237	0.75 - 4.75
Organic EL/HTL	0.580	-1.104*	0.222	11.552	4.026	0.75 - 4
Fe oxide	1.787	0.510	0.179	3.282	1.149	0.75 - 4.75
GaN	2.145	2.620	0.048	4.298	0.290	0.65 - 3
GeO ₂	1.576	1.172	0.199	1.262	6.064	0.75 - 2
Glass	1.130	3.428	0.009	16.953	0.282	0.65 - 5
Glass 1737	0.626	3.238	0.0549	41.765	2.309	1.5 - 6.5
HfO ₂	1.436	4.831	0.259	12.138	2.718	0.75 - 4.5
InN	2.200	1.900	0.200	2.250	0.250	
In _x Ga _{1-x} N	1.857	-6.448*	0.006	4.915	0.762	0.75 - 3
In ₂ O ₃	1.526	3.429	0.111	3.631	0.484	1.5 - 5
LiAlO ₃	2.139	-12.354*	0.003	2.533	0.518	
LiNbO ₃	2.149	4.183	0.478	4.902	0.556	1.5 - 5
MoSi	2.089	-5.109*	0.172	2.928	4.385	0.75 - 4.5
LPCVD Nitride	1.834	2.192	0.164	5.412	2.170	1.5 - 4.7
NPD	1.781	2.763	0.054	3.374	0.318	0.75 - 4

Materials	nø	ω _g	f ₁	ω ₁	Γ ₁	S. R. (eV)
e-PEDOT-PSS	1.563	4.149	0.0027	5.163	0.152	
o-PEDOT-PSS	1.471	3.374	0.025	6.542	0.773	0.8 - 5.4
PI	1.485	3.604	0.092	5.375	1.322	1.5 - 6.5
Photoresist	1.557	1.301	0.011	5.352	0.490	1.5 - 6.5
Polymer	1.660	0.875	0.0434	2.435	0.442	0.6 - 4
РММА	1.456	3.667	0.130	4.212	7.144	0.6 - 6
e - PPV	1.516	2.878	0.0832	3.144	0.126	08-53
o-PPV	1.538	1.766	0.0763	2.383	0.241	0.0 0.0
PZT	2.345	3.000	0.183	4.350	0.424	0.6 - 4.7
Resist	1.537	2.629	0.00364	5.312	0.079	1.5 - 3
SiGeC	3.023	0.500	0.100	4.000	0.900	0.8 - 3
SiN _x	2.071	4.388	0.657	5.288	2.848	1.5 - 6.5
SiN _x	1.750	3.420	0.160	7.310	1.700	0.75 - 4.5
Si ₃ N ₄	1.794	2.221	0.031	7.340	0.623	0.75 - 4.75
SiO	1.500	2.362	0.0841	5.878	1.561	0.6 - 6
TAC	1.448	2.947	0.0507	5.244	1.966	1.5 - 4
Ta ₂ O ₅	1.788	3.892	0.225	5.170	0.700	1.4 - 5
TeO ₂	2.096	2.952	0.278	4.282	0.652	0.75 - 4
TiO ₂	2.096	2.952	0.278	4.282	0.652	0.75 - 4
TiVO _x	1.879	1.479	0.124	3.848	0.849	0.75 - 4
WO ₃	1.818	2.199	0.025	4.734	0.325	1.5 - 4.5
WO _x	2.646	2.952	0.802	3.105	1.011	1.5 - 4.5
ZnO _x	1.405	2.737	0.102	3.385	0.305	1.5 - 4
ZnO ₂	1.852	3.062	0.127	3.372	0.218	1.5 - 4

Materials following the Double New Amorphous model

Materials	nø	ω_{g}	f ₁	ω ₁	
Alq3	1.630	2.590	0.029	4.642	
AZO	1.605	1.825	0.001	3.520	
EIL	1.329	0.376	0.0203	5.420	
EML	1.329	0.376	0.0203	5.420	
ETL	1.702	2.881	0.0979	2.976	
FeCo	1 266	0 524	0 142	2 230	
oxide	1.200	0.024	0.142	2.200	
Blue	1 616	1 463	0 0333	1 620	
Filter	1.010	1.100	0.0000	1.020	
Ge	2.389	-0.918 [*]	0.366	4.436	
HIL	1.695	2.695	0.135	3.275	
HTL	1.702	2.658	0.097	3.201	
In ₂ O ₃	1.427	3.047	0.020	4.922	
ITO	1.555	2.866	0.777	3.385	
LaTeMnO	1.650	0.227	0.049	4.307	
NPD	1.717	2.893	0.066	3.167	
p-Si	1.800	2.250	0.550	3.250	
TPD	1.571	2.515	0.00723	3.933	
Varnish	1.604	-0.519*	0.053	-1.662*	

Materials	Γ ₁	f ₂	ω ₂	Γ ₂	S. R. (eV)	
Alq3	0.613	0.015	2.990	0.187	0.75 - 4	
AZO	0.176	0.022	4.126	0.761		
EIL	0.385	0.003	3.031	0.169	0.8 - 4	
EML	0.385	0.003	3.031	0.169	0.8 - 4	
ETL	0.100	-0.060*	3.398	-0.538*	0.8 - 4	
FeCo	0 976	0 905	0 346	-0.016*	0 75 - 4	
oxide	0.370	0.000	0.040	-0.010	0.70 - 4	
Blue	0.0634	0 00588	1 928	0 117	07-2	
Filter	0.0034	0.00000	1.520	0.117	0.7 - 2	
Ge	1.862	0.003	2.248	0.119	0.8 - 2	
HIL	0.369	0.0109	3.688	0.229	0.8 - 4	
HTL	0.298	0.0071	3.604	0.167	0.8 - 4	
In ₂ O ₃	0.740	0.049	4.049	0.662	1.5 - 5	
ITO	0.230	0.003	5.025	0.075	0.8 - 4	
LaTeMnO	0.771	0.059	1.815	0.459		
NPD	0.185	0.031	3.493	0.232	0.75 - 4	
p-Si	0.450	0.250	4.250	0.500		
TPD	0.232	0.0206	3.440	0.202		
Varnish	0.201	0.001	6.367	0.654	1.5 - 6.5	

Materials following the Triple New Amorphous model

Materials	nø	ω_{g}	f ₁	ω ₁	Γ ₁	f ₂
CuPc	1.300	1.000	0.003	1.700	0.050	0.002
PET	1.653	2.216	0.00036	4.174	0.133	0.0095
Poly- carbonate	1.000	3.500	0.010	4.500	0.200	0.100
ARC coating Polymer	1.565	4.373	0.00289	5.451	0.189	0.0316
Photoresist	1.480	3.980	0.00048	5.426	0.0536	0.0046

Materials	ω ₂	Γ2	f ₃	ω ₃	Γ ₃	S. R. (eV)
CuPc	2.000	0.050	0.020	3.750	0.479	0.6 - 4
PET	5.018	0.303	0.00704	6.421	0.339	1.5 - 6.5
Poly- carbonate	5.500	1.000	0.050	6.300	1.000	1.5 - 6.5
ARC coa- ting Polymer	4.385	0.0204	0.0248	6.363	0.455	1.5 - 6.5
Photoresist	4.397	0.115	0.0242	6.453	0.366	1.5 - 6.5

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Appendix A5: Tauc-Lorentz Dispersion Formula

Spectroscopic ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected and polarized light in order to characterize thin film optical functions and other properties. The measured SE data are used to describe a model where layers refer to given materials. The model uses mathematical relations called dispersion formulae that help to evaluate the material's optical properties by adjusting specific fit parameters.

This technical note deals with Tauc-Lorentz dispersion formula.

Theoretical model

Jellison and Modine developed this model (1996) using the Tauc joint density of states and the Lorentz oscillator. The complex dielectric function is :

$$\widetilde{\varepsilon}_{TL} = \varepsilon_{r,TL} + i \cdot \varepsilon_{i,TL} = \varepsilon_{r,TL} + i \cdot \left(\varepsilon_{i,T} \times \varepsilon_{i,L}\right) (1)$$

Here the imaginary part $\varepsilon_{i,TL}$ of the dielectric function is given by the product of imaginary part of Tauc's (1966) dielectric $\epsilon_{i,T}$ function with Lorentz one $\epsilon_{i,L}$. In the approximation of parabolic bands, Tauc's dielectric function describes inter-band transitions above the band edge as :

$$\varepsilon_{i,T}(E > E_g) = A_T \cdot \left(\frac{E - E_g}{E}\right)^2$$
(2)

where :

- A_T is the Tauc coefficient

- E is the photon energy

- E_q is the optical band gap

The imaginary part of Tauc's dielectric function gives the response of the material caused by inter-band mechanisms only : thus $\varepsilon_{i, T}$ ($E \le E_g$) = 0. The imaginary part of the Lorentzian oscillator model is

given by :

$$\varepsilon_{i,L}(E) = \frac{A_L \cdot E_0 \cdot C \cdot E}{\left(E^2 - E_0^2\right)^2 + C^2 \cdot E^2}$$
(3)

where :

- A_L is the strength of the $\varepsilon_{2, TL}(E)$ peak
- C is the broadening term of the peak
- E_0 is the peak central energy

By multiplying equation (2) by equation (3), Jellison sets up a new expression for $\varepsilon_{i,L}(E)$:

$$\varepsilon_{i,TL}(E) = \varepsilon_{i,L} \times \varepsilon_{i,T} = \begin{cases} \frac{1}{E} \frac{AE_0 C(E - E_g)^2}{(E^2 - E_0^2)^2 + C^2 E^2} & \text{for } E > E_g \\ 0 & \text{for } E \le E_g \end{cases}$$
(4)

where $A = A_T \times A_I$.

The real part $\epsilon_{r,TL}$ of the dielectric function is derived from the expression of ϵ_i using the Kramers-Kronig integration. Then, it comes the following expression for ε_i :

$$\varepsilon_r(E) = \varepsilon_r(\infty) + \frac{2}{\pi} \cdot P \cdot \int_{E_g}^{\infty} \frac{\zeta \cdot \varepsilon_i(\zeta)}{\zeta^2 - E^2} d\zeta \quad (5)$$

where P is the Cauchy principal value containing the residues of the integral at poles located on lower half of the complex plane and along the real axis.

According to Jellison and Modine (Ref. 1), the derivation of the previous integral yields :

$$\begin{split} \varepsilon_{r,TL}(E) &= \varepsilon_{\infty} + \frac{A \cdot C \cdot a_{\ln}}{2 \cdot \pi \cdot \zeta^{4} \cdot a \cdot E_{0}} \cdot \ln\left[\frac{E_{0}^{2} + E_{g}^{2} + a \cdot E_{g}}{E_{0}^{2} + E_{g}^{2} - a \cdot E_{g}}\right] \\ &- \frac{A}{\pi} \cdot \frac{a_{a \tan}}{\zeta^{4} \cdot E_{0}} \cdot \left[\pi - \arctan\left(\frac{2 \cdot E_{g} + a}{C}\right) + \dots \right. \\ &+ \arctan\left(\frac{a - 2 \cdot E_{g}}{C}\right)\right] \\ &+ \frac{4 \cdot A \cdot E_{0} \cdot E_{g} \cdot \left(E^{2} - \gamma^{2}\right)}{\pi \cdot \zeta^{4} \cdot a} \times \dots \\ &\times \left[\arctan\left(\frac{a + 2 \cdot E_{g}}{C}\right) + \arctan\left(\frac{a - 2 \cdot E_{g}}{C}\right)\right] \\ &- \frac{A \cdot E_{0} \cdot C \cdot \left(E^{2} + E_{g}^{2}\right)}{\pi \cdot \zeta^{4} \cdot E} \cdot \ln\left(\frac{|E - E_{g}|}{E + E_{g}}\right) + \dots \\ &+ \frac{2 \cdot A \cdot E_{0} \cdot C}{\pi \cdot \zeta^{4}} \times E_{g} \cdot \ln\left[\frac{|E - E_{g}| \cdot \left(E + E_{g}\right)}{\sqrt{\left(E_{0}^{2} - E_{g}^{2}\right)^{2} + E_{g}^{2} \cdot C^{2}}}\right] \end{split}$$
(6)

where:

$$\begin{cases} a_{\ln} = \left(E_{g}^{2} - E_{0}^{2}\right) \cdot E^{2} + E_{g}^{2} \cdot C^{2} - E_{0}^{2} \cdot \left(E_{0}^{2} + 3 \cdot E_{g}^{2}\right) \\ a_{a \tan} = \left(E^{2} - E_{0}^{2}\right) \cdot \left(E_{0}^{2} + E_{g}^{2}\right) + E_{g}^{2} \cdot C^{2} \\ \alpha = \sqrt{4 \cdot E_{0}^{2} - C^{2}} \\ \gamma = \sqrt{E_{0}^{2} - C^{2} / 2} \\ \zeta^{4} = \left(E^{2} - \gamma^{2}\right)^{2} + \alpha^{2} \cdot C^{2} / 4 \end{cases}$$
(7)

Extension to multiple oscillators

In the case of multiple oscillators (N>1) the expression of ε_i contains now a series over all the oscillators.

$$\varepsilon_{i} = \begin{cases} \sum_{i=1}^{N} \frac{1}{E} \times \frac{A_{i} \cdot E_{i} \cdot C_{i} \cdot (E - E_{g})^{2}}{(E^{2} - E_{i}^{2})^{2} + C_{i}^{2} \cdot E^{2}} & \text{for } E > E_{g} \\ 0 & \text{for } E \le E_{g} \end{cases}$$
(8)

and ε_r is re-written as the following sum :

$$\varepsilon_r(E) = \varepsilon_r(\infty) + \sum_{i=1}^N \frac{2}{\pi} \cdot P \cdot \int_{E_g}^{\infty} \frac{\xi \cdot \varepsilon_i(\xi)}{\xi^2 - E^2} \,\mathrm{d}\xi \quad (9)$$

Deriving this integral yields the analytical expression of the real part of the dielectric function :

$$\varepsilon_{r}(E) = \varepsilon_{r}(\infty) + \sum_{i=1}^{N} \left\{ \frac{A_{i} \cdot C_{i} \cdot a_{\ln}}{2 \cdot \pi \cdot \zeta^{4} \cdot \alpha \cdot E_{i}} \cdot \ln\left[\frac{E_{i}^{2} + E_{g}^{2} + \alpha \cdot E_{g}}{E_{i}^{2} + E_{g}^{2} - \alpha \cdot E_{g}}\right] - \frac{A_{i}}{\pi} \cdot \frac{a_{a\tan}}{\zeta^{4} \cdot E_{i}} \cdot \left[\pi - \arctan\left(\frac{2 \cdot E_{g} + \alpha}{C_{i}}\right) + \dots + \arctan\left(\frac{\alpha - 2 \cdot E_{g}}{C_{i}}\right)\right] + \frac{4 \cdot A_{i} \cdot E_{i} \cdot E_{g} \cdot (E^{2} - \gamma^{2})}{\pi \cdot \zeta^{4} \cdot \alpha} \cdot \left[\arctan\left(\frac{\alpha + 2 \cdot E_{g}}{C_{i}}\right) + \arctan\left(\frac{\alpha - 2 \cdot E_{g}}{C_{i}}\right)\right] - \frac{A_{i} \cdot E_{i} \cdot C_{i} \cdot (E^{2} + E_{g}^{2})}{\pi \cdot \zeta^{4} \cdot E} \cdot \ln\left(\frac{|E - E_{g}|}{E + E_{g}}\right) + \frac{2A_{i}E_{i}C_{i}}{\pi \cdot \zeta^{4}} \cdot E_{g} \ln\left[\frac{|E - E_{g}|(E + E_{g})}{\sqrt{(E_{0}^{2} - E_{g}^{2})^{2} + E_{g}^{2}}C_{i}^{2}}\right]\right\} (10)$$

The parameters of the equations

The name of the different Tauc-Lorentz formulae present in the DeltaPsi2 software depending on the number of oscillators are given below. It indicates the number of parameters too.

Formula	Number of oscillators	Number of parameters
Tauc-Lorentz	N=1	5
Tauc-Lorentz 2	N=2	8
Tauc-Lorentz 3	N=3	11

Increasing the number of oscillators leads to a shift of the peaks of absorption toward the ultraviolet region.

1 parameter is linked to the real part of the dielectric function

ε_r (∞) = ε_∞ is the high frequency dielectric constant. This is an additional fitting parameter that prevents ε₁ from converging to zero for energies below the band gap. Generally, ε_∞>1.

At least 4 parameters describe the imaginary part of the dielectric function

- A_i (in eV) is related to the strength of the ith absorption peak. The subscript «i» refers to the number (i=1, 2 or 3) of oscillators. As A_i increases, the amplitude of the peak increases and the Full Width At Half Maximum (FWHM) of that peak gets slightly larger. Generally, 10<A_i<200.
- C_i (in eV) is the broadening term ; it is a damping coefficient linked to FWHM of the ith peak of absorption. The higher it is the larger that peak becomes and at the same time the smaller its amplitude is. Generally $0 < C_i < 10$.
- E_g (in eV) is the optical band gap energy.
- E_i (in eV) is the energy of maximum transition probability or the energy position of the peak of absorption. The subscript «i» refers to the number (i=1, 2 or 3) of oscillators. Always, Eg<Ei.

Limitation of the model

The Tauc-Lorentz model requires ε_i to be zero for energies less than the band gap. Consequently, Tauc-Lorentz model does not take into account intra-band absorption: any defect or intra-band absorption increases ε_i below the band gap and generates bad fits in that region.

Valid spectral range

The Tauc-Lorentz model well describes the behaviour of materials for energies $E \le E_i$ where E_i is the transition energy of the oscillator of highest order.



Parameter setup

The Tauc-Lorentz model works particularly well for amorphous materials exhibiting an absorption in the visible and/or FUV range (absorbing dielectrics, semiconductors, polymers).

Note that :

- The graph below shows the different contributions (in red dashed lines) to the imaginary part of the Tauc Lorentz dielectric function (in red bold line).
- The sign « x » before a given parameter means that either the amplitude or the broadening of the peak is linked to that parameter.

Tauc-Lorentz function



Dielectric function of a-Si represented by a single Tauc-Lorentz oscillator

Double Tauc-Lorentz function



Dielectric function of a blue filter represented by a double Tauc-Lorentz oscillator

Applications to materials

Materials following the Tauc-Lorentz model

The asterisk * refers to parameters that are negative and thus do not have any physical meaning but represent good starting values to perform the fit on the material.

Materials	Eg	Q 3	А	E ₀	С	S. R. (eV)
a-C	1.305	3.774	38.835	6.335	6.227	1.5 - 5
AlGaN	3.300	3.000	100.000	4.500	1.000	1.5 - 2.5
As_2S_3	2.349	1.240	133.819	4.574	5.935	0.75 - 4.75
AsSSe	1.000	1.600	50.000	3.500	2.000	0.75 - 4.5
DLC	0.374	2.960	11.558	3.533	3.346	0.7 - 2
GaN	3.200	3.500	100.000	4.500	1.000	1,5 - 3
GeSbSe	1.851	2.062	54.703	3.412	2.045	0.65 - 3
InGaN	3.000	3.500	100.000	4.500	1.000	1,5 - 3
Polymer	3.027	1.701	32.921	5.759	2.416	0.7 - 4
poly-Si	0.882	3.937	49.597	3.711	0.321	1.5 - 3.25
a-Si	2.030	1.692	142.599	3.840	1.908	0.75 - 3.65
a-Si:B	1.267	2.148	143.380	3.617	1.994	0.75 - 4.75
a-Si:H	1.393	0.626	171.105	3.582	2.201	0.75 - 4.75
Si _x O _y N _z	3.864	1.760	77.340	8.363	3.550	1.5 - 6.5
Ta-C	1.395	2.558	67.312	8.713	9.760	1.5 - 5
ZnS	2.976	4.500	57.167	3.954	1.200	1.5 - 6.5
GeSbTe	0,450	2,024	158,795	1,688	2,397	0,65 - 3

Materials following the Double Tauc-Lorentz model

Materials	Eg	و _ي ع	A ₁	E ₁
Blue Filter	1.451	2.346	6.236	1.617
Oxide coating	1.168	0.894	41.060	6.464
Polymer	3.420	2.771	113.260	4.234
SiC	1.521	-1.657*	4.026	5.678
Ta ₂ O ₅	4.261	1.827	188.580	4.188

Materials	C ₁	A ₂	E ₂	C 2	S. R. (eV)
Blue Filter	0.146	2.227	1.987	0.377	0.6 - 2.2
Oxide coating	4.140	-6.269*	3.763	2.390	1.5 - 5
Polymer	1.455	6.870	4.806	8.345	0.7 - 5
SiC	-5.129*	51.264	9.561	3.869	1.5 - 6.5
T a ₂ O ₅	5.278	58.367	5.354	1.246	1.5 - 6.5

Materials following the Triple Tauc-Lorentz model

			•			
Materials	Eg	્ય 3	A ₁	E ₁	C ₁	A ₂
a - Si	1.405	1.029	-16.227*	6.800	5.454	1.310
HfO₂	4.972	2.764	80.221	7.451	0.876	4.604

Materials	E ₂	C ₂	A ₃	E ₃	C ₃	S. R. (eV)
a - Si	0.807	2.215	185.343	3.707	2.379	0.6 - 6
HfO ₂	4.192	6.834	4.469	6.368	0.599	1.5 - 6.5

References

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- 2) 3) Erratum, G. E. Jellison & F. A. Modine, Appl. Phys. Lett. 69 (14), 2137 (1996) H. Chen, W. Z. Shen, Eur. Phys. J. B. 43, 503-507 (2005).

Appendix A6: Cauchy and related Empirical Dispersion Formulae for Transparent Materials

Spectroscopic ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected and polarized light in order to characterize thin film optical functions and other properties. The measured data are used to describe a model where each layer refers to a given material. The model uses mathematical relations called dispersion formulae that help to evaluate the thickness and optical properties of the material by adjusting specific fit parameters.

This technical note deals with the Cauchy and related empirical transparent dispersion formulae to calculate the real (n) and imaginary (k) parts of the complex refractive index for a material.

Theoretical Model

Equation of Cauchy Transparent

The earliest dispersion formula was established by Cauchy (1836) who set up simple empirical dispersion law. The "Cauchy Transparent" dispersion works best when the material has no optical absorption in the visible spectral range and consequently generally has a normal dispersion which means a monotonous decreasing refractive index with increasing wavelength in this way:

$$1 < n(\lambda_{red}) < n(\lambda_{blue})$$

The following equation connects the refractive index to the wavelength (in nm):

$$\begin{cases} n(\lambda) = A + \frac{10^4 \cdot B}{\lambda^2} + \frac{10^9 \cdot C}{\lambda^4} \\ k(\lambda) = 0 \end{cases}$$
(1)

Equation of Cauchy Absorbent

A second formulation of the Cauchy model is the «Cauchy Absorbing» dispersion and it is more suitable to describe the optical properties of weakly absorbing materials. Here a non-zero extinction coefficient is given in an expression similar to the previous used for the refractive index:

$$\begin{cases} n(\lambda) = A + \frac{10^4 \cdot B}{\lambda^2} + \frac{10^9 \cdot C}{\lambda^4} \\ k(\lambda) = 10^{-5} \cdot D + \frac{10^4 \cdot E}{\lambda^2} + \frac{10^9 \cdot F}{\lambda^4} \end{cases}$$
(2)

The parameters of the equations

3 parameters are used in the equation of the Cauchy transparent model and 6 parameters in the Cauchy absorbing model.

Parameters describing the refractive index

- *A* is a dimensionless parameter: when $\lambda \to \infty$ then $n(\lambda) \to A$.
- *B* (nm²) affects the curvature and the amplitude of the refractive index for medium wavelengths in the visible.
- *C* (nm⁴) affects the curvature and amplitude for smaller wavelengths in the UV. Generally,

$$0 < |C| < |B| < 1 < A$$
 (3).

3 parameters describe the extinction coefficient.

- *D* is a dimensionless parameter similar to *A*,
- $E (nm^2)$ is analogous to B,
- $F(nm^4)$ behaves like C.

Limitation of the model

Cauchy's formulation cannot be easily applied to metals and semiconductors.

The parameters used do not have any physical meaning and therefore, these empirical relations are not Kramers-Kronig consistent. From first principles, the Kramers-Kronig relation relates the index of refraction and extinction coefficient parts; it means they are not independent quantities. In other words, if the value of the extinction coefficient is known over the entire spectral range, the index of refraction coefficient can be calculated.



Parameter setup

Cauchy Transparent function



Optical properties of SiO₂ given by the Cauchy transparent function

Cauchy Absorbent function



Optical properties of SiN given by the Cauchy absorbent function

Application to materials

The Cauchy model is used for transparent materials like insulators, glasses exhibiting no or very low optical absorption in the Far Ultra Violet.

Materials	А	В	С	D	E	F	S.R (eV)
ChG Glass	6.5	0.7	0	0	0	0	
HfO	1.993	1.303	0.158	0	0	0	1.5 - 5.5
MgF	1.386	0.117	0.109	0	0	0	1.5 - 5.5
PI	1.631	0.497	1.337	771.776	- 0.587	1.117	
SiN	2.000	0.638	0.690	-0.658	-0.236	0.212	1 - 6.0
SiO ₂	1.447	3660	0	0	0	0	1 - 6.0
TiO₂	2.374	1.932	6.855	0	0	0	1.5 - 6.0

References

- 1. L. Cauchy, Bull. des sc. Math., 14, 9 (1830).
- http://en.wikipedia.org/wiki/Cauchy equation.
 G. Ghosh, Handbook of Thermo-Optic Coefficients of optical Materials with Applications, Academic Press.

Appendix A6a: Briot Dispersion Formula

Theoretical model

This dispersion model is based on a Laurent series (1864) expressing the equation for the refractive index of transparent materials:

$$n^{2}(\lambda) = A_{0} + 10^{-2} \cdot A_{1} \cdot \lambda^{2} + \frac{10^{-2} \cdot A_{2}}{\lambda^{2}} + \frac{10^{-4} \cdot A_{3}}{\lambda^{4}} + \frac{10^{-6} \cdot A_{4}}{\lambda^{6}} + \frac{10^{-7} \cdot A_{5}}{\lambda^{8}}$$
(1)

with:

$$k(E)=0 \ (2).$$

The previous equation of the refractive index is also known as the Schott equation because it was used by the Schott company until 1992 when it was abandoned and replaced by the Sellmeier formula. However, Briot's equation remains widespread use elsewhere.

The parameters of the equations

Parameters describing the refractive index

- A_0 (dimensionless), A_1 (in nm⁻²), A_2 (in nm²), A_3 (in nm⁴), A_4 (in nm⁶) and A_5 (in nm⁸) are the different coefficients of the development in Laurent series.
- A_0 is a positive parameter that prevents $n^2(\lambda)$ from being negative.
- The low order terms of the development ($A_1 < 0$ and A_2 contribute to the intensity of the refractive index curve for long wavelengths in the visible and IR regions.
- The high order terms of the development (A_3, A_4) and A_5 contribute to the intensity of the refractive index curve for small wavelengths in the UV region.

Limitation of the model

Briot's formulation cannot be easily applied to metals and semiconductors.

This empirical relation is not Kramers-Kronig consistent and therefore does not have any physical meaning.

Parameter setup

Schott-Briot function



Optical properties of SiO₂ given by the Schott-Briot function

Application to materials

The Briot model is used for transparent materials such as crystalline quartz across the range 300nm \sim 1600nm.

References

- 1. C. Briot, Essais sur la théorie mathématique de la lumière. Paris, Mallet-Bachelier, (1864). 2. www.optics.arizona.edu/Palmer/cgi-bin/index/dispeqns.pdf

Appendix A6b: Hartmann Dispersion Formula

Theoretical model

The Hartmann's (1926) model, has the following general form:

$$n(E) = A + \frac{C}{\lambda(E) - B} (1)$$

This equation works for transparent films that do not have absorption:

$$k(E)=0 \ (2).$$

The parameters of the equations

3 parameters are used in the equation of the Hartmann model and allow to define the refractive index.

- A (A > 1) is a dimensionless parameter; when λ
 →∞ then n(λ) → A.
- *B* and *C* have the dimension of wavelengths (nm).
- *B* is the resonance wavelength for which the refractive
- index diverges. The fit must be performed for wavelengths λ ≠ B otherwise n(λ) → ∞.
- C determines the amplitude (strength) of the refractive index for wavelengths λ → B.

Limitation of the model

The Hartmann formula cannot be easily applied to metals and semiconductors.

This empirical formula is not Kramers-Kronig consistent and therefore does not have any physical meaning.

Parameter setup

The Hartmann function has similar application to the Cauchy Transparent dispersion. It is useful for transparent materials with normal dispersion.

Hartmann function



Optical properties of SiO₂ given by the Hartmann function

Application to materials

The Hartmann model is used for transparent materials like insulators, glasses.

Materials	А	В	С	S.R (eV)
SiO ₂	1.429	124.312	8.335	0.7 - 6
TiO ₂	1.890	178.621	203.804	1.5 - 4

References

- 1. http://scienceworld.wolfram.com/physics/HartmannDispersion-Formula.html
- H. W. Lee, Trans. Opt. Soc., Volume 28, Issue 3, 161-167 (1926).
- 3. www.optics.arizona.edu/Palmer/cgi-bin/index/dispeqns.pdf



Appendix A6c: Conrady Dispersion Formula

Theoretical model

Conrady's equation (1960) allows the refractive index to be derived using the following equation for $n(\lambda)$:

$$n(\lambda) = A + \frac{10^2 \cdot B}{\lambda} + \frac{10^9 \cdot C}{\lambda^{3.5}} (1)$$

for a transparent film for which:

$$k(E)=0 \ (2).$$

The parameters of the equations

3 parameters are used in the expression of the Conrady model.

Parameters describing the refractive index

- *A* is a dimensionless parameter: when $\lambda \to \infty$ then $n(\lambda) \to A$.
- *B* (nm) has the same dimension as a wavelength.

This parameter affects the curvature and amplitude of the refractive index for visible wavelengths.

• *C* (nm^{3.5}) influences the behaviour (strength) of the refractive index at small wavelengths in the UV.

Generally,

$$0 < |C| < |B| < 1 < A$$
 (3)

Limitation of the model

Conrady's equation cannot be easily applied to metals and semiconductors.

This empirical formula is not Kramers-Kronig consistent and therefore does not have any physical meaning.

Parameter setup

Conrady function has similar application to the Cauchy Transparent formula. It is useful for transparent materials with normal dispersion.

Conrady function



Optical properties of SiO_2 given by the Conrady function

Application to materials

The Conrady model is used for transparent materials like insulators, glasses.

Materials	А	В	С	S.R (eV)
SiO ₂	1.427	0.111	0.00513	0.7 - 6
TiO ₂	2,500	-1.148	0.731	1.5 - 4

References

 A. E. Conrady, Applied Optics and Optical Design, Dover (1960).



Appendix A6d: Sellmeier Dispersion Model

Theoretical model

Equation of Sellmeier Transparent

The Sellmeier formula (1871) is semi-empirical but it remains more accurate than the Cauchy dispersion law for characterizing the refractive index of a material across a wider spectral range. The transparent Sellmeier equation for the index of refraction looks like:

$$n^{2}\left(\lambda\right) = A + B \times \frac{\lambda^{2}}{\lambda^{2} - \lambda_{0}^{2}} \ \left(1\right)$$

without absorption:

$$k(\lambda) = 0$$
 (2)

Parameters of the Sellmeier Transparent function

3 parameters are used in the equation of the Sellmeier Transparent model to describe the refractive index variation.

Parameters describing the refractive index

- A (1 ≤ A) is a dimensionless parameter that determines the value of the refractive index when when λ → ∞ and B ~ 0. It represents the contribution of the ultraviolet term.
- $B(A \le B)$ is another dimensionless parameter that determines the shape of the refractive index in the visible range.
- λ₀ (in nm²) is the resonance wavelength for which the refractive index diverges. The fit must be performed for wavelengths λ ≠ λ₀ otherwise n²(λ) →∞.

Equation of Sellmeier Absorbent

The Sellmeier Absorbent equation for the refractive index is written as follows:

$$n^{2}(\lambda) = \frac{1+A}{1+\frac{10^{4}\cdot B}{\lambda^{2}}}$$
(3)

and the extinction coefficient is given by the relation below:

$$k(\lambda) = \frac{C}{10^{-2} \cdot n \cdot D \cdot \lambda + \frac{10^2 \cdot E}{\lambda} + \frac{1}{\lambda^3}}$$
(4)

Parameters of the Sellmeier Absorbent function

2 parameters describe the refractive index

- \mathcal{A} is a dimensionless parameter that is linked to the amplitude of the refractive index. When $\lambda \rightarrow \infty$ then $n(\lambda) = \sqrt{1+A}$.
- B (in nm²) has an influence on the curvature of the refractive index: if B > 0, n increases with λ and the bigger B is the more straight n²(λ) is. If B < 0, n decreases with λ.

3 parameters describe the extinction coefficient

- *C* is a dimensionless parameter that determines the strength of the absorption coefficient curve.
- D (nm⁻¹) and E (nm) are terms corresponding to a development series similar to that of Schott-Briot formula. Increasing D and E involves a decrease of the absorption coefficient.

Limitation of the model

Sellmeier equations cannot be easily applied to metals and semiconductors.

These empirical relations are not Kramers-Kronig consistent and therefore do not have any physical meaning.

Parameter setup

The Sellmeier Transparent function behaves like the Cauchy Transparent function. It exhibits normal dispersion, so the refractive index decreases with increasing wavelengths.



Sellmeier Transparent function



Optical properties of SiO₂ given by the Sellmeier Transparent function

The Sellmeier Absorbent function behaves like the Cauchy Absorbing function. This function also gives decreasing absorption and dispersion with increasing wavelength. Note that this is not physical in the presence of absorption as the Kramers-Kronig relation states that the refractive index should normally increase with increasing wavelength.

Sellmeier Absorbent function



Optical properties of a polymer given by the Sellmeier absorbent function

Application to materials

The Sellmeier model is used for transparent materials such as organics, insulators, glasses.

Valid spectral range

The Sellmeier equation mostly holds for wavelengths far from λ_0 in the range of transparency.

References

- 1. http://en.wikipedia.org/wiki/Sellmeier_equation
- 2. http://www.schott.com/optics_devices/english/download/tie-29_refractive_index_v2.pdf



Appendix A6e: Fixed Index Dispersion Formula

Theoretical model

The Fixed index dispersion formula is given by constant refractive index and extinction coefficient for any wavelength:

$$\begin{cases} n(\lambda) = const. = n \\ k(\lambda) = const. = k \end{cases}$$

The Fixed index model is applicable to non dispersive

Application to materials

materials like air. It is often used when the measurement is performed in different ambient media (liquids). It can be used to decrease the number of fitting parameters when the material dispersion can be neglected.

The parameters of the equations

Only 2 parameters are needed to describe the model:

- *n* is the value of the refractive index.
- *k* is the value of the extinction coefficient.

Limitation of the model

This formula strictly describes the vacuum. The Fixed index equation does not apply to metals and semiconductors.

Parameter setup

Fixed index function



Fixed Index formula given a refractive index equal to 1.5



Appendix A7: Forouhi-Bloomer alias Amorphous Dispersion Formula

Spectroscopic ellipsometry (SE) is a technique based on the measurement of the relative phase change of reflected and polarized light in order to characterize thin film optical functions and other properties. The measured data are used to describe a model where each layer refers to a given material. The model uses mathematical relations called dispersion formulae that help to evaluate the thickness and optical properties of the material by adjusting specific fit parameters.

This application note deals with the original Forouhi & Bloomer dispersion formula also known as «Amorphous» (name used in the DeltaPsi2 software). We recommend the use of the updated «New Amorphous» formulation, please refer to the corresponding technical note.

Forouhi-Bloomer alias «Amorphous» theoretical model

The Forouhi-Bloomer formulation (1986, 1988) is applicable to amorphous semiconductors and dielectrics and is based on the quantum-mechanical theory of absorption. It takes into account the optical band gap in the inter-band region. It is supposed that the valence and conduction bands are parabolic and are separated by a forbidden band whose width is E_g . Peaks that can be seen in the optical spectrum correspond to transitions of electrons between two states. For an amorphous material, a single peak is present in the optical spectrum which refers to the transition between the bonding state in the valence band and the anti-bonding state in the conduction band.

For crystalline semiconductors dielectrics and metals, several peaks can be observed, indicating the transitions occurring between the critical-point in the valence and conduction bands.

In the interband region, metals, semiconductors and dielectrics have similar structure in their optical functions. Thus, the same physical processes are involved for these three types of materials.

Optical properties depend on inter-band transitions of electrons that are related to photon absorption.

The equation of $\alpha(\omega)$ below derives from the quantum expression of the absorption.

Equation of $\alpha(\omega)$

$$\alpha(\omega) = \lim_{\Delta x \to 0} \left[-\frac{1}{I} \cdot \frac{\Delta I}{\Delta x} \right] = \frac{\Theta(\omega) \cdot \Phi(\omega)}{I_0} = \frac{2 \cdot \omega \cdot k}{c}$$
(1)

where:

- $\Theta(\omega)$ is the total number of ways a photon, of energy ω is removed from the incident intensity in a unit volume and a layer of infinitesimal thickness Δx ; $\Theta(\omega)$

varies with the type (amorphous, dielectric, metallic) of the material considered.

- $\Phi(\omega)$ is the transition probability rate at which a photon is absorbed from the incident intensity in the frequency range $[\omega; \omega + d\omega]$.
- I_0 is the incident photon intensity.

With $\Phi(\omega)$ defined as:

The equation of $\Phi(\omega)$ is deduced from first-order timedependant perturbation theory for direct and finite electron transitions. $\Phi(\omega)$ depends on the dipole position matrix element squared between initial final states and also on the lifetime of the excited state through the following equation. For an amorphous material, the expression of $\Phi(\omega)$ is given by:

$$\Phi(\omega) = \frac{4\pi\hbar}{3c} e^2 \omega I_0 \left| \left\langle \sigma^* \middle| \vec{x} \middle| \sigma \right\rangle \right|^2 \left\{ \frac{\gamma}{\left(E_{\sigma^*} - E_{\sigma} - \hbar \cdot \omega \right)^2 + \frac{\hbar^2 \cdot \gamma^2}{4}} \right\}$$
(2.1)



Fig. 1 Orbital states of the energies of the energies E_{σ} and E_{σ}^{*} (Ref. 2)



where:

- e is the electron charge
- c is the light speed in vacuum
- \hbar is the Planck constant
- $E = h\omega$ is the photon energy
- E_{σ} is the energy of the initial state
- $\mathsf{E}_{\sigma}{}^{*}$ is the energy of the final excited state
- $\langle \sigma^* | \bar{x} | \sigma \rangle$ is the dipole matrix element squared between the bonding and anti-bonding states.

In the case of crystalline semiconductors, dielectrics and metals, $\Phi(\omega)$ is the sum over the number of types of transitions.

$$\Phi(\omega) = \frac{4 \cdot \pi \cdot \hbar}{3 \cdot c} \cdot e^2 \cdot \omega \cdot I_0 \cdot \sum_{j=1}^{N} \left| \left\langle \vec{k}_{crit}^c \middle| \vec{x} \middle| \vec{k}_{crit}^v \right\rangle \right|_j^2 \times \dots \\ \times \left\{ \frac{\gamma_j}{\left\{ \left[E_c\left(\vec{k}_{crit}\right) - E_v\left(\vec{k}_{crit}\right) \right]_j - \hbar \cdot \omega \right\} + \frac{\hbar^2 \cdot \gamma_j^2}{4} \right\}}$$
(2.2)

where:

- the sum over N refers to the sum over the number of observed peaks associated with the transitions.
- $\left|\left\langle \bar{k} \, {}_{crit}^{e} \, \left| \bar{x} \, \right| \bar{k} \, {}_{crit}^{v} \, \right\rangle \right|_{j}^{2}$ is the dipole matrix element squared between the ith critical-point state in the valence band and the ith critical-point state in the conduction band.

With $\Theta(\omega)$ defined as:

 $\Theta(\omega)$ is proportional to the number of possible transitions for a given photon of energy $\hbar\omega$ to be removed. $\Theta(\omega)$ depends on the product of the number of occupied electron states in the valence band times the number of unoccupied electron states in the conduction band. The densities of states in the valence and conduction bands are assumed to have a parabolic shape.

Therefore, $\Theta(\omega)$ is written as

$$\Theta(\omega) \propto \int dE_{\nu} \eta_{\nu}(E_{\nu}) \cdot \eta_{c}(E_{\nu} + \hbar \omega)$$
(3.1)

where:

- $\eta_{v}(E) = cst \times (E_{v}^{top} - E)^{1/2}$ (3.2) is the density of states in the valence band.

- $\eta_c(E) = cst \times (E E_c^{bottom})^{1/2}$ (3.3) is the density of states in the conduction band.
- the band gap energy is $E_g = E_c^{bottom} E_v^{top}$ (3.4).

For amorphous and crystalline materials the final expression of $\Theta_{\alpha}(\omega)$ is

$$\Theta_a(\omega) = \Theta_c(\omega) = cst \times (\hbar \cdot \omega - E_g)^2 \quad (3.5)$$

For metals $E_q = 0$, $\Theta m(\omega)$ has the following expression:

$$\Theta_m(\omega) = cst \times (\hbar \cdot \omega)^2 \quad (3.6)$$

Equation of $k(\omega)$

Knowing the equations of $\Theta(\omega)$, $\Phi(\omega)$ and $\alpha(\omega)$ it is then possible to determine $k(\omega)$:

$$k(\omega) = \frac{c}{2\omega} \alpha(\omega) = \frac{c}{2\omega} \frac{1}{I_0} \Phi(\omega) \Theta(\omega)$$
(4.1)

By neglecting all the second order processes and replacing all $\Theta_{\alpha}(\omega)$ and $\Phi_{\alpha}(\omega)$ by their respective expressions the corresponding extinction coefficient is derived through these equations:

The final equation for the extinction coefficient of amorphous materials is written as:

$$k(E) = \begin{cases} \frac{A \cdot (E - E_g)^2}{E^2 - B \cdot E + C} & ; \text{ for } E > E_g \\ 0 & ; \text{ for } E \le E_g \end{cases}$$
(4.2)

with

$$\begin{vmatrix} A = cst \cdot \left| \left\langle \sigma^* | x | \sigma \right\rangle \right|^2 \times \gamma \\ B = 2 \cdot (E_{\sigma^*} - E_{\sigma}) \\ C = (E_{\sigma^*} - E_{\sigma})^2 + \frac{\hbar^2 \cdot \gamma^2}{4} \end{vmatrix}$$
(4.3)

$$\begin{cases} A = cst \cdot \left| \left\langle \sigma^* | x | \sigma \right\rangle \right|^2 \times \gamma \\ B = 2 \cdot (E_{\sigma^*} - E_{\sigma}) \quad (4.3) \\ C = (E_{\sigma^*} - E_{\sigma})^2 + \frac{\hbar^2 \cdot \gamma^2}{4} \end{cases}$$

The final expression for the extinction coefficient of crystalline materials is written this way:
$$k(E) = \begin{cases} \sum_{j=1}^{N} \frac{A_j \times (E - E_g)^2}{E^2 - B_j \cdot E + C_j} & ; \text{ for } E > E_g \\ 0 & ; \text{ for } E \le E_g \end{cases} \quad (4.4)$$

where A_i , B_i and C_i are given by:

$$\begin{cases} A_{j} = \gamma_{j} \cdot cst \cdot \left| \left\langle \vec{k}_{crit}^{c} | \vec{x} | \vec{k}_{crit}^{v} \right\rangle \right|_{j}^{2} \\ B_{j} = 2 \cdot \left[E_{c} \left(\vec{k}_{crit} \right) - E_{v} \left(\vec{k}_{crit} \right) \right]_{j} \\ C_{j} = \left[E_{c} \left(\vec{k}_{crit} \right) - E_{v} \left(\vec{k}_{crit} \right) \right]_{j}^{2} + \frac{\hbar^{2} \cdot \gamma_{j}^{2}}{4} \end{cases}$$
(4.5)

The final expression for the extinction coefficient of metals is written as:

$$k(E) = \begin{cases} \sum_{j=1}^{N} \frac{A_j \times E^2}{E^2 - B_j \cdot E + C_j} & \text{; for } E > E_g \\ 0 & \text{; for } E \le E_g \end{cases}$$
(4.6)

where A_i , B_i and C_i are given by:

$$\begin{cases} A_{j} = \gamma_{j} \cdot cst \cdot \left| \left\langle \vec{k}_{crit}^{c} | \vec{x} | \vec{k}_{crit}^{v} \right\rangle \right|_{j}^{2} \\ B_{j} = 2 \cdot \left[E_{c} \left(\vec{k}_{crit} \right) - E_{v} \left(\vec{k}_{crit} \right) \right]_{j} \\ C_{j} = \left[E_{c} \left(\vec{k}_{crit} \right) - E_{v} \left(\vec{k}_{crit} \right) \right]_{j}^{2} + \frac{\hbar^{2} \cdot \gamma_{j}^{2}}{4} \end{cases}$$

$$(4.7)$$

Crystalline semiconductors and metals may have several peaks present in the structure of the extinction coefficient which reveals the presence of several oscillators. The number of oscillators determines the name of the dispersion formula as shown below:

Formula	Number of oscillators	Number of parameters
Amorphous	N=1	5
Double Amorphous	N=2	8
Triple Amorphous	N=3	11
Quatre Amorphous	N=4	14

Equation of $n(\omega)$

Then Kramers-Kronig (K-K) relation is used to determine the refractive index through a Hilbert transform of k(E):

$$n(E) - n(\infty) = \frac{1}{\pi} \cdot P \cdot \int_{-\infty}^{+\infty} \frac{k(E') - k(\infty)}{E' - E} dE' \quad (5.1)$$

where P is the Cauchy principal value containing the residuals of the integrand at poles located on the lower half of the complex plane and along the real axis. The derivation of K-K yields the final expression for the refractive index:

In the case of an amorphous material:

$$n(E) = \sqrt{\varepsilon_{\infty}} + \frac{B_0 \cdot E + C_0}{E^2 - B \cdot E + C}$$
(5.2)

where

$$\begin{cases} B_0 = \frac{A}{Q} \cdot \left(\frac{-B^2}{2} + B \cdot E_g - E_g^2 + C\right) \\ C_0 = \frac{A}{Q} \cdot \left(\left(E_g^2 + C\right) \cdot \frac{B}{2} - 2 \cdot E_g \cdot C\right) (5.3) \\ Q = \frac{1}{2} \cdot \sqrt{4 \cdot C - B^2} \end{cases}$$

In the case of a crystalline semiconductors and metals:

$$n(E) = \sqrt{\varepsilon_{\infty}} + \sum_{j=1}^{N} \frac{B_{0j} \cdot E + C_{0j}}{E^2 - B_j \cdot E + C_j}$$
(5.4)

where:

$$\begin{cases} B_{0j} = \frac{A_j}{Q_j} \left(-\frac{B_j^2}{2} + E_g \cdot B_j - E_g^2 + C_j \right) \\ C_{0j} = \frac{A_j}{Q_j} \left(\left(E_g^2 + C_j \right) \cdot \frac{B_j}{2} - 2 \cdot E_g \cdot C_j \right) (5.5) \\ Q_j = \frac{1}{2} \cdot \sqrt{4 \cdot C_j - B_j^2} \end{cases}$$



The parameters of the equation

Parameter describing the refractive index

The term $\varepsilon(\infty)$ is an additional parameter corresponding to the high-frequency dielectric constant. It is at least superior to one and equal to the value of the dielectric function when $\omega \rightarrow \infty$.

At least 4 parameters describe the extinction coefficient.

- A, B, C are positive non zero parameters referring to the electronic structure of the material. For crystalline semiconductors, dielectrics and metals the subscript «*j*» (*j* = 1, 2, 3) for the parameters $A_{j^{*}}$ $B_{j^{*}}C_{j}$ refers to the number of oscillators.
- A_j (in eV) is related to the dipole matrix squared and gives the strength (amplitude) of the extinction coefficient peak. Due to correlation between A and C, the higher A is the higher the absorption peak and the smaller its width C is. Generally, 0<A_j<2.
- $B_j/2$ (in eV) is approximately the energy at which the extinction coefficient is maximum (peak of absorption). As the value of *B* increases the absorption peak is shifted towards the UV region. Generally, $3 < B_j < 30$.
- C_j (in eV²) is related to the broadening term of the absorption peak. It depends on the energy difference between different states and on the lifetime of transition. If there is no lifetime contribution in the expression of *B* then $C_j \ge B^2/4$. The larger C_j is the larger the absorption peak but the smaller its amplitude. Generally, $3 < C_j < 150$.
- E_g (in eV) is the energy band gap. It is equal to the minimum of energy required for a transition from the valence band to the conduction band. It is the energy from which the absorption starts to be non-zero: $k(E \ge E_g) \ge 0$. $E_g < B$ for all cases.

Limitation of the model

The Forouhi-Bloomer dispersion has some drawbacks due to the generation of unphysical results. Thus according to Jellison & Modine (Ref. 4):

- The Forouhi-Bloomer model accuracy is rather bad below the band gap.
- After the fit, several energy gaps in the Forouhi-Bloomer equations might be negative.
- The limit of k(E) → cst as E → ∞ is inconsistent with what is predicted by experiments, that is k(E) → 0.
- Forouhi-Bloomer does not respect time reversal symmetry because k(*E*) ≠ k(-*E*).
- The *B* and *C* parameters are often correlated.

Parameter set up

Note that:

- The graphs below show the different contributions (in red dashed lines) to the imaginary part of the triple Amorphous dielectric function (red bold line).
- The sign « a » before a given parameter means that either the amplitude or the broadening of the peak is linked to that parameter.

Amorphous function



Optical properties of a-Si given by the Amorphous function

Triple Amorphous function



Optical properties of Alq3 given by Triple Amorphous function



Applications to materials

The amorphous model works particularly well for amorphous materials exhibiting an absorption in the visible and/or FUV range (absorbing dielectrics, semiconductors, polymers).

Materials following the Amorphous model

The asterisk * refers to parameters that are negative and thus do not have any physical meaning but represent good starting values to perform the fit on the material.

Materials	લ ૩	А	В	С	Eg	S. R. (eV)
AIN	2.472	0.066	17.670	80.000	0.809	1.5 - 3
AlInGaP	5.521	0.0229	6.714	11.282	2.046	0.75 - 3
a-C	2.275	0.194	7.342	28.634	0.132	1.5 - 3.5
a-GaAs	4.766	0.407	5.872	9.605	1.418	0.6 - 3
a-Si	3.453	0.865	6.703	13.237	0.906	0.6 - 5
a-Si:H	3.710	1.090	6.670	12.720	1.270	0.6 - 5
a-SiGe	1.037	0.63	8.085	19.125	- 0.894*	0.8 - 3
SiN	3.569	0.225	13.491	48.649	4.299	0.6 - 6
TaOx	3.443	0.294	10.925	30.523	3.801	1.4 - 5
TiOx	4.470	0.470	8.300	17.799	3.190	0.6 - 5
WOx	3.962	0.060	7.756	15.175	2.895	1.5 - 4.5
ZnSe	5.521	0.022	6.714	11.282	2.046	0.6 - 3

Materials following the Triple Amorphous model

Materials	લ 3	Eg	A ₁	B ₁	C ₁	A ₂
Alq2	2.360	2.439	0.005	9.384	22.042	0.0162
8						
Materials	B ₂	C ₂	A ₃	B ₃	C ₃	S. R. (eV)
Alq2	6.079	9.293	0.003	11.208	31.444	0.75 - 4

Materials following the Quatre Amorphous model

Materials	લ 3	Eg	A ₁	B ₁
GaAs	4.324	1.253	0.000	5.870
GaP	4.148	0.932	0.002	7.470
Ge	4.879	0.784	0.144	4.526
In G a P	4.996	1.296	0.040	6.681
c-Si	5.664	2.065	0.011	6.860

Materials	C ₁	A ₂	B ₂	C ₂		
GaAs	8.618	0.148	6.248	10.037		
GaP	13.967	0.024	7.946	15.957		
Ge	5.280	0.304	6.789	12.694		
In G a P	11.297	0.164	3.395	3.124		
c-Si	11.778	0.054	7.361	13.625		
Materials	Α ₃	В ₃	C ₃	Α ₄		
GaAs	0.116	9.790	24.412	0.018		
GaP	0.090	10.249	26.699	0.005		
Ge	0.013	8.710	19.046	0.014		
In G a P	0.114	8.033	16.913	0.098		
c-Si	0.153	8.632	18.817	0.047		
Materials	B ₄	C ₄	S. R. (eV)			

Materials	В4	C_4	S. R. (eV)
GaAs	12.450	40.969	0.6 - 3
GaP	12.719	40.528	0.6 - 3
Ge	9.997	25.250	1.5 - 3
In G a P	13.008	39.464	0.6 - 3
c-Si	10.314	26.859	0.6 - 3

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Appendix BPropagation of Polarized Light:Jones and Mueller Formalism

The two most important mathematical descriptions of polarization are the Jones and the Mueller Formalism. The Jones formalism ia applicable only to completely polarized waves, but it can deal with coherent beams. The Mueller formalism instead is applicable also to partially polarized light, however, it can deal only with incoherent beams.

B1 Jones formalism

During a reflection under non normal incidence, the plane of incidence is defined by the incident beam and the normal direction on the sample:



The propagation vector of the light is z, the electromagnetic field can be described by two components, one perpendicular to the plane of incidence (s) and one parallel to the plane of incidence (p). These two axis define an orthogonal basis to describe the light polarization. A representation of the electric field vector is then:

$$\dot{\mathbf{E}} = \mathbf{E}_{s}\vec{\mathbf{s}} + \mathbf{E}_{p}\vec{\mathbf{p}}$$

or written as a *Jones vector*:

$$\vec{\mathrm{E}} = \begin{bmatrix} \mathrm{E}_{\mathrm{s}}(t) \\ \mathrm{E}_{\mathrm{p}}(t) \end{bmatrix}$$

 ${\rm E}_p(\textit{t})$ and ${\rm E}_s(\textit{t})$ are the instantaneous scalar components of $\,E$. By preserving the phase information it is possible to handle coherent waves:

Jones vector for an arbitrarily polarized beam

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$$\vec{\mathbf{E}} = \begin{bmatrix} \mathbf{E}_{0s} e^{i\varphi_s} \\ \mathbf{E}_{0p} e^{i\varphi_p} \end{bmatrix}$$

If for example $E_{op} = E_{0s}$ and $p = {}_{s}$ we get:

$$\vec{\mathrm{E}} = \mathrm{E}_{0\mathrm{p}} e^{i\varphi_{\mathrm{p}}} \begin{bmatrix} 1\\1 \end{bmatrix}$$

Representing a light polarized linearly at 45°.

Since absolute amplitudes and phases are usually not interesting the irradiance can be normalized to unity so that we get:

$$\vec{\mathrm{E}}_{45^{\circ}} = \frac{1}{\sqrt{2}} \begin{bmatrix} 1\\1 \end{bmatrix}$$

and also:

$$\vec{\mathrm{E}}_{p} = \begin{bmatrix} 0\\1 \end{bmatrix}_{\text{and}} \vec{\mathrm{E}}_{s} = \begin{bmatrix} 1\\0 \end{bmatrix}$$

For right-circular light we have $E_{op} = E_{0s}$ and but the s-component is leading the p-component by 90°:

$$\vec{\mathrm{E}}_{\mathrm{R}} = \begin{bmatrix} \mathrm{E}_{0\mathrm{s}} e^{i\varphi\mathrm{s}} \\ \mathrm{E}_{0\mathrm{p}} e^{i(\varphi_{p} - \pi/2)} \end{bmatrix}$$

Dividing by $E_{0s}e^{i\varphi s}$ yields:

$$\begin{bmatrix} 1 \\ e^{-i\pi/2} \end{bmatrix} = \begin{bmatrix} 1 \\ -i \end{bmatrix}$$

So that the normalized Jones-vectors for right-circular and respectively left-circular polarized light are:

$$\vec{\mathrm{E}}_{R} = \frac{1}{\sqrt{2}} \begin{bmatrix} 1\\ -i \end{bmatrix}_{\text{and}} \vec{\mathrm{E}}_{L} = \frac{1}{\sqrt{2}} \begin{bmatrix} 1\\ i \end{bmatrix}$$

One refers to two polarization states as being orthogonal when their Jones vectors are orthogonal. \vec{E}_R and \vec{E}_L form such a set, as do \vec{E}_p and \vec{E}_s .

Any polarization state can be described by a linear combination of the vectors of such an orthonormal set.

An optical element transforms an incident beam represented by \vec{E}_i into a transmitted beam \vec{E}_t . This can be described now mathematically using a 2 x 2 matrix:

$$\vec{\mathrm{E}}_{t} = \begin{bmatrix} a_{11} & a_{12} \\ a_{21} & a_{22} \end{bmatrix} \vec{\mathrm{E}}_{t} = A\vec{\mathrm{E}}_{t}$$

With the **Jones matrix** A. The effect of n optical elements can then simply be determined by matrix multiplication:

$$\vec{\mathbf{E}}_t = A_n \cdots A_2 A_1 \vec{\mathbf{E}}_t$$

Examples for Jones matrices:

Horizontal linear polarizer:

	$\begin{bmatrix} 1 & 0 \\ 0 & 0 \end{bmatrix}$
Vertical linear polarizer:	$\begin{bmatrix} 0 & 0 \\ 0 & 1 \end{bmatrix}$
Linear polarizer at 45°:	$\begin{bmatrix} 0 & 1 \end{bmatrix}$
Linear polarizer at -45°:	$2\begin{bmatrix} 1 & 1 \end{bmatrix}$ $\underline{1}\begin{bmatrix} 1 & -1 \end{bmatrix}$
QWP, fast axis vertical:	$2 \begin{bmatrix} -1 & 1 \end{bmatrix}$ $e^{i\pi/4} \begin{bmatrix} 1 & 0 \end{bmatrix}$
QWP, fast axis horizontal:	$\begin{bmatrix} 0 & -i \end{bmatrix}$
Circular polarizer right:	$\begin{bmatrix} e & \\ 0 & i \end{bmatrix}$ $1\begin{bmatrix} 1 & i \end{bmatrix}$
Circular polarizer left:	$\frac{1}{2}\begin{bmatrix} -i & 1 \end{bmatrix}$
	$\frac{1}{2} \begin{bmatrix} 1 & -i \\ i & 1 \end{bmatrix}$

B2 Mueller formalism

Stokes vector for an arbitrarily polarized beam

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All possible states of polarization, including partial polarization, can be described by a set of three real numbers. If we include the total light intensity we have four real, measurable quantities, the *Stokes parameters* S_0 , S_1 , S_2 , S_3 forming the *Stokes vector*:

$$\vec{\mathbf{S}} = \begin{bmatrix} S_0 \\ S_1 \\ S_2 \\ S_3 \end{bmatrix} = \begin{bmatrix} I_0 \\ I_x - I_y \\ I_{\pi/4} - I_{-\pi/4} \\ I_R - I_L \end{bmatrix}$$



where:

 $I_0 =$ total light intensity

 $I_x =$ intensity through a linear polarizer at 0°

 $I_v =$ intensity through a linear polarizer at 90°

 $I_{\pi/4}$ = intensity through a linear polarizer at 45°

 $I_{-\pi/4}$ = intensity through a linear polarizer at -45°

 I_R = intensity through a right circular polarizer

 I_L = intensity through a left circular polarizer

and $I_0 = I_x + I_y = I_{\pi/4} + I_{-\pi/4} = I_R + I_L$.

It is often convenient to normalize the Stokes parameters by dividing each one by the S_0 . The set of parameters for natural light is then:



Examples for Stokes vectors:



For completely polarized light we must have:

$$S_0^2 = S_1^2 + S_2^2 + S_3^2$$

Furthermore, for partially polarized light we can now calculate the degree of polarization by:

$$Deg_{Pol} = \sqrt{S_1^2 + S_2^2 + S_3^2} / S_0$$

Also in the Mueller formalism we have a matrix method to deal with the vectors representing the state of polarization. These Mueller, $4 \ge 4$ matrices are applied in much the same way as are the Jones matrices:

with

$$\vec{\mathbf{S}}_t = M \vec{\mathbf{S}}_i$$

$$M = \begin{bmatrix} M_{11} & M_{12} & M_{13} & M_{14} \\ M_{21} & M_{22} & M_{23} & M_{24} \\ M_{31} & M_{32} & M_{33} & M_{34} \\ M_{41} & M_{42} & M_{43} & M_{44} \end{bmatrix}$$

Examples for Mueller matrices:

Horizontal linear polarizer:

Vertical linear polarizer:

		1	-1	0	0
	1	-1	1	0	0
	2	0	0	0	0
		0	0	0	0_
arizer at 45°.					

Linear polarizer at 45°:

Linear polarizer at -45°:

$$\frac{1}{2} \begin{bmatrix} 1 & 0 & -1 & 0 \\ 0 & 0 & 0 & 0 \\ -1 & 0 & 1 & 0 \\ 0 & 0 & 0 & 0 \end{bmatrix}$$

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QWP, fast axis vertical:

	[1	0	0	0	
	0	1	0	0	
	0	0	0	-1	
	0	0	1	0	
QWP, fast axis horizont	al:				
	∏ 1	0	0	0	
	0	1	0	0	
	0	0	0	1	
	0	0	-1	0	
Circular polarizer right:					
	L L	1 () 0	1]
	1) () 0	0	
	$\overline{2}$) () 0	0	
		1 () 0	1	
Circular polarizer left:	_				
	[1	() 0	_	1
	1 0	() 0	0	
	$\overline{2} \mid 0$	() 0	0	
		1 () 0	1	

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Appendix C Deduction of Psi and Delta from raw data

The measured intensity of a phase-modulated ellipsometer can be written as:

$$I(t) = I \left[I_0 + I_s \sin(\delta_0 + A_m \sin \omega t) + I_c \cos(\delta_0 + A_m \sin \omega t) \right]$$
(1)

with the amplitude of modulation A_m and the frequency of modulation ω . *I* is a pre-factor (lamp intensity, sample reflectivity, transmission of optical elements), which is omitted during later calculations. δ_0 is the rest-birefringence of the modulator.

The factors I_0 , I_s and I_c contain the azimuth of the polarizer P, the modulator M and the analyzer A in respect to the plane of incidence and the ellipsometric angles ψ and Δ :

$$I_0 = 1 + \cos 2A \cos 2M \cos 2(P - M) - \{\cos 2A + \cos 2M \cos 2(P - M)\} \cos 2\psi$$
(1a)
$$I_s = \sin 2(P - M) \sin 2A \sin 2\psi \sin \Delta$$
(1b)

 $I_{c} = \sin 2(P - M) \{ \sin 2M (\cos 2\psi - \cos 2A) + \sin 2A \cos 2M \sin 2\psi \cos \Delta \}$ (1c)

We make the following transformation:

$$\sin(\delta_{\theta} + A_m \sin \omega t) = \sin \delta_{\theta} \cos(A_m \sin \omega t) + \cos \delta_{\theta} \sin(A_m \sin \omega t)$$

(and analogous for the cos-Term)

and the following series expansion:

$$\sin(A_m \sin \omega t) = 2 J_1(A_m) \sin \omega t + \dots$$
$$\cos(A_m \sin \omega t) = J_0(A_m) + 2 J_2(A_m) \cos 2\omega t + \dots$$

whereby $J_n(\mathcal{A}_m)$ is the n-th Bessel-function of A_m .

these terms will be used again in equation (1).

The detected signal can as well be written as:

$$S(t) = S_0 + S_1 e^{it} + S_2 e^{2it} + \dots$$
(2)

Where again ω = modulation frequency of the photoelastic modulator. By Fourier-analysis we get the coefficients S_0 , S_1 and S_2 . A comparison of equations (1) and (2) yields:

$$S_0 = I \left[I_0 + J_0(\mathcal{A}_m) \left(I_c \cos \delta_0 + I_s \sin \delta_0 \right) \right]$$
(3a)

$$S_1 = I \left[2 J_1(A_m) \left(I_s \cos \delta_0 - I_\iota \sin \delta_0 \right) \right]$$
(3b)

$$S_2 = I \left[2 J_2(A_m) \left(I_c \cos \delta_0 + I_s \sin \delta_0 \right) \right]$$
(3c).


and further by normalization

$$S_{\omega} \equiv \frac{S_1}{S_0} \tag{4a}$$

and

$$S_{2\omega} \equiv \frac{S_2}{S_0}$$
(4b)

Is and Ic can now be calculated by inversion of the equations from S and S₂ (whereby $I_0 = 1$ in the given configurations). The values for A_m and δ_0 are determined during the calibration of the system.

$$Is = \frac{\frac{S_{\omega}}{2T_{1}J_{1}(A_{m})}\cos\delta_{0} + \frac{S_{2\omega}}{2T_{2}J_{2}(A_{m})}\sin\delta_{0}}{1 - J_{0}(A_{m})\frac{S_{2\omega}}{2T_{2}J_{2}(A_{m})}}$$
$$Ic = \frac{\frac{S_{2\omega}}{2T_{2}J_{2}(A_{m})}\cos\delta_{0} - \frac{S_{\omega}}{2T_{1}J_{1}(A_{m})}\sin\delta_{0}}{1 - J_{0}(A_{m})\frac{S_{2\omega}}{2T_{2}J_{2}(A_{m})}}$$

 $(T_1 \text{ and } T_2 \text{ the norms of the electronic transfer functions of the system for each harmonic)}$

All the measurement is funded on the determination of the values I_s and I_c , which are directly proportional to the ellipsometric angles and .

Summary: determination of the ellipsometric angles and from the detected signal:



Examples for the last step:

configuration II (for example P = 45°, M = 0°, A = 45°): $I_0 = 1$ $I_s = \sin 2 \sin 1$ $I_c = \sin 2 \cos 1$

configuration III (for example P = 90°, M = 45°, A = 45°): I₀ = 1 I_s = sin2 sin I_{c'} = cos2

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Measuring in these two configurations we already can determine a polarization factor for isotropic samples:

 $P = I_S + I_C + I_C'$

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Appendix DDeduction of Mueller Matrix andJones Matrix elements fromUVISEL Data

D1 UVISEL data and Mueller matrix elements

The relations for the intensities I_0 , I_s and I_c of a phase-modulated ellipsometer in the configuration ASMP (analyzer - sample - modulator - polarizer) with a sample represented by the corresponding Mueller-Matrix M are:

$$\begin{split} &I_0 = M_{11} + M_{12}\cos 2A + M_{13}\sin 2A + \cos 2(P-M) \times \\ &\times \left[\cos 2M \left(M_{21} + M_{22}\cos 2A + M_{23}\sin 2A\right) + \sin 2M \left(M_{31} + M_{32}\cos 2A + M_{33}\sin 2A\right)\right] \end{split}$$

 $I_s = sin2(P-M) (M_{41} + M_{42} cos2A + M_{43} sin2A)$

$$\begin{split} &I_{c} = \sin 2(\text{P-M}) \times \\ &\times \left[\cos 2M \left(M_{31} + M_{32} \cos 2A + M_{33} \sin 2A\right) - \sin 2M \left(M_{21} + M_{22} \cos 2A + M_{23} \sin 2A\right)\right] \end{split}$$

whereby M_{ij} (i, j = 1, 2, 3, 4) the elements of the Mueller-Matrix of the sample.

P = azimuth of polarizerM = azimuth of modulatorA = azimuth of analyzer

In the following we show the expressions for the most important configurations: For a Polarizer setting of $+45^{\circ}$ (P-M = $+45^{\circ}$) we get the following simplified expressions:

$$\begin{split} I_0 &= M_{11} + M_{12}\cos 2A + M_{13}\sin 2A \\ I_s &= M_{41} + M_{42}\cos 2A + M_{43}\sin 2A \\ I_c &= \cos 2M\left(M_{31} + M_{32}\cos 2A + M_{33}\sin 2A\right) - \sin 2M\left(M_{21} + M_{22}\cos 2A + M_{23}\sin 2A\right) \end{split}$$

and further:

for the "alignment" configuration (M = 0° or 90°, A = 0° or 90°) : $I_0 = M_{11} + (-1)^A M_{12}$ $I_s = M_{41} + (-1)^A M_{42}$ $I_c = (-1)^M [M_{31} + (-1)^A M_{32}]$

for the "calibration" configuration I (M = ±45°, A = 0° or 90°) : $\rm I_0 = M_{11} + (-1)^A \ M_{12}$

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$$\begin{split} {\rm I}_{\rm s} &= {\rm M}_{41} + (\text{-1})^{\rm A} \; {\rm M}_{42} \\ {\rm I}_{\rm c} &= \text{-} (\text{-1})^{\rm M} \; [{\rm M}_{21} + (\text{-1})^{\rm A} \; {\rm M}_{22}] \end{split}$$

for the "measurement" configuration II (M = 0° or 90° , A = $\pm 45^{\circ}$):

$$\begin{split} \mathbf{I}_0 &= \mathbf{M}_{11} + (\text{-1})^{\mathrm{A}} \, \mathbf{M}_{13} \\ \mathbf{I}_{\mathrm{s}} &= \mathbf{M}_{41} + (\text{-1})^{\mathrm{A}} \, \mathbf{M}_{43} \\ \mathbf{I}_{\mathrm{c}} &= (\text{-1})^{\mathrm{M}} \, [\mathbf{M}_{31} + (\text{-1})^{\mathrm{A}} \, \mathbf{M}_{33}] \end{split}$$

for the "measurement" configuration III (M = ±45°, A = ±45°) : $I_0 = M_{11} + (-1)^A M_{13}$ $I_s = M_{41} + (-1)^A M_{43}$ $I_c = - (-1)^M [M_{21} + (-1)^A M_{23}].$

(the factor $(-1)^{E} = +1$ for an element (A or M) with azimuth $E = 0^{\circ}$ or $+45^{\circ}$ and $(-1)^{E} = -1$ for $E = 90^{\circ}$ or -45°).

D2 Determination of the Mueller matrix elements

We measure in the following 8 configurations:

1. M = -45	A = -45	Ic = A	
2. M = -45°	A = 45	Ic = B	
3. M = 0	A = - 45	Ic = C	Is = D
4. M = 0	A = 45	Ic = E	Is = F
5. M = -45°	$\mathbf{A} = 0$	Ic = G	
6. M = 0	A = 90	Ic = H	Is = I
7. $M = 0$	$\mathbf{A} = 0$	Ic = J	
8. M = -45	A = 90	Ic = L	

in matrix formalism that means:

(0	A	1	0	-1	0	0	0	0	0	0		(M12)		(A)
0	-B	1	0	1	0	0	0	0	0	0		<i>M</i> 13		B
0	С	0	0	0	1	0	-1	0	0	0		<i>M</i> 21		<i>C</i>
0	D	0	0	0	0	0	0	1	0	-1		M22		D
0	-E	0	0	0	1	0	1	0	0	0		M23		E
0	-F	0	0	0	0	0	0	1	0	1	×	<i>M</i> 31	=	F
- G	0	1	1	0	0	0	0	0	0	0		M32		G
H	0	0	0	0	1	-1	0	0	0	0		M33		H
Ι	0	0	0	0	0	0	0	1	-1	0		<i>M</i> 41		Ι
-J	0	0	0	0	1	1	0	0	0	0		M42		J
L	0	1	-1	0	0	0	0	0	0	0)		(M43)		(L)

So that we can determine the following elements of the normalized (M11=1) mueller matrix: M12 ; M13 ; M21 ; M22 ; M23 ; M31 ; M32 ; M33 ; M41 ; M42 ; M43.

$$M_{UVISEL} = \begin{bmatrix} 1 & M_{12} & M_{13} & - \\ M_{21} & M_{22} & M_{23} & - \\ M_{31} & M_{32} & M_{33} & - \\ M_{41} & M_{42} & M_{43} & - \end{bmatrix}$$

Note that if one inverts the direction of light (i.e. PMSA instead of ASMP) t would be possible to measure M^T and thus further 3 elements: M_{14} , M_{24} and M_{34} . This means that theoretically it is possible to measure 14 of the 15 normalized elements of the Mueller matrix. Only M_{44} cannot be determined!

D3 Deduction of Jones matrix elements from the Mueller matrix elements

Non-depolarizing Mueller matrices are transformed into Jones matrices using the following relations:

Jones matrix:
$$J = \begin{bmatrix} j_{xx} & j_{xy} \\ j_{yx} & j_{yy} \end{bmatrix} = \begin{bmatrix} \rho_{xx} e^{i\phi_{xx}} & \rho_{xy} e^{i\phi_{xy}} \\ \rho_{yx} e^{i\phi_{yx}} & \rho_{yy} e^{i\phi_{yy}} \end{bmatrix}$$

where the amplitudes are:

$$\rho_{xx} = \frac{1}{\sqrt{2}} \sqrt{M_{11} + M_{12} + M_{21} + M_{22}}$$
$$\rho_{xy} = \frac{1}{\sqrt{2}} \sqrt{M_{11} + M_{12} + M_{21} - M_{22}}$$
$$\rho_{yx} = \frac{1}{\sqrt{2}} \sqrt{M_{11} + M_{12} - M_{21} - M_{22}}$$



$$\rho_{yy} = \frac{1}{\sqrt{2}} \sqrt{M_{11} - M_{12} - M_{21} + M_{22}}$$

and the relative phases are:

$$\phi_{xy} - \phi_{xx} = \arctan\left(\frac{-M_{14} - M_{24}}{M_{13} + M_{23}}\right)$$
$$\phi_{yx} - \phi_{xx} = \arctan\left(\frac{M_{41} + M_{42}}{M_{41} + M_{32}}\right)$$
$$\phi_{yy} - \phi_{xx} = \arctan\left(\frac{M_{43} - M_{34}}{M_{33} + M_{43}}\right)$$

The phase ϕ_{xx} is not determined; it represents the absolute phase relative to which the other phases are determined.



